

# Novel portable Paris-Edinburgh presses for synchrotron time-resolved 3-D micro- imaging under extreme conditions

Y. Le Godec, E. Boulard, G. Bromiley, N. Guignot, G. Hamel, J.P. Itié, A. King,  
M. Mezouar, G. Morard, J.P. Perrillat, J. Phillippe  
etc.....

**IMPMC**

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(Sorbonne University, Paris, France)*

Y. Le Godec 30'

## Intro Paris-Edinburgh-press

- state-of-the-art
- Recent developments for tomography experiments (Rotopec)



Brief introduction of  
the Paris-Edinburgh  
press



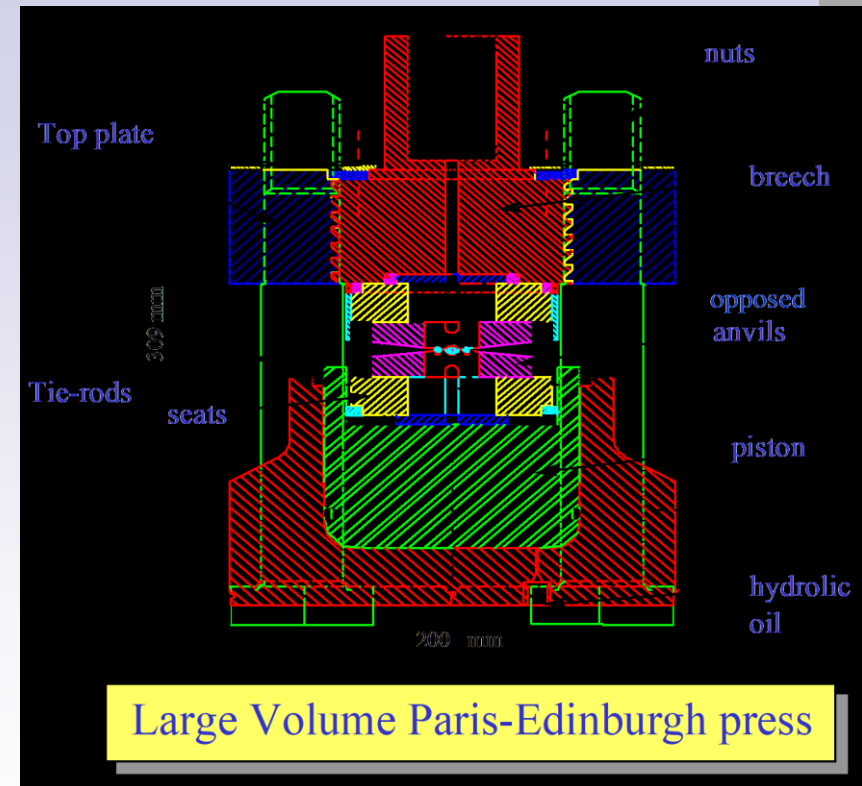
# Paris-Edinburgh press

Designed by *J.M. Besson*, *G. Hamel* and *G. Weill* (1991) in *France*

an alternative, in some cases, to the usual *very big large volume presses* developed in the other countries



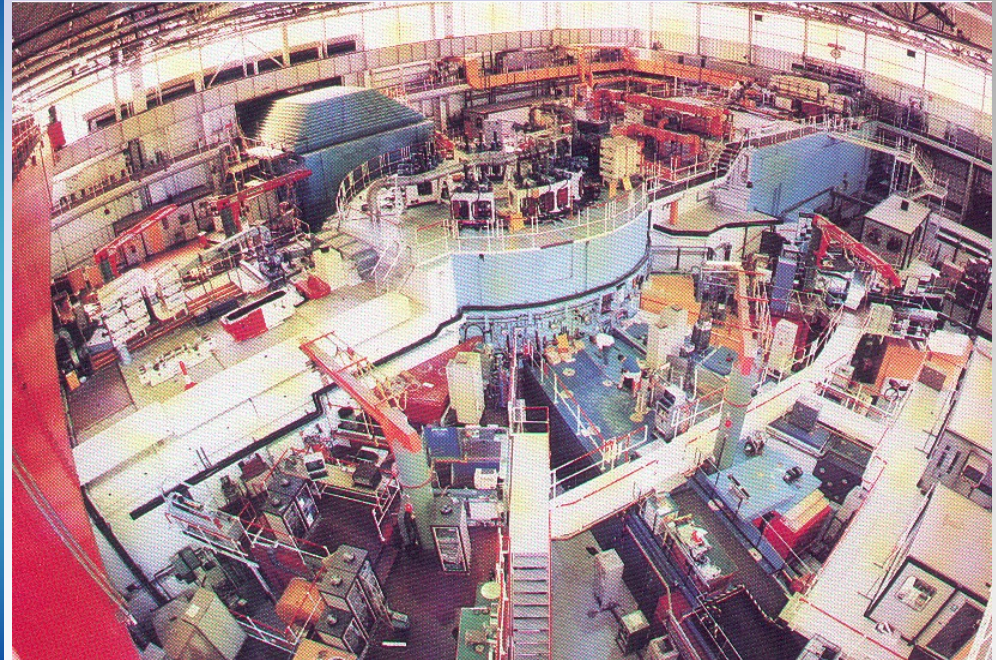
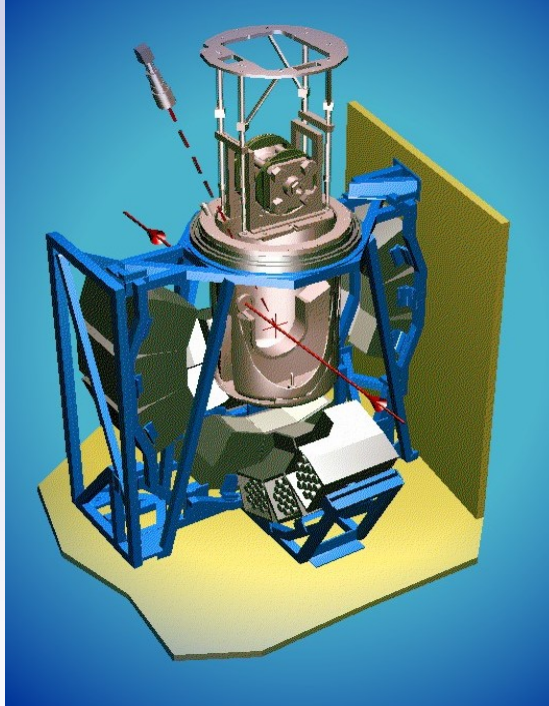
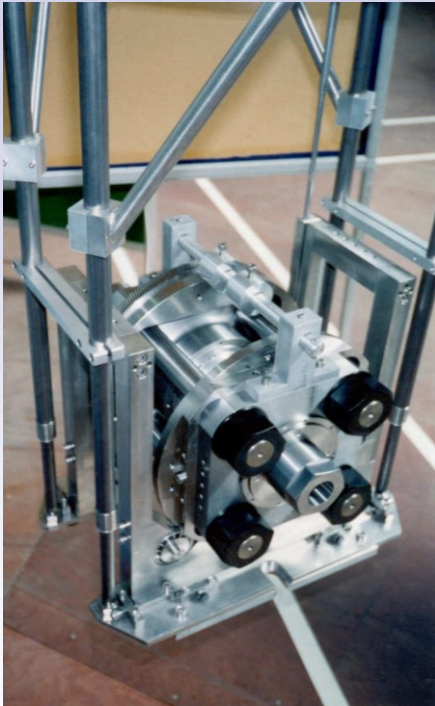
← Correct scale



Its originality comes from its *hydraulic ram*, which is a key breakthrough. It has been very carefully designed (by finite element calculations) to minimize both *size* and *weight*. The result is a ram with a *250 tonnes capacity* which weighs only *50 kilogrammes* and fits into a *30 cm cube*



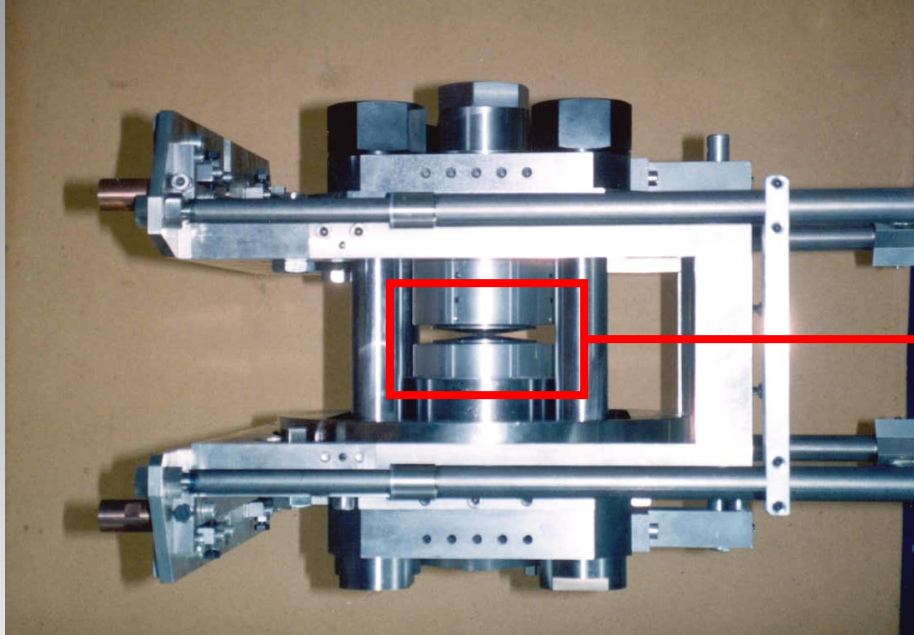
# Originally developed for time-of-flight neutron scattering RT experiments at ISIS



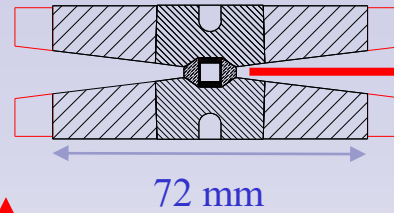
J.M. Besson, R.J. Nelmes, G. Hamel, J.S. Loveday, G. Weill and  
S. Hull, *Physica (Amsterdam)* **180&181B**, 907 (1992)



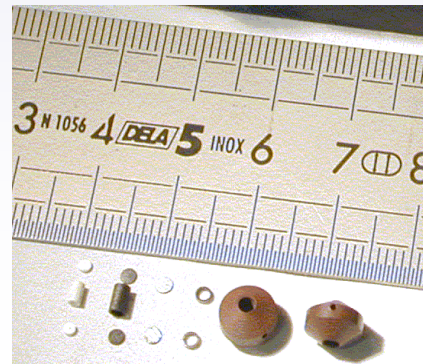
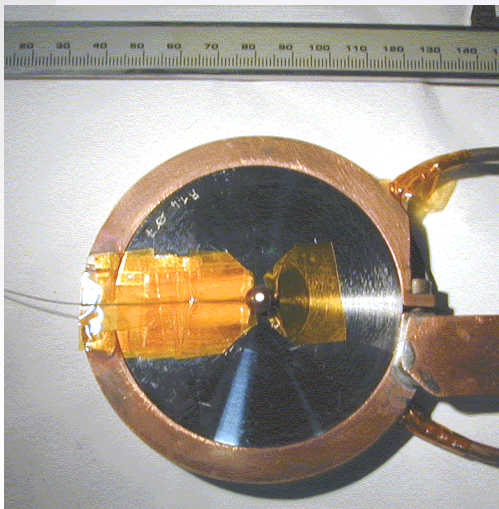
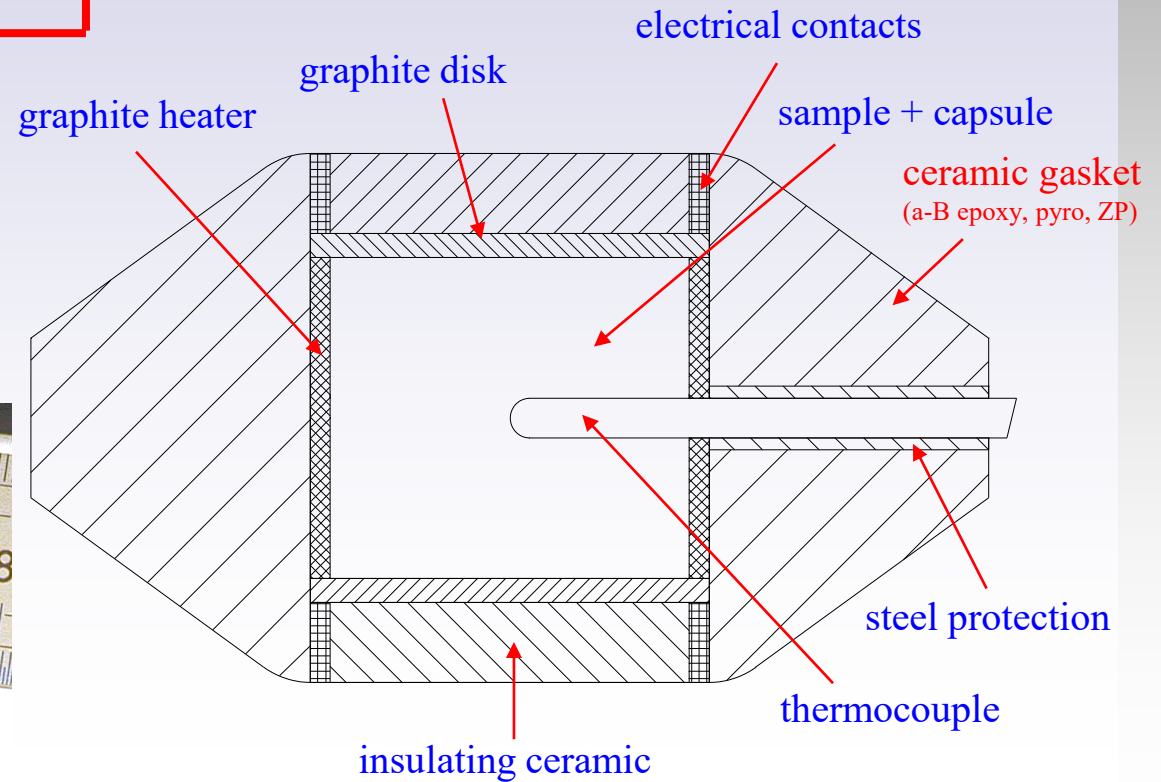
# This press has been adapted later for **high (P,T)** measurements



typical set-up



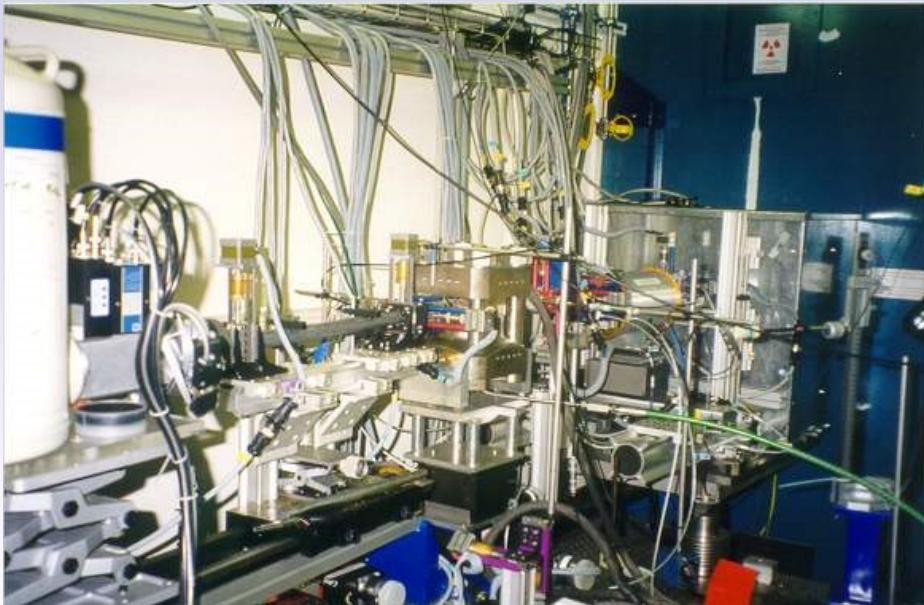
*Pseudo-conoidal profile for anvils*  
*ceramic gasket*



Various parts of the set-up

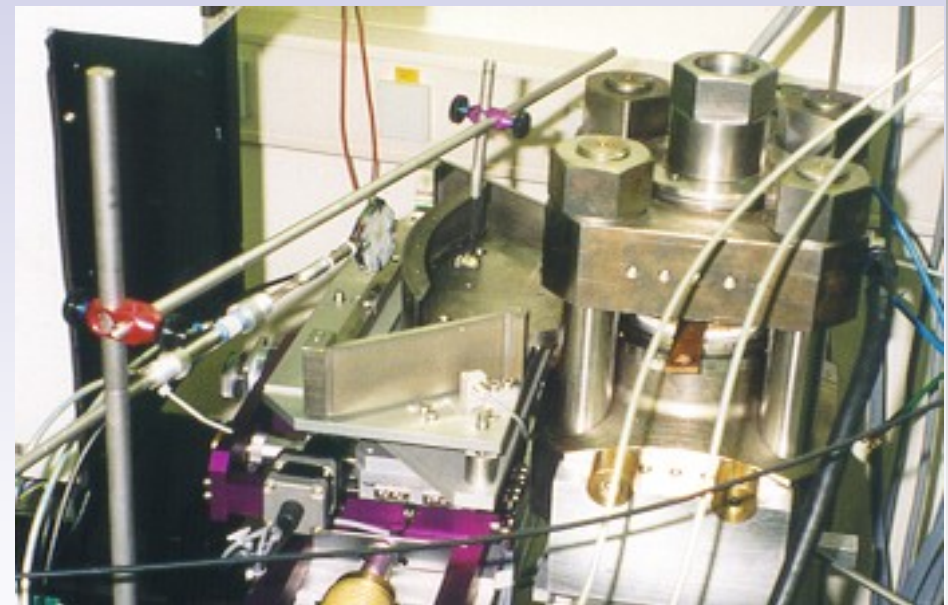
This basic design of the PE press with internal heating has been adapted for a wide range of *in situ* high (P,T) measurements :

## X-ray diffraction (LURE and ID30, ID27, ESRF)



### Energy-dispersive mode

**P. Grima**, A. Polian, M. Gauthier, J.P. Itié, M. Mezouar, G. Weill and J.M. Besson *J. Chem. Solids.* **56**, 525 (1995)



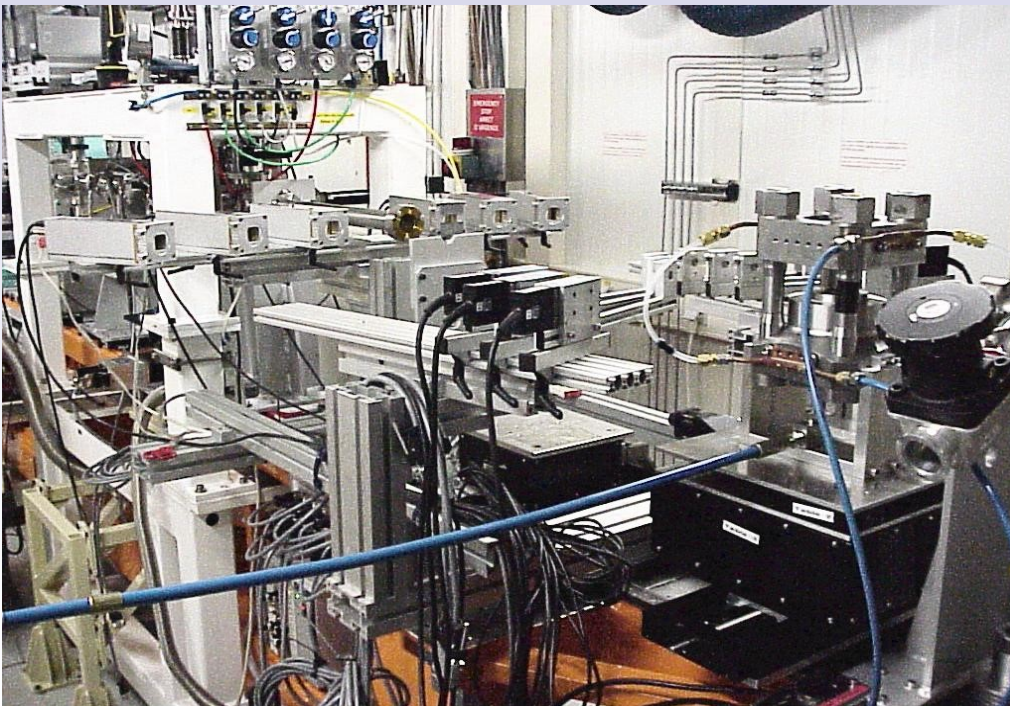
### Angle-dispersive mode

**M. Mezouar**, T. Le Bihan, H. Libotte, Y. Le Godec and D. Häusermann, *Journal of Synch. Rad.* **6**, 1115 (1999)

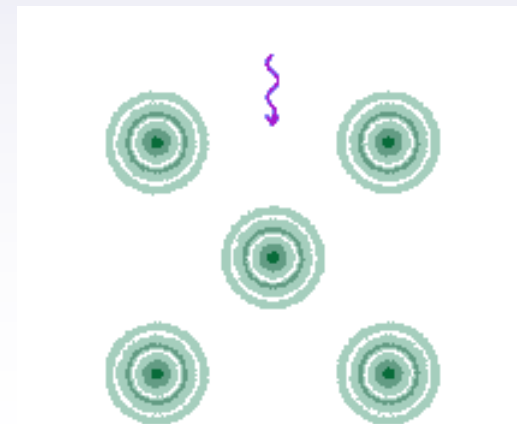


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## EXAFS (BM29, ESRF)

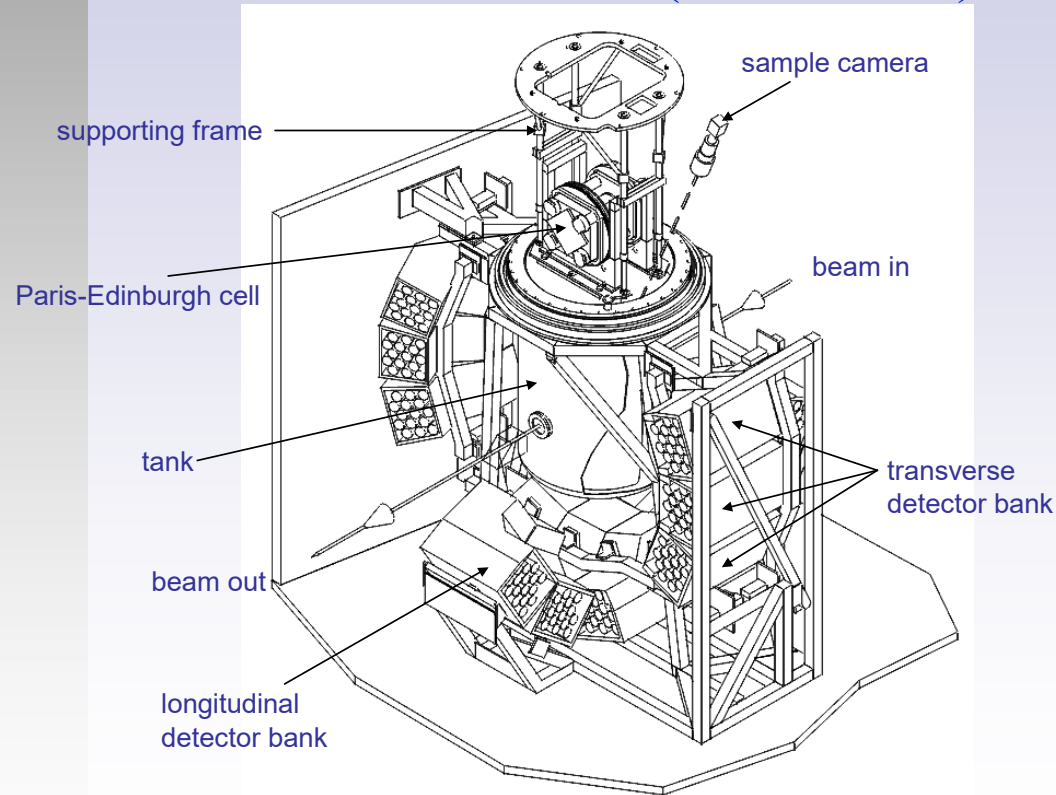


Y. Katayama, M. Mezouar, J.P. Itié,  
J. M. Besson, G. Syfosse, P. Le Fevre,  
and A. Di Cicco,  
*J. Physique IV, Colloq. 7, C2-1011*  
(1997)



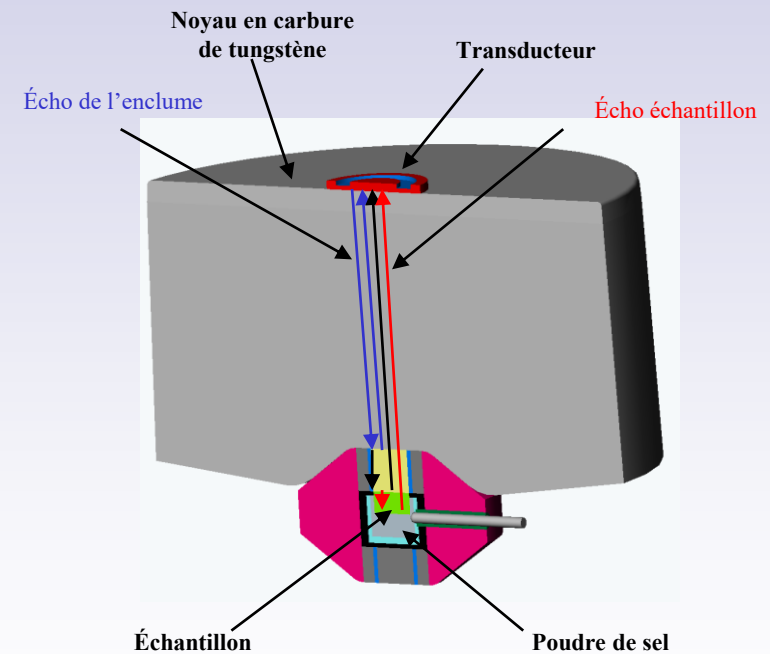
This basic design of the PE press with internal heating has been adapted for a wide range of *in situ* high (P,T) measurements :

## Neutron diffraction (Pearl, ISIS)



Y. Le Godec, et al *High Pres. Res.*, **21**, 263. (2001)

## Ultrasonic studies (IMPMP)



R. Debord, D. Leguillon, G. Syfosse, M. Fisher, *High Pres. Res.*, **23**(4), 451. (2003)

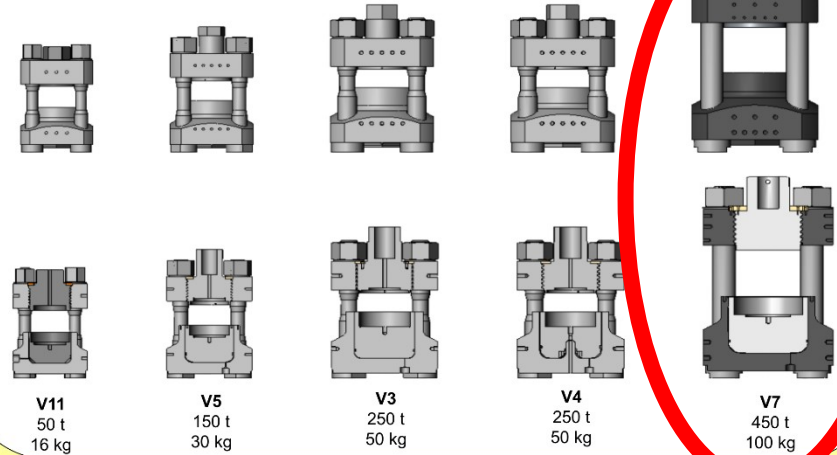




**Type V :**

**4 colomns**

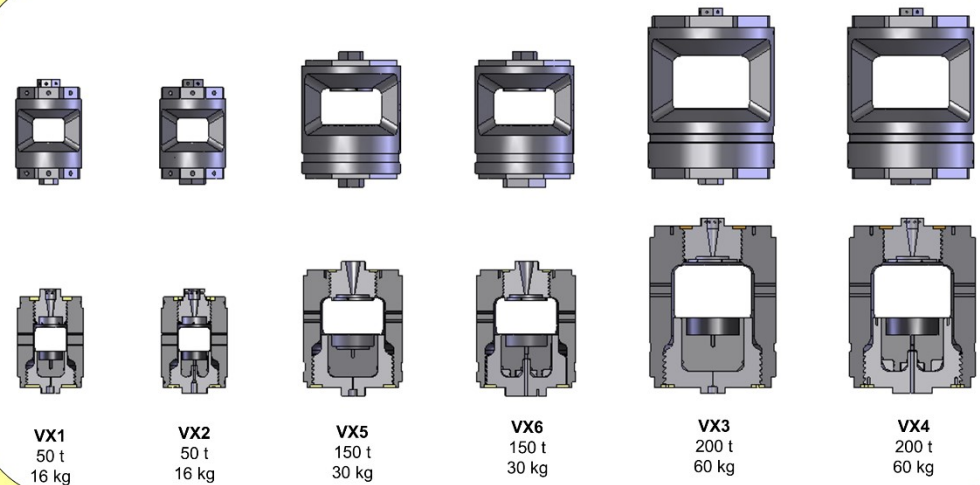
*J.M. Besson et al., Physica  
180&181B, 907 (1992).*



**Type VX :**

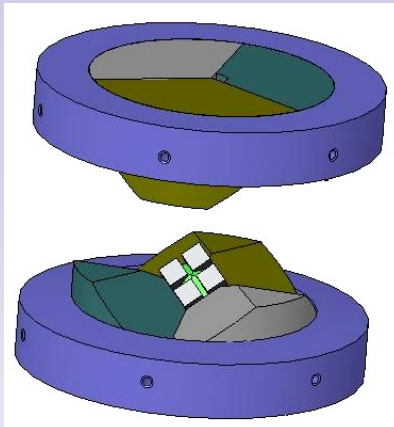
**2 colomns**

*S. Klotz, et al. High Press.  
Res., 24, 219–223 (2004).*

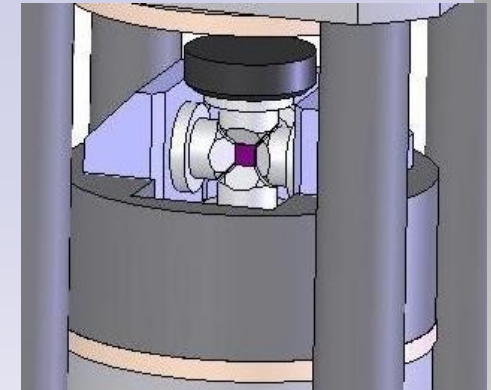
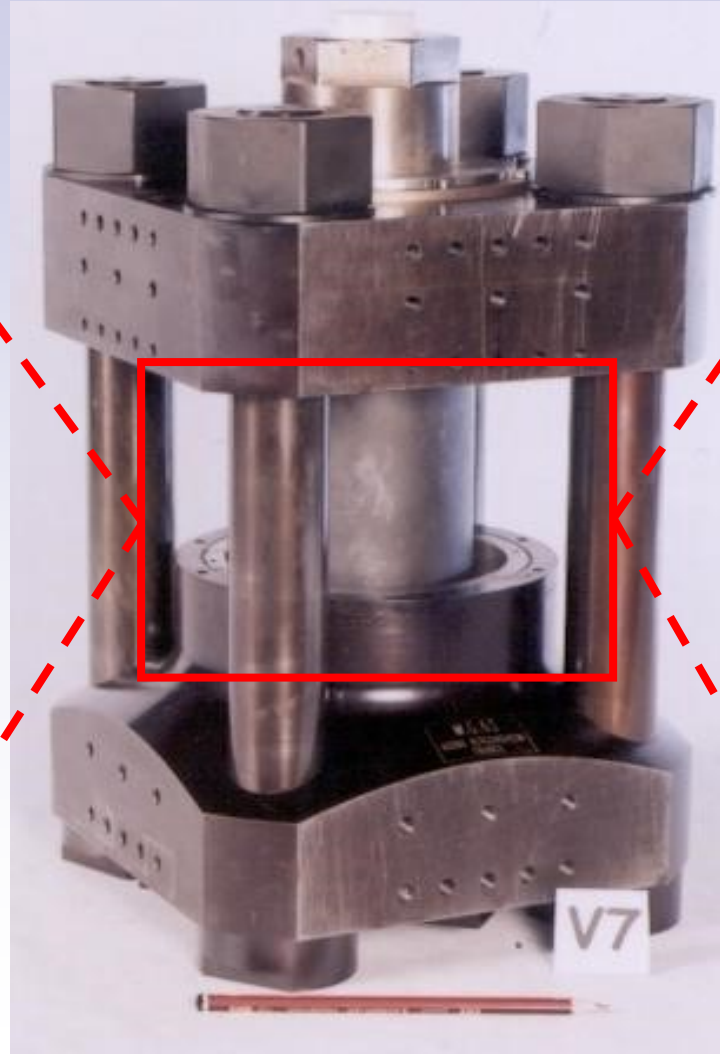


To show the modularity of this press to various techniques, I will continue this talk with the V7, the biggest press of the PE Press family

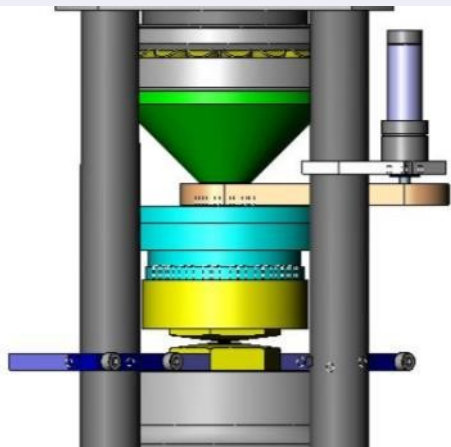
During the last years we developed various modules that can be easily and quickly adapted into the V7/PE press in order to perform various *in situ* or *ex situ* experiments



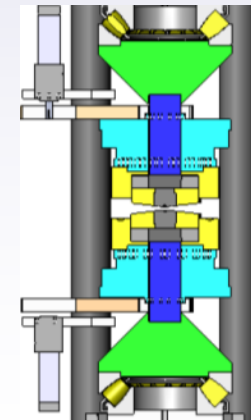
T-cup



DIA



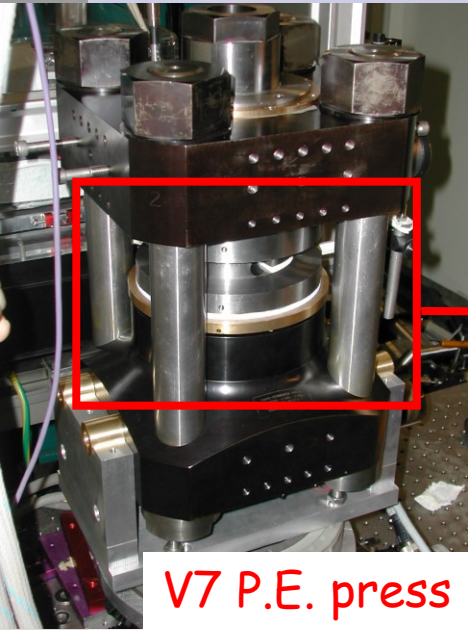
roPEc



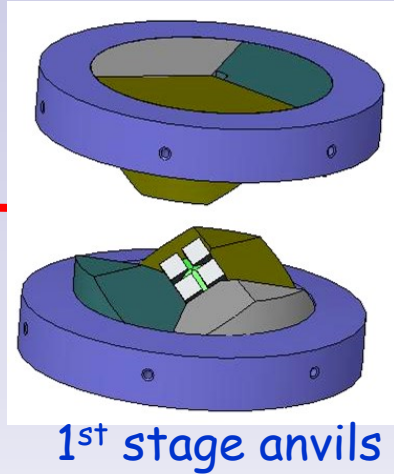
Ro-tomography



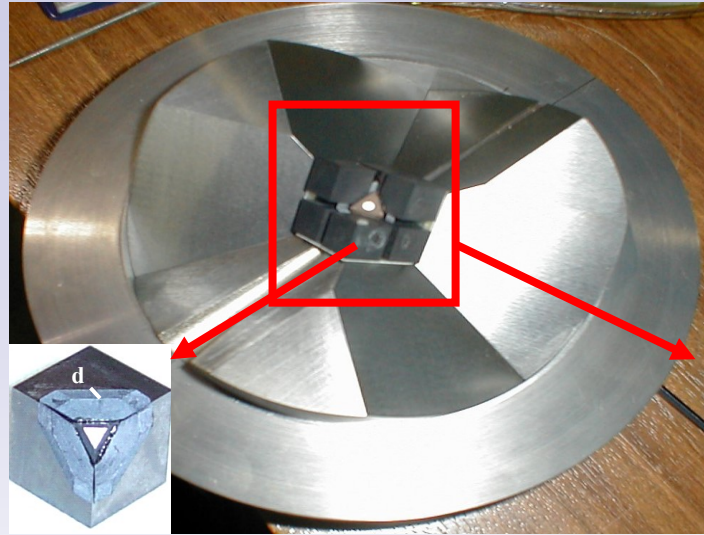
# The PE press with T-cup module



V7 P.E. press



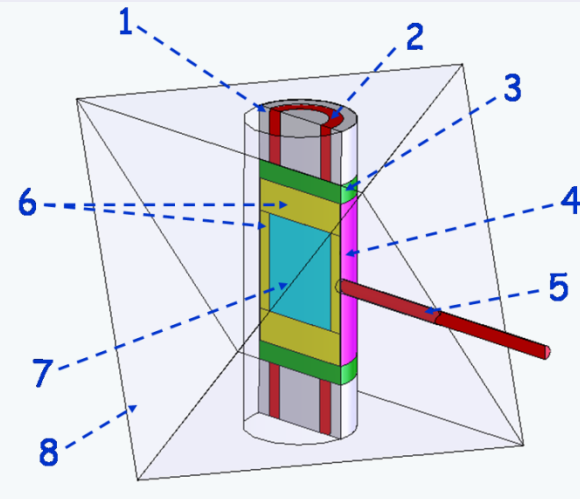
1<sup>st</sup> stage anvils



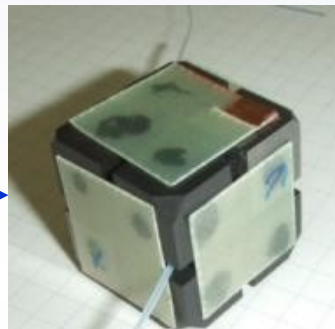
a miniaturized 6-8 KAWAI-type LV device similar to the Stony Brook "T-cup system" which operates routinely to 25 GPa and 2500 K



The second stage consists of **eight c-BN cubes of 10 mm** edge length with a **2 mm truncation** separated by **gaskets**

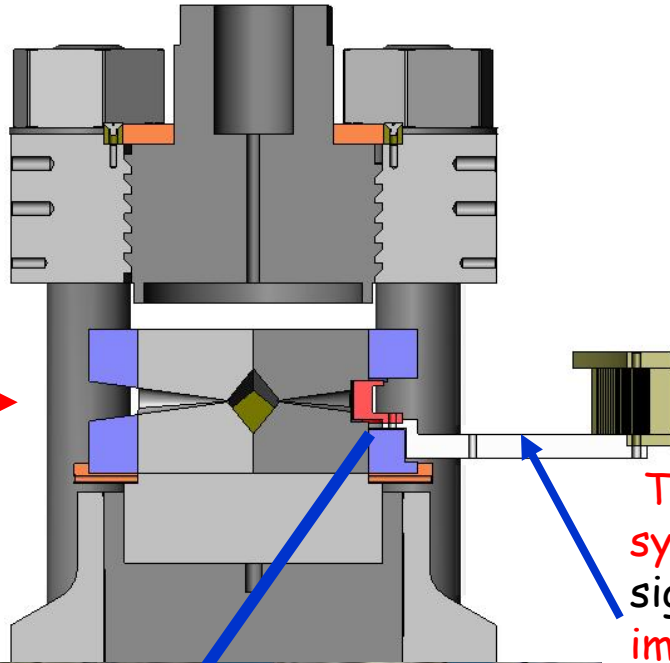
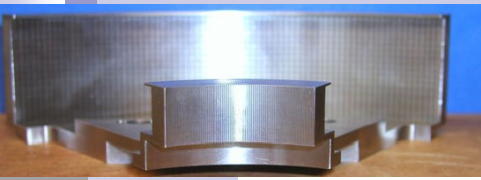
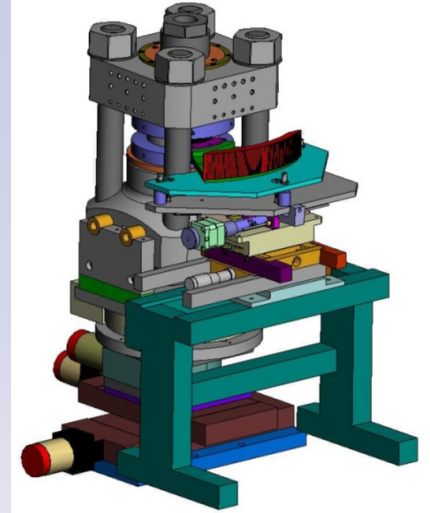


The high pressure set-up is hence a **7 mm amorphous boron epoxy octahedron**



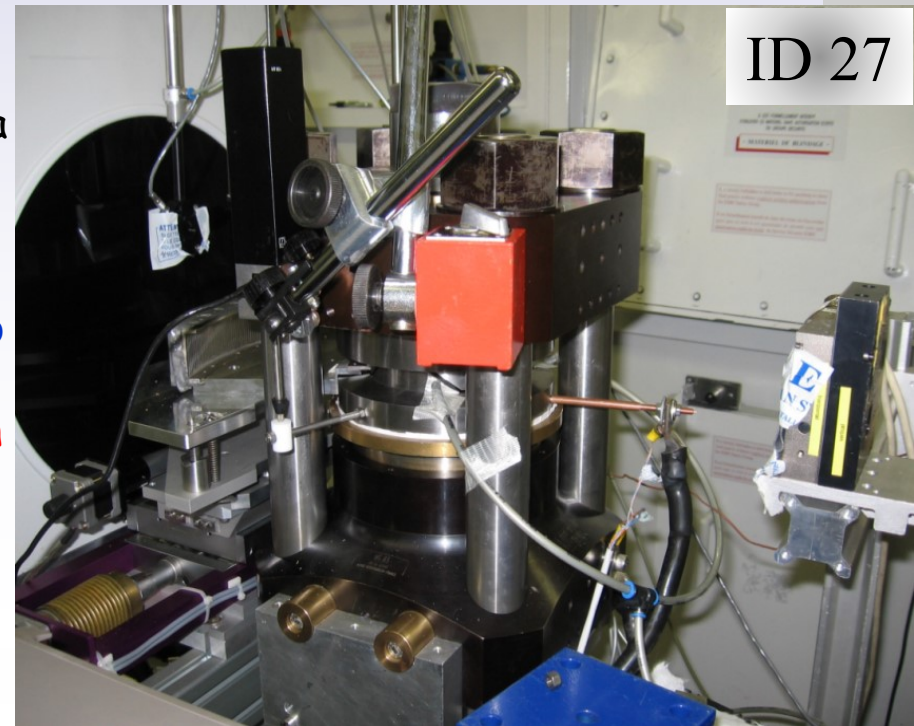
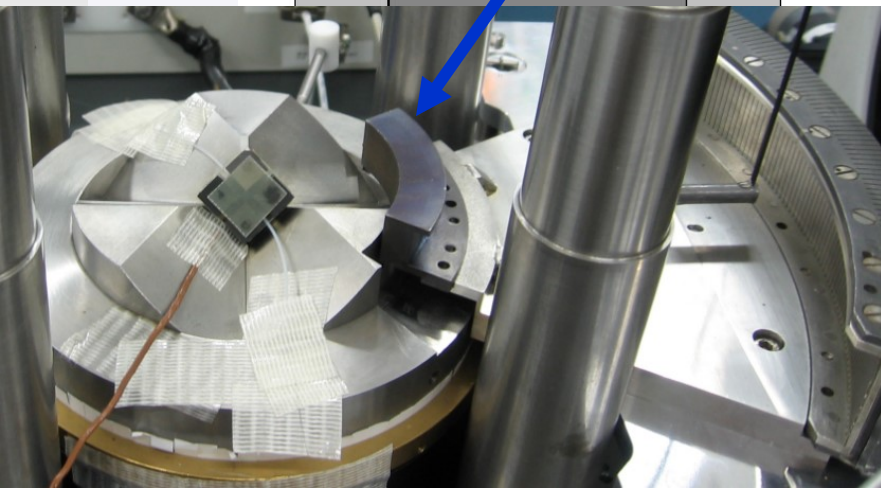
# The PE press with T-cup module

The novel feature of this LV press is its **extreme stiffness** which permits to **update optimised Sollers** and **large area CCD detectors** for **angle dispersive diffraction** at ESRF-ID27



**Incident X-ray beam**

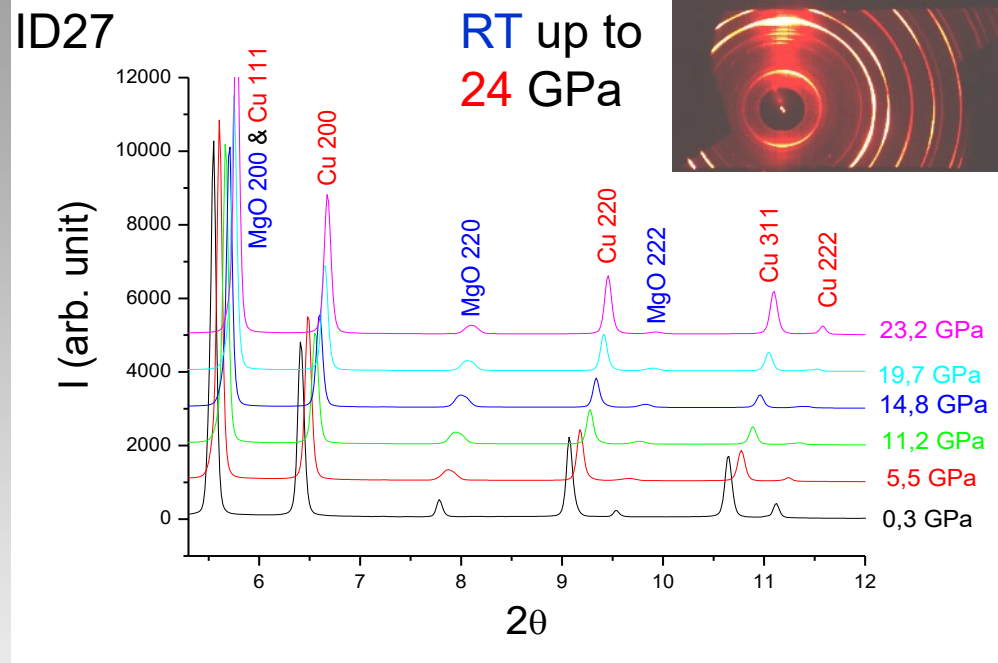
This **Soller slit system** permits a significant **improvement** of the **signal-to-background ratio** and provides **clean diffraction patterns** of **crystalline, amorphous and even liquid samples**



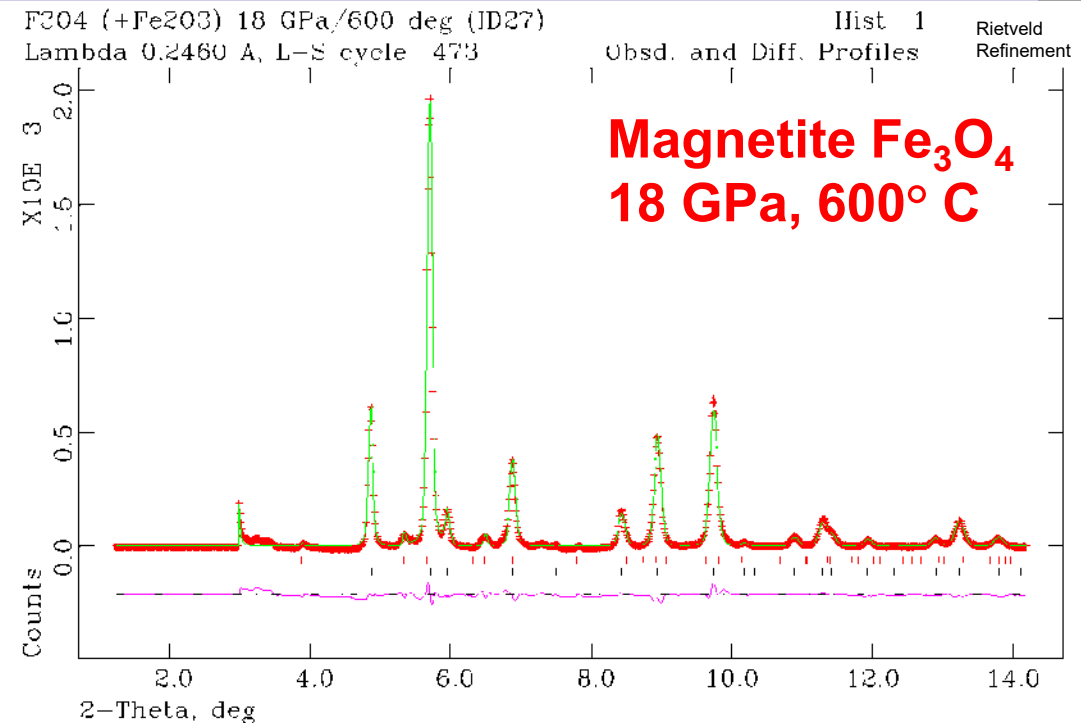
ID 27



# Some typical diffraction patterns



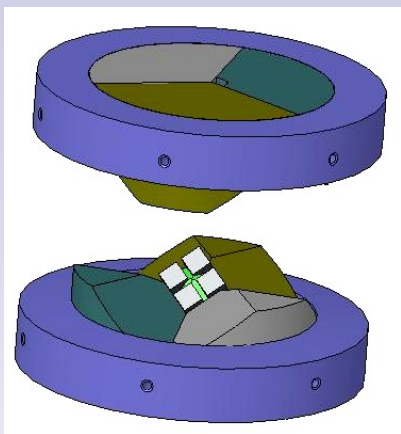
Sample of Copper and MgO



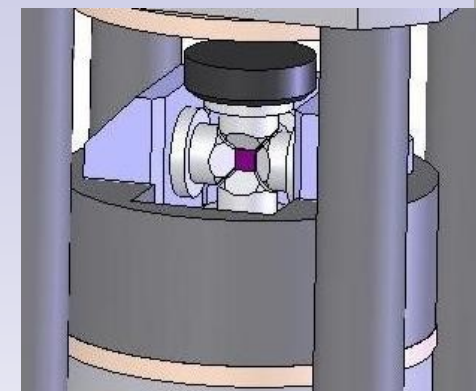
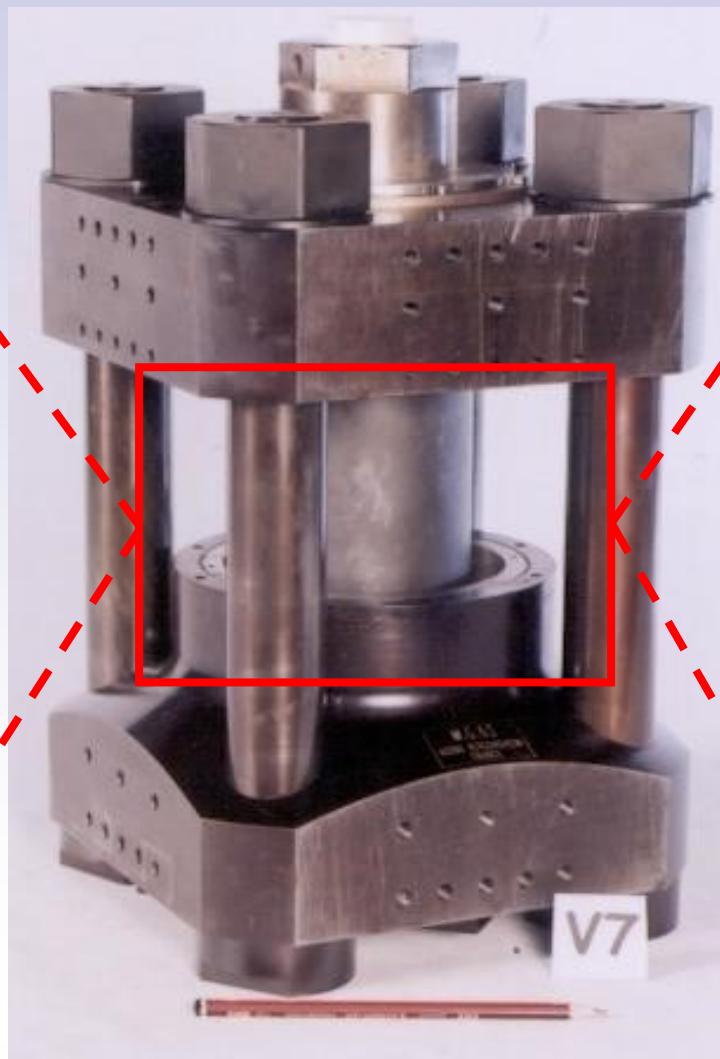
*All reflections provide from the sample, there is no contaminant signal from the sample environment*

Finally, these experiments showed that **the quality of diffraction patterns** which can be obtained at HPHT **is excellent**, and that the data quality is comparable to that obtained with **the standard opposed-anvil setup in the Paris-Edinburgh press**

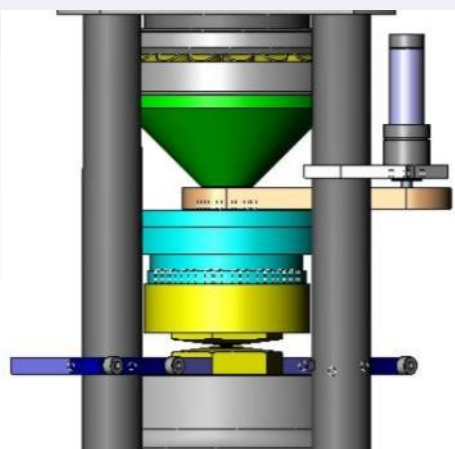
The V7/T-cup set-up with this  $\sim 20\%$  larger overall dimensions allows to accommodate very easily and quickly some other interesting HP modules



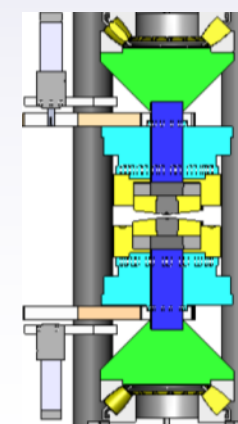
T-cup



DIA

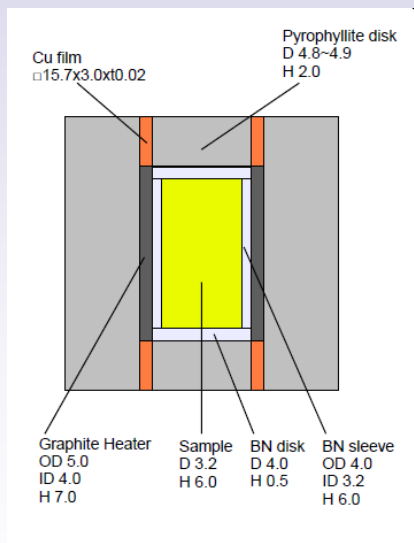
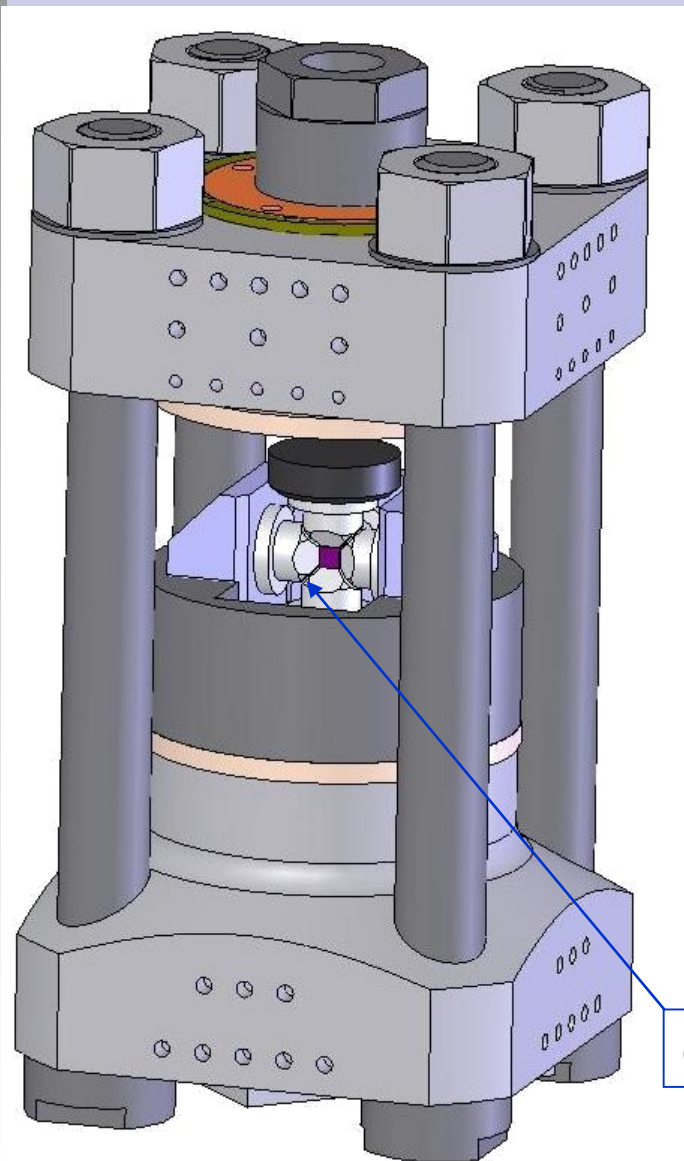


roPEc

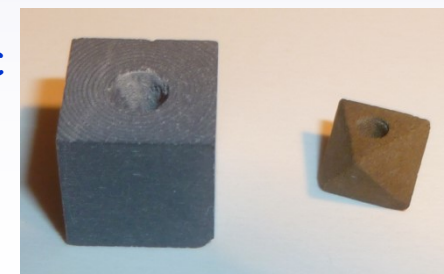


Ro-tomography

# DIA module



Gasket is cubic



a miniaturized DIA module

larger sample volume can be achieved



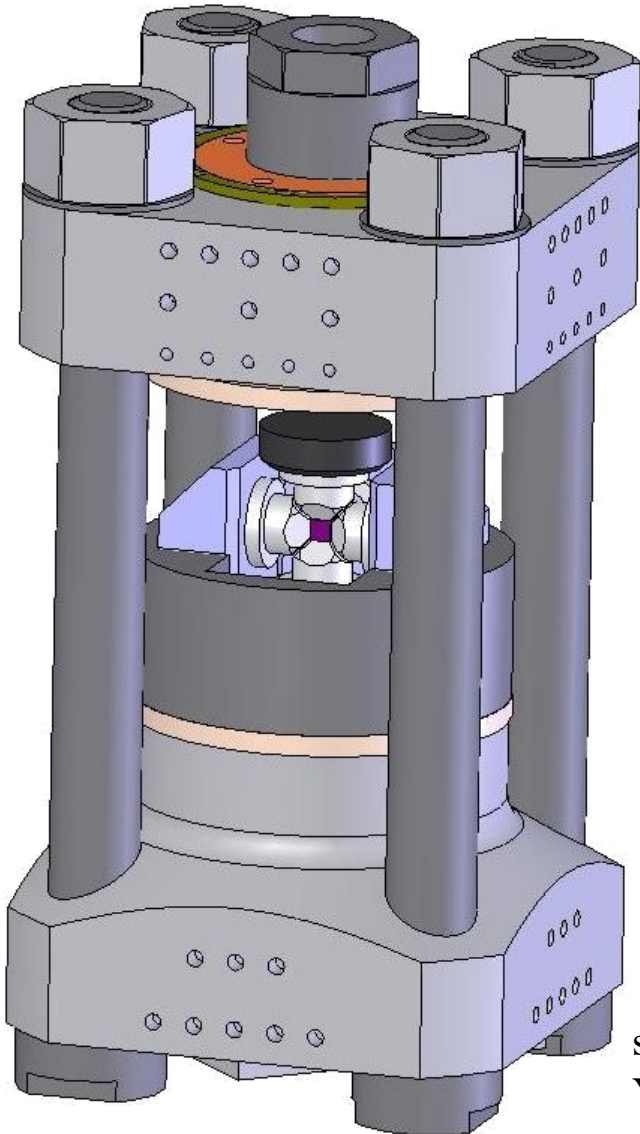
## D-DIA press

Hence, in the DIA configuration, **one pair of opposing anvils** can be moved independently from the other anvils

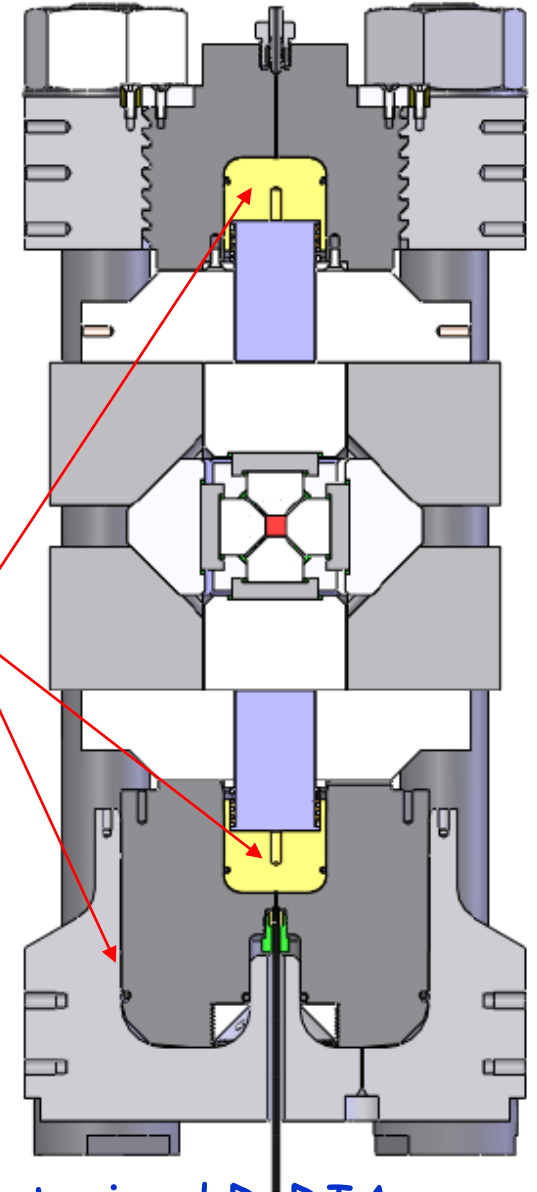
Four miniaturized displacement transducers will be used to measure the **displacements of anvils during deformation**

**three independent hydraulic rams**

similar to the system of  
Y. Wang, Rev. Sci. Instrum. 74, 3002 (2003)

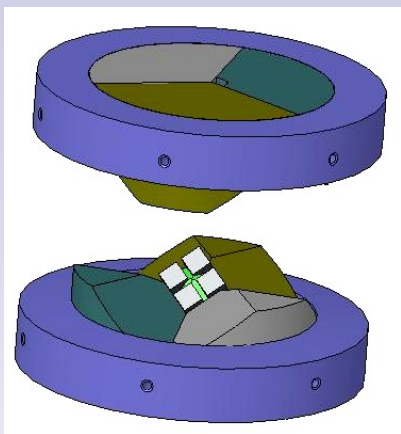


## Another possibility

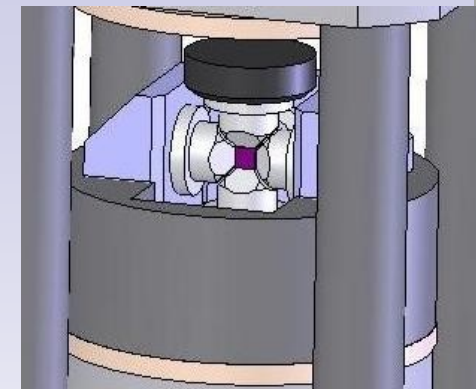
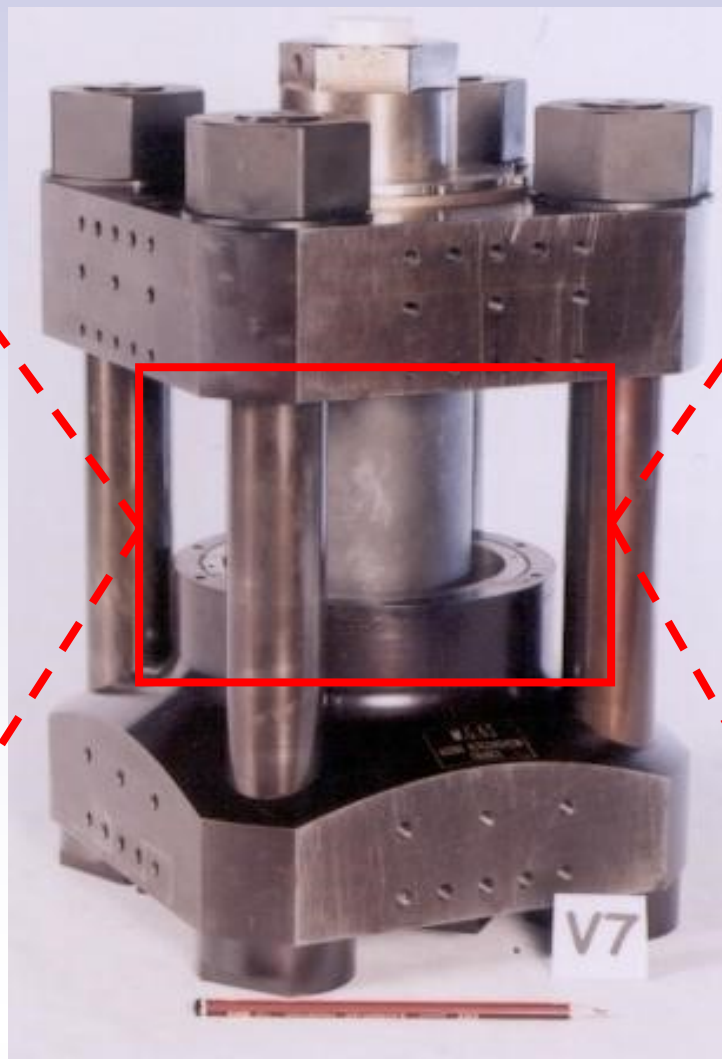


miniaturized D-DIA press

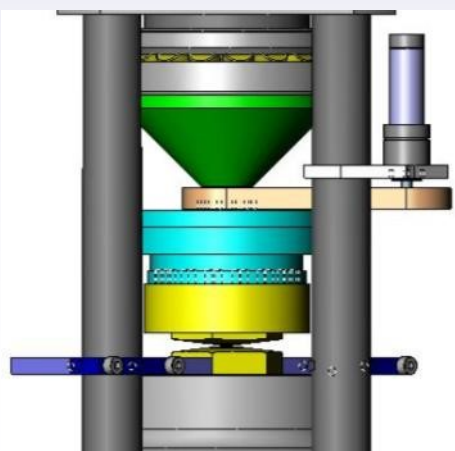
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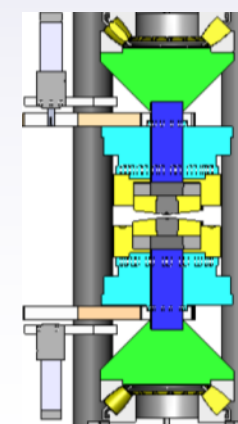
T-cup



DIA

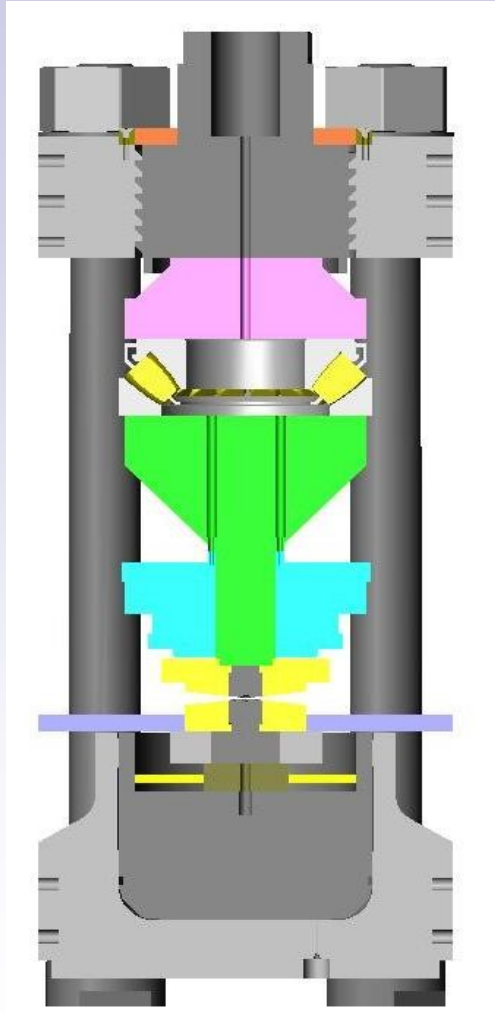
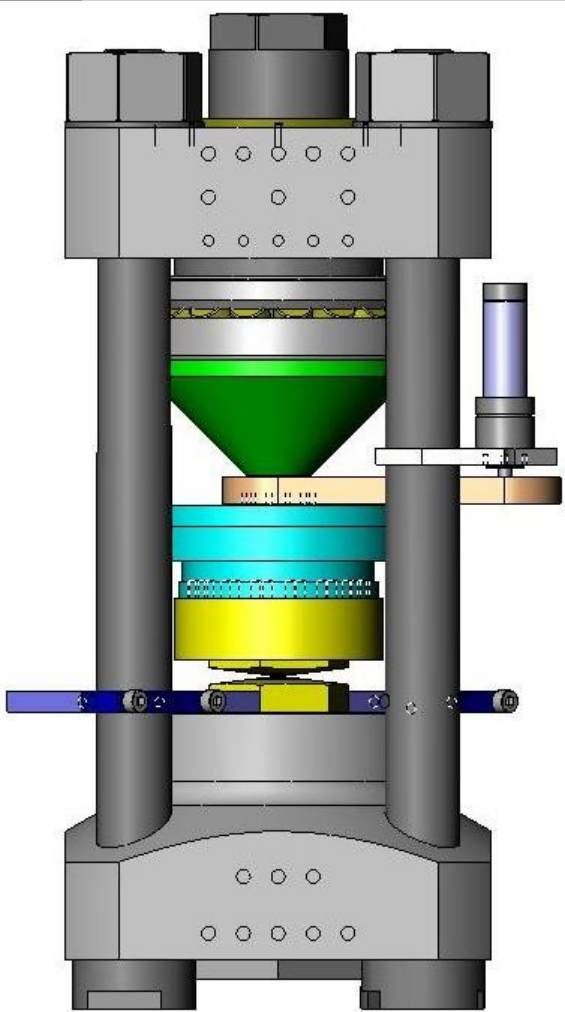


roPEc



Ro-tomography

roPEc module : Torsional deformation under simultaneous high (P,T) conditions (can also synthesize new materials in these unusual conditions !)



Strains exceeding 5 or 6 are possible

A rotational actuator attached to the upper anvil (while the lower anvil is fixed) allows deformation of the sample in simple shear geometry (torsion) under controlled strain rate



# Mechanochemistry studies at HPHT

It is well known that the effect of plastic shear deformation leads to many interesting effects, summarized here :

A significant (by a factor 3 to 10) reduction of pressure transformation for a well known phase transition (PT) and chemical reaction

The synthesis of new phases which could not be produced by hydrostatic pressure

New chemical reactions (shear-induced metallization and oxidation, polymerisation)

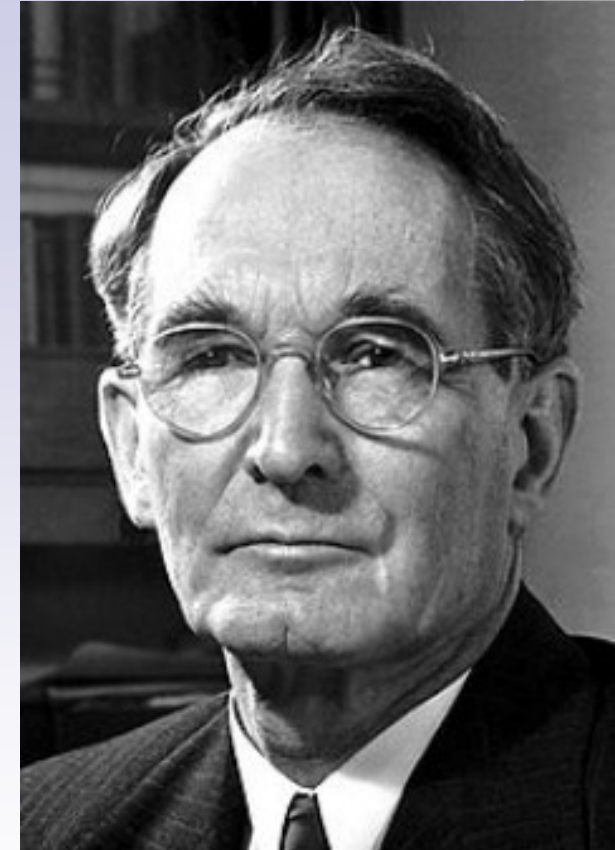
A substitution of a reversible PT by an irreversible one

A strain-controlled (rather than time-controlled) kinetics. It is possible to play with the kinetics of PT

A possibility to modulate the microstructure and to form bulk nanostructured materials

Phys. Rev. 7, 215–223 (1916)

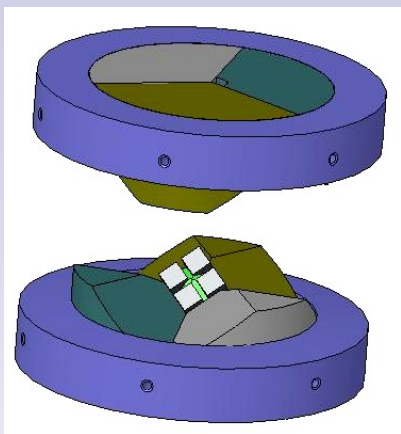
On The Effect of General Mechanical Stress on the Temperature of Transition of Two Phases, with a Discussion of Plasticity



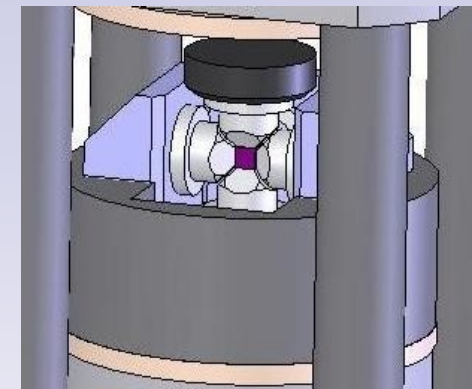
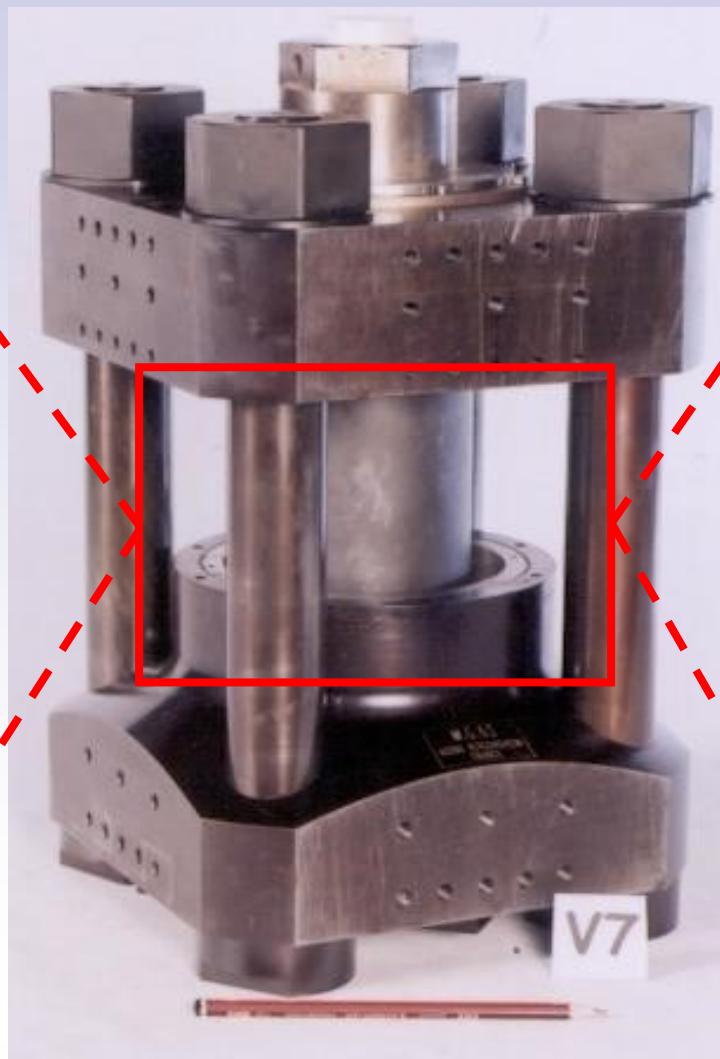
**Bridgman (1916)**

*An old idea but a new unexplored field which could be explored NOW with the roPEc module*

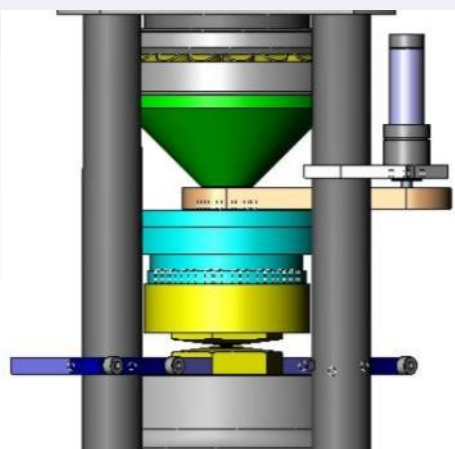
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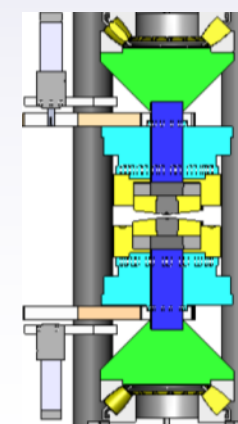
T-cup



DIA

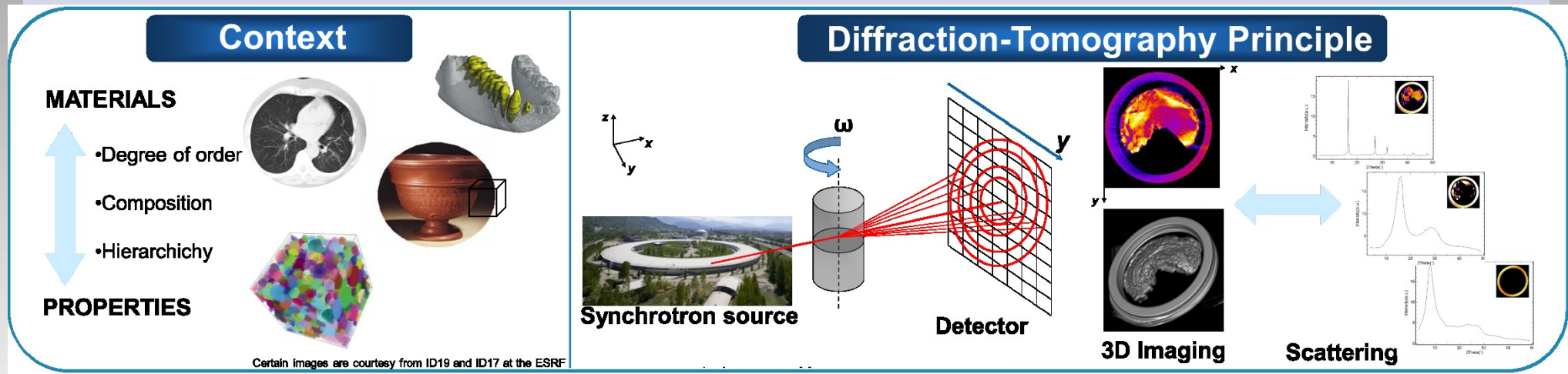


roPEc

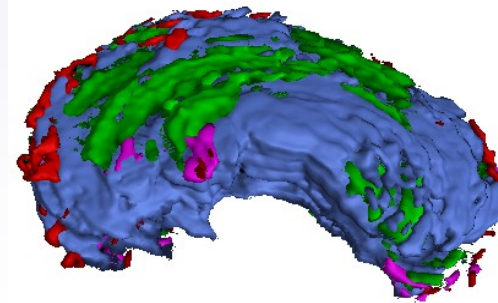


Ro-tomography

SB tomographic imaging is nowadays a **powerful technique for non-destructive, high-resolution investigations** of a broad kind of materials. High-brilliance and high-coherence third generation synchrotron radiation facilities allow **micrometer and sub-micrometer, quantitative, three-dimensional imaging within very short time**. Coupling this technique with HP would be very interesting !!!



- Penetrating radiation (X-rays)
- Contrast : absorption, diffraction, fluo
- 2D angular projections
- Reconstruction of 'virtual slices'
- Volume elements (voxels)



Set-up in U.S :

Y. Wang et al., REVIEW OF SCIENTIFIC INSTRUMENTS 76, 073709 (2005)

- 3D probe
- Non destructive
- Phase selective
- High sensitivity
- High spatial resolution




# Ex situ tomographic study on recovered samples demonstrated how useful *in situ* high P/T/stress tomographic imaging could be!

- ability to **study evolution of textures** in multi-phase systems
- time-resolved studies of development of melt networks
- Study of **microstructure evolution of composite materials** under high pressure, temperature, and during deformation
- Study of **strain partitioning** in composite materials (strains in the inclusions and the matrix can be determined by mapping the locations and shape change of the inclusions)
- **a useful tool in directly measuring** volume changes of noncrystalline materials as a function of both pressure and temperature : **density of noncrystalline materials at HPHT**
- A useful tool for the new "Diffraction-micro-tomography" method to **examine *in situ*** some **minor unidentified phases in high (P,T) synthesised polycrystalline materials**. This method could help to **precisely determine the local mechanism of high (P,T) synthesis** (Pierre Bleuet et al., *Nature Materials*, 7:468472, June 2008 and M. Alvarez et al., *Applied Crystallography* 44, 163, January 2011 and *Phys. Rev. Lett.* 109, 025502 (2012))


 nature  
materials

Probing the structure of heterogeneous diluted materials by diffraction tomography

PIERRE BLEUET<sup>1</sup>, ELÉONORE WELCOMME<sup>2</sup>, ERIC DOORYHÉE<sup>3</sup>, JEAN SUSINI<sup>1</sup>, JEAN-LOUIS HODEAU<sup>3\*</sup> AND PHILIPPE WALTER<sup>2</sup>


 Physical Review Letters

moving physics forward

PRL 109, 025502 (2012)

week ending  
13 JULY 2012

“Compressed Graphite” Formed During C<sub>60</sub> to Diamond Transformation as Revealed by Scattering Computed Tomography

M. Álvarez-Murga,<sup>1,2</sup> P. Bleuet,<sup>3</sup> G. Garbarino,<sup>1</sup> A. Salamat,<sup>1</sup> M. Mezouar,<sup>1</sup> and J. L. Hodeau<sup>2\*</sup>

The design of a cell for *in situ* High P/T/stress Tomographies has to satisfy several specific requirements :

- Large X-rays beams (absorption)
- Monochromatic micro-beams (diffraction)
- Large sample volume ( $\sim 2 \text{ mm}^3$ ) to obtain meaningful structural information on microstructure and its evolution
- Angular access of  $180^\circ$  to the sample
- Coaxiality of the anvils, Flatness of sample
- Accurate and simultaneous rotation of the anvils under High Pressure and High Temperature
- Possibility to rotate the anvils in opposite directions (deformation)
- High Energies and high flux
- Transparency of sample environment (absorption)
- Eliminate scattering signal from gasket (diffraction) : compatible with a sollers slits system
- Maintain sample geometry during compression (limit gasket lateral extrusion)
- Easy to handle, portative. The press has to fit on the ID27 Beamline of the ESRF and easily exportable to other beamlines (SOLEIL, DIAMOND)

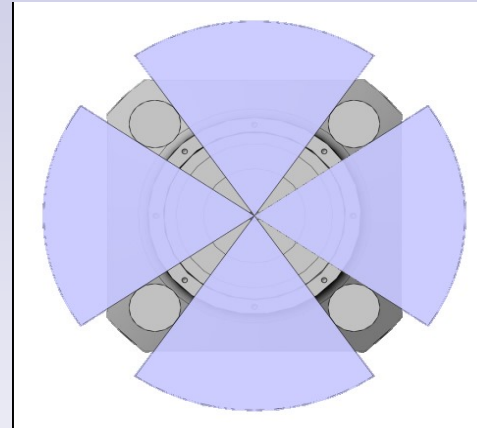
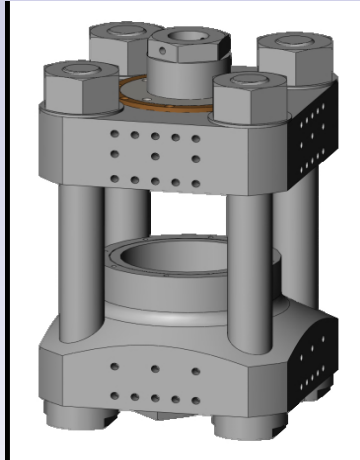


# Paris-Edinburgh presses

the first **simple** idea is to rotate the press itself !

## V4 Type

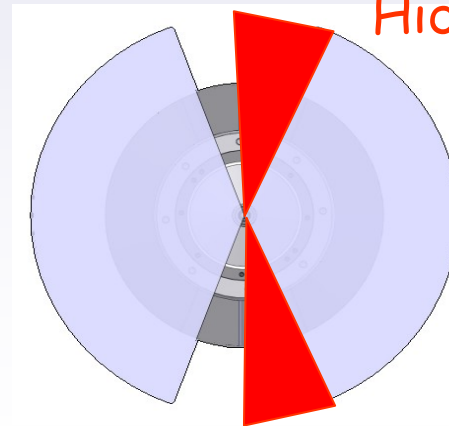
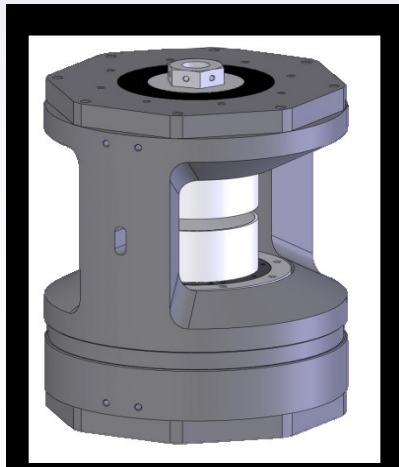
4 columns



68,8°

## VX3 Type

2 columns



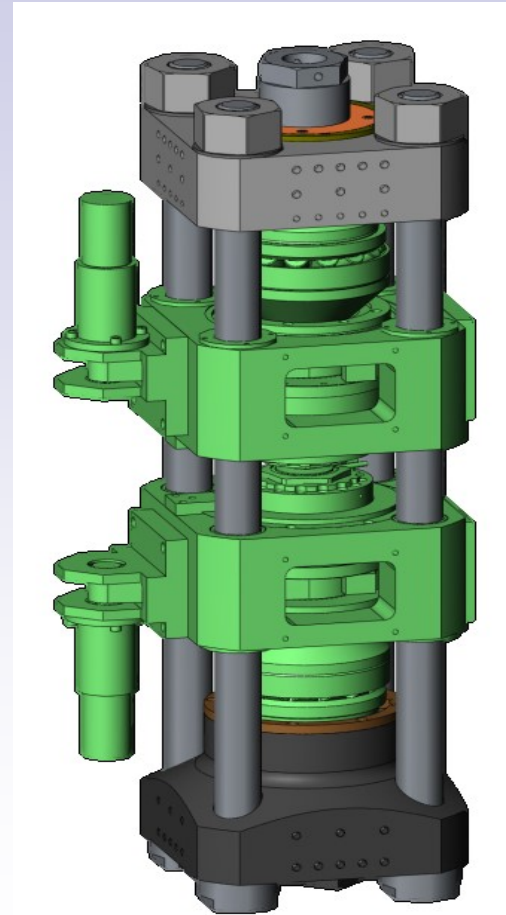
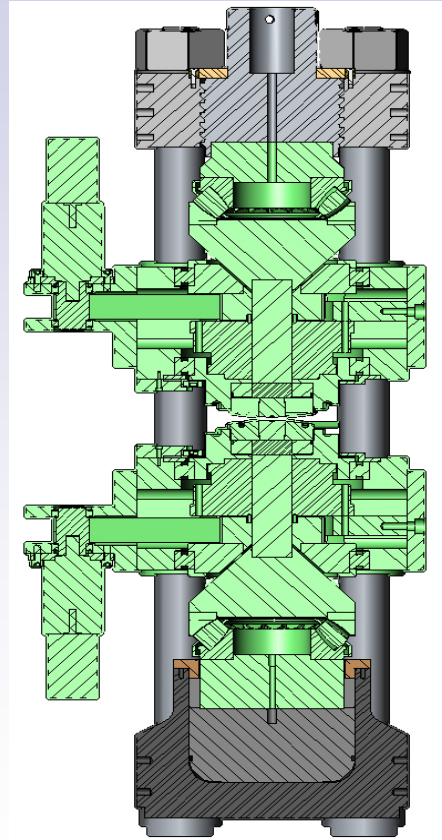
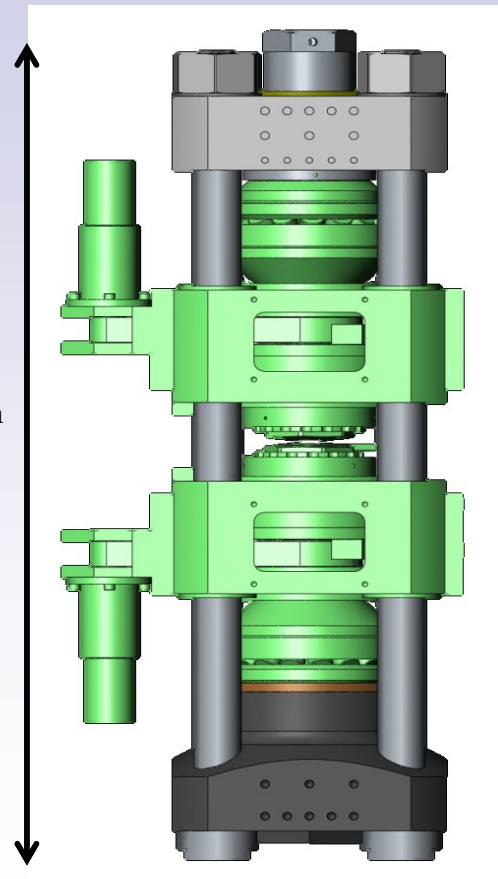
Hidden !

140°

columns are a problem !

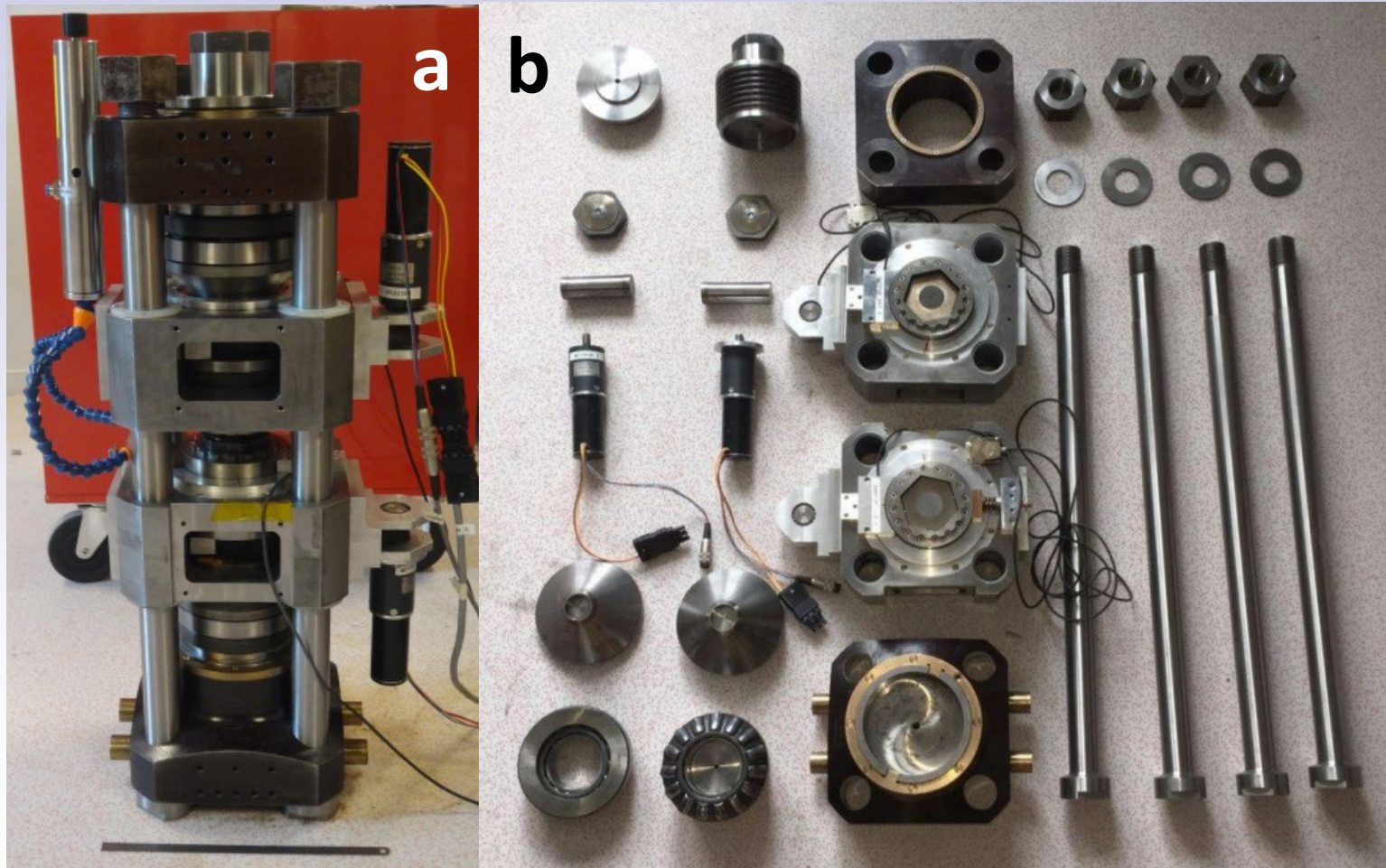
it is not possible to scan in excess of **140°**. For the tomography scan a minimum rotation of **180°** is required. The only solution to this issue is the development of a **new system of rotating anvils**, which will allow the **180°** scanning of the slice

# Rotational Tomography PE module



The **rotoPEc** is **easy** to **install** to any beamline (within 20 minutes) and can be **removed quickly**

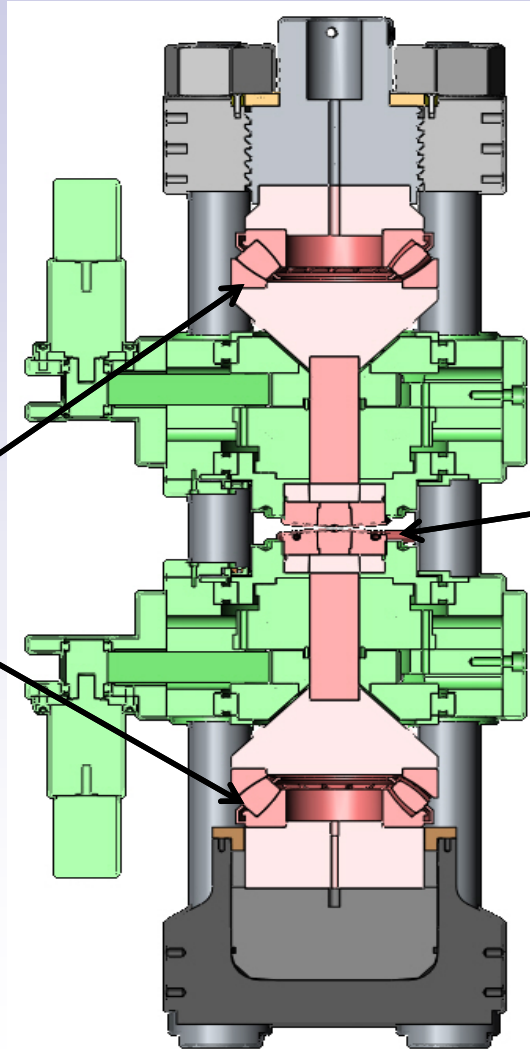
## Rotational Tomography PE module



The **rotoPEc** is **easy** to **install** to any beamline (within 20 minutes) and can be **removed** **quickly** (10 minutes)



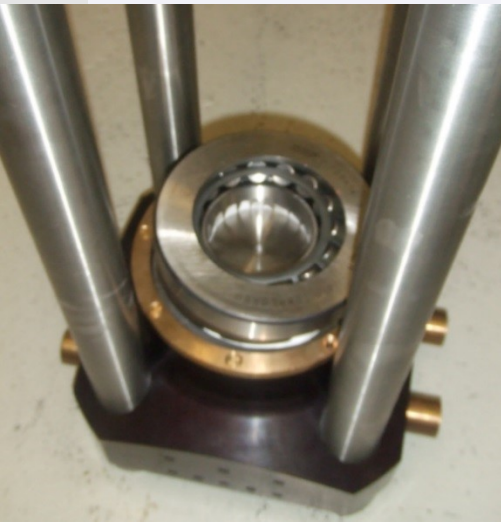
# Pressure transmission



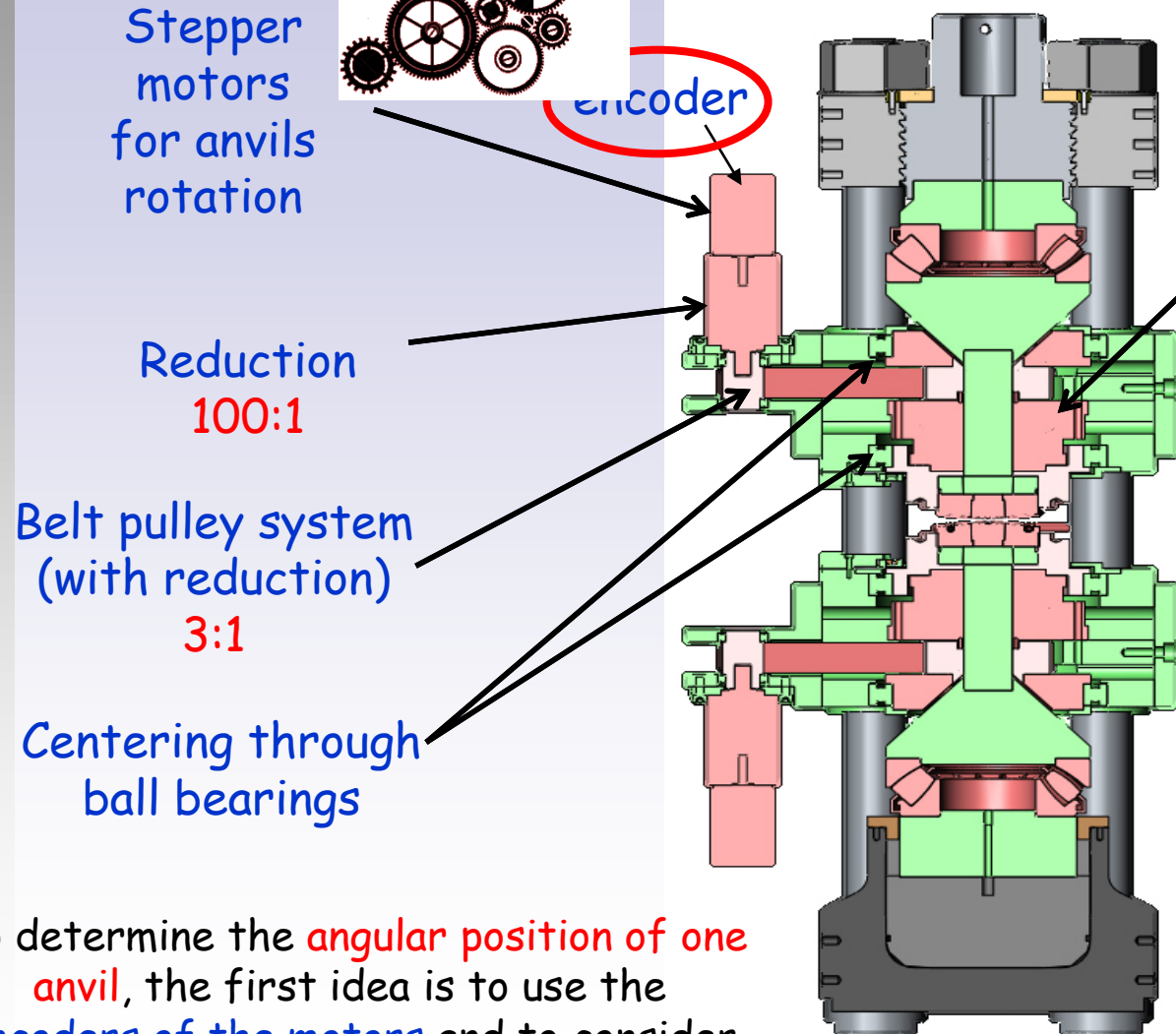
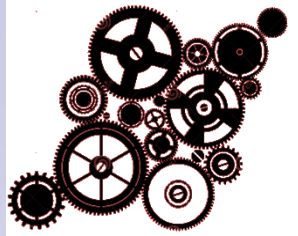
2 thrust bearings with a capacity of 125 tonnes

Hexagonal anvils and sample

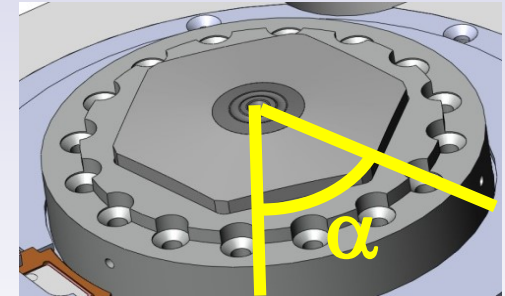
sufficient to reach **15 GPa** in our experiments



## Torque transmission



Gear reducer  
Max torque: 892 Nm  
160:1



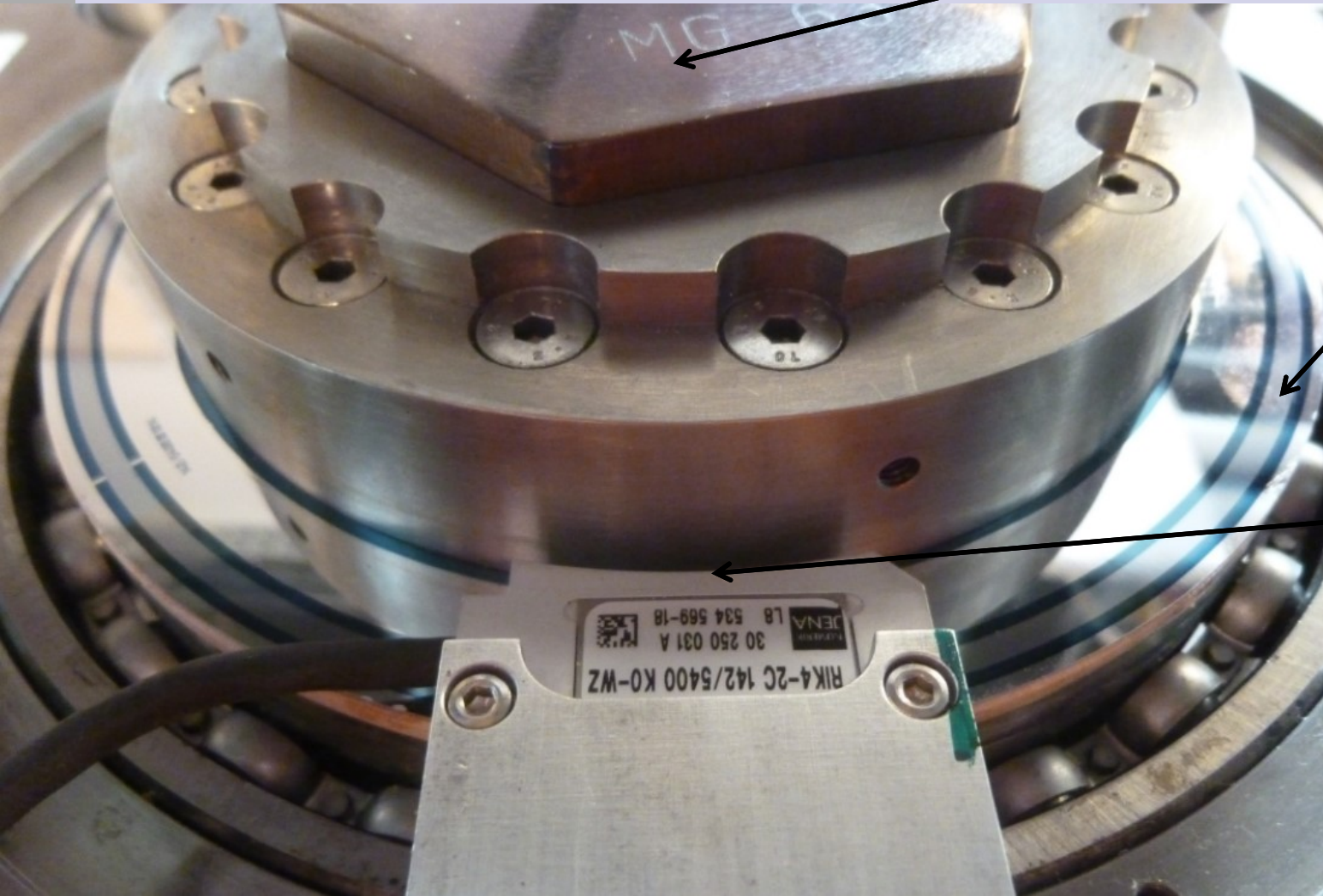
Reduction ratio: 48000:1  
48 000 engine rotations to perform a complete rotation of an anvil (in 9 min with the mean speed of the motor)

Accuracy was not sufficient for tomography!  
Our idea is to control the motors with two encoders placed directly on the anvils

To determine the angular position of one anvil, the first idea is to use the encoders of the motors and to consider that the system is perfect (lossless) with the reduction ratio

The idea is to **control the motors** with two encoders placed directly on the anvils

Hexagonal anvil



incremental  
encoder

Optical Drive can read the  
various positions

*Resolution*  
*0,005 °*



We have developed a **software to control the two motors with these incremental encoders** in order **to monitor** the angular **position** and/or **velocity** of the anvils

**List serial port com found**

Port Com: COM5

Initiate Device Connections

Connect

**SAVE FILE**

File Name: C:\

Comments:

Period Save (s): 5

Save Data

**ENCODER NUMERIK JENA**

GO TOMOGRAPHY MODE

Pulse Width: 100 (ms)

Target Position (°): 180

Step (°): 2.5

Stop Duration (s): 1

State Encoder Jena:

**DYNAMIC MOVEMENT**

Create Profile Move Profile Reset Profile

Profile Parameter In

- Signal type: sine
- Nb of samples: 250
- Amplitude (°): 5
- Periode (s): 30
- Cycle Number: 1
- DC Offset (°): 0

Graph: Anvil Position (degree) vs Temps (s)

Duration (s): 0

Period sample (ms): 0

Speed motor max (rpm): 0

Speed Anvil (°/s): 0.05

Speed Motor (rpm): 0

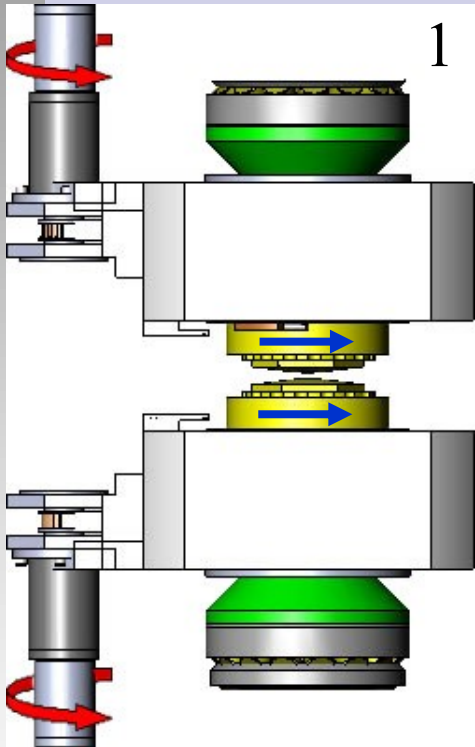
Acceleration (°/s<sup>2</sup>): 1.00

Position (°): 5

Mode: Relatif Absolute

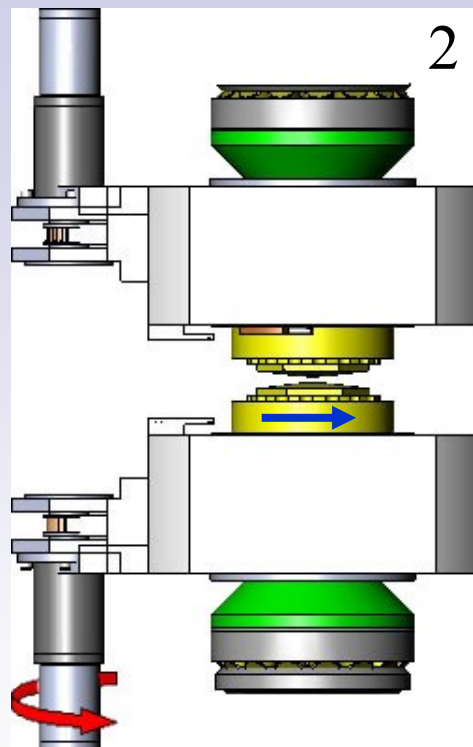
MOVE RESET

This module can operate **under HPHT** with **4 modes of operation**



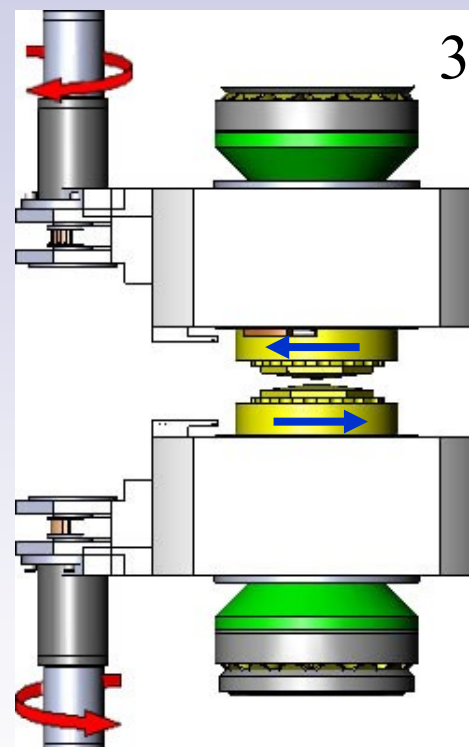
The two anvils rotate in the same direction simultaneously

**High P/T tomographies**



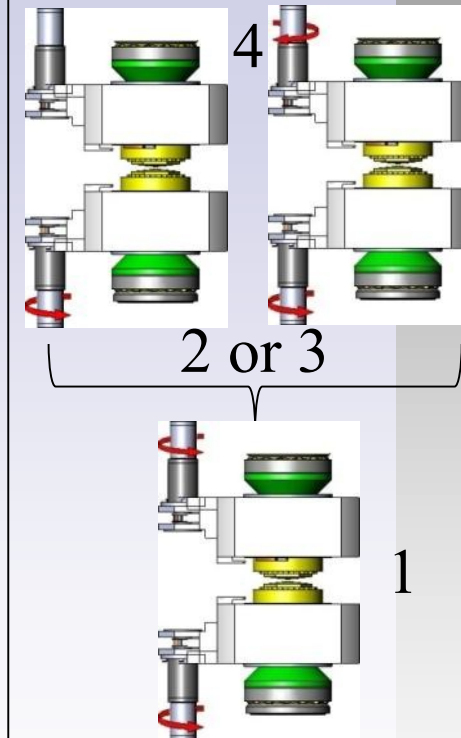
Only one anvil rotates, the other is fixed

**High P/T/stress diffraction**



The two anvils rotate in opposite directions

**High P/T/stress diffraction (fast)**

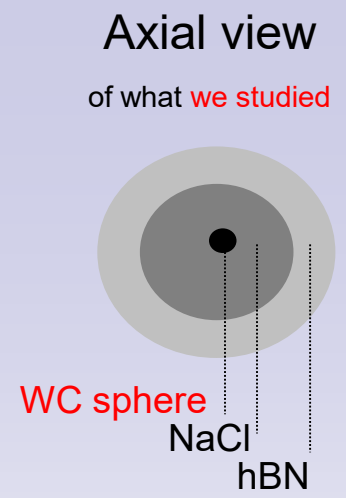
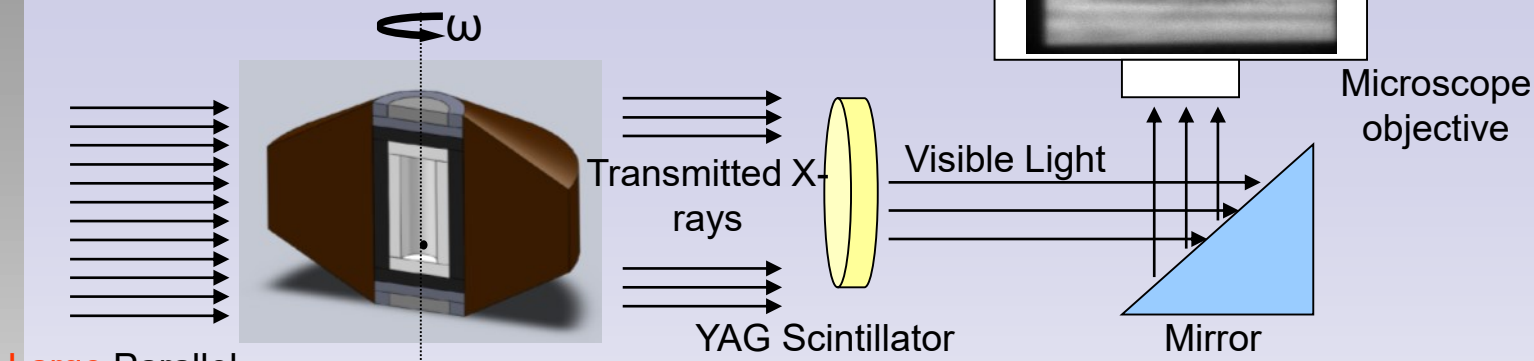


Combination of mode 2 or 3 followed by mode 1

**High P/T/stress tomographies**

# In situ high (P,T) Absorption Tomography

**P = 8 GPa and 1000K**

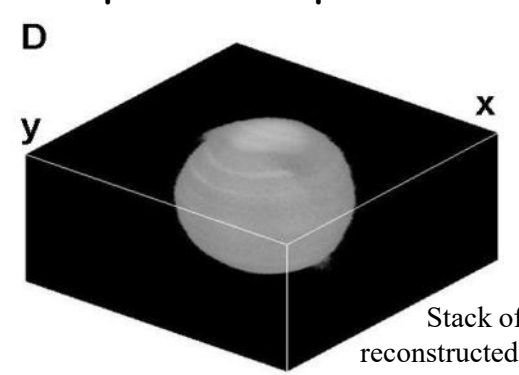
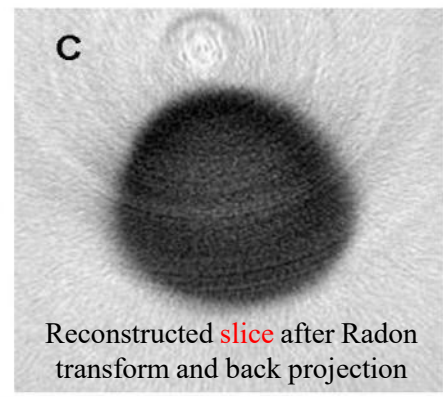
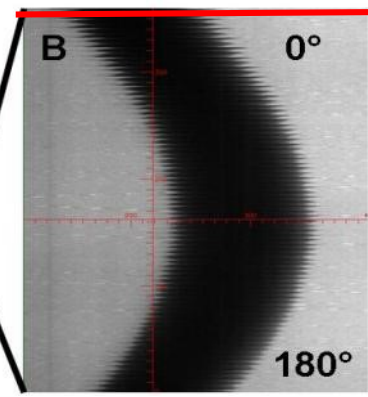
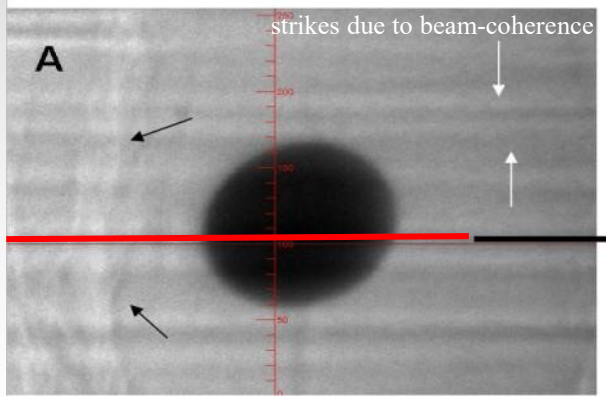


Large Parallel X-ray beam

RoToPEC

After tomography, the shape of the object has been perfectly reconstructed. The resolution of the new device will be determined from area/volume measurements for each pressure point

360 radiographies, 0.5° angular step





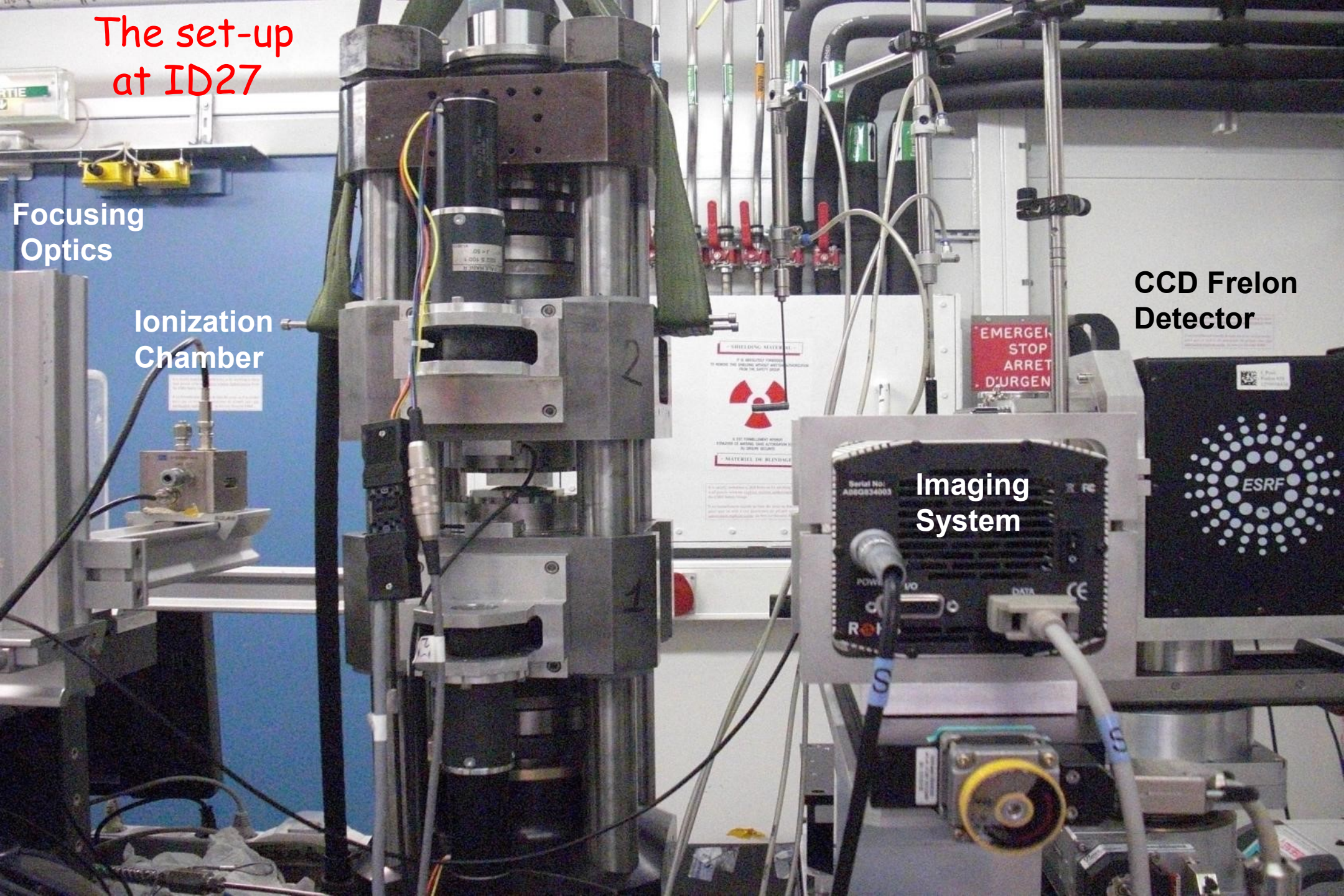
# The set-up at ID27

Focusing  
Optics

Ionization  
Chamber

CCD Frelon  
Detector

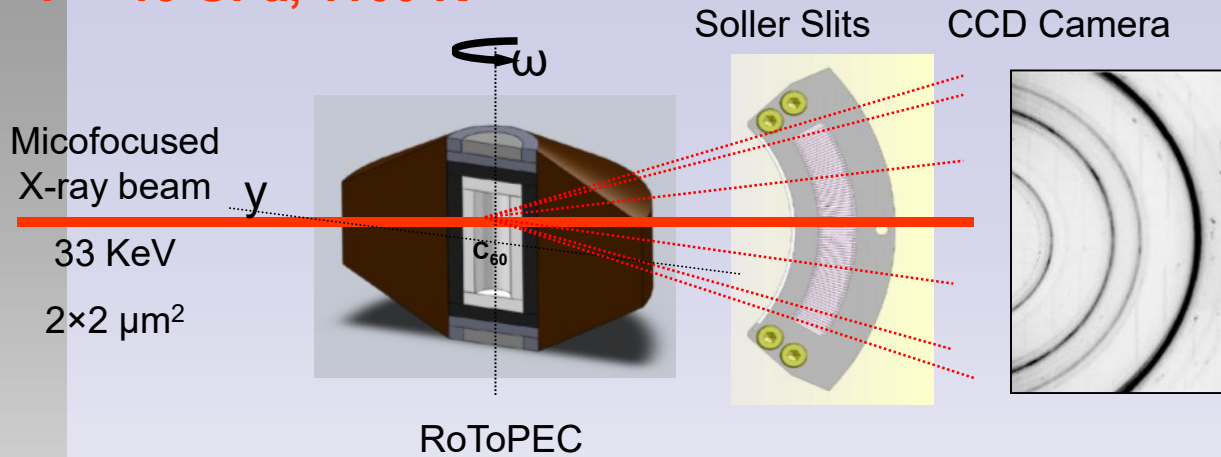
Imaging  
System



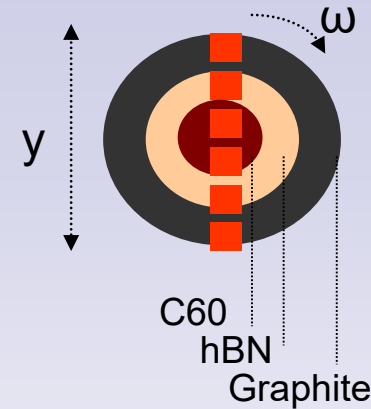


# In situ Diffraction Tomography

**P = 13 GPa, 1100 K**



## Preliminary experiment



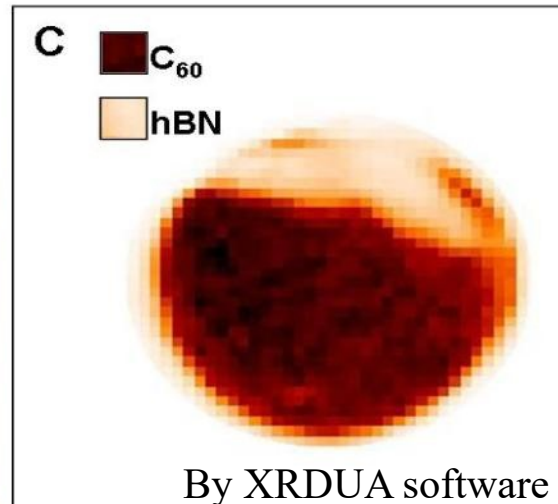
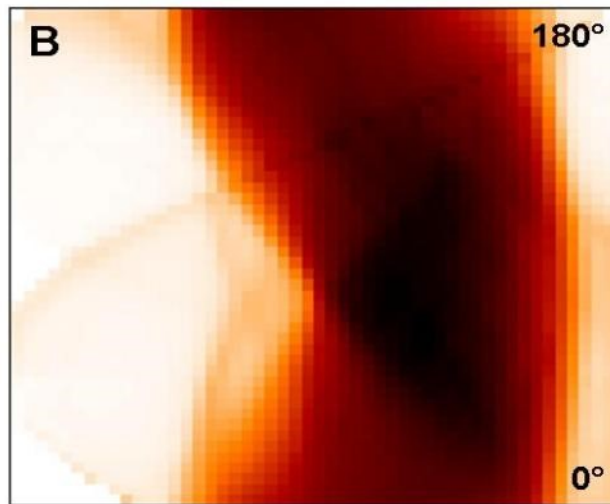
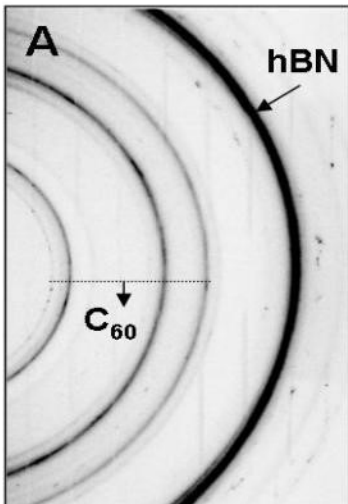
4° angular step, 26 μm linear step  
length 1.3 mm

Scanned

**C<sub>60</sub>**

To get very interesting information at high P/T/stress conditions :

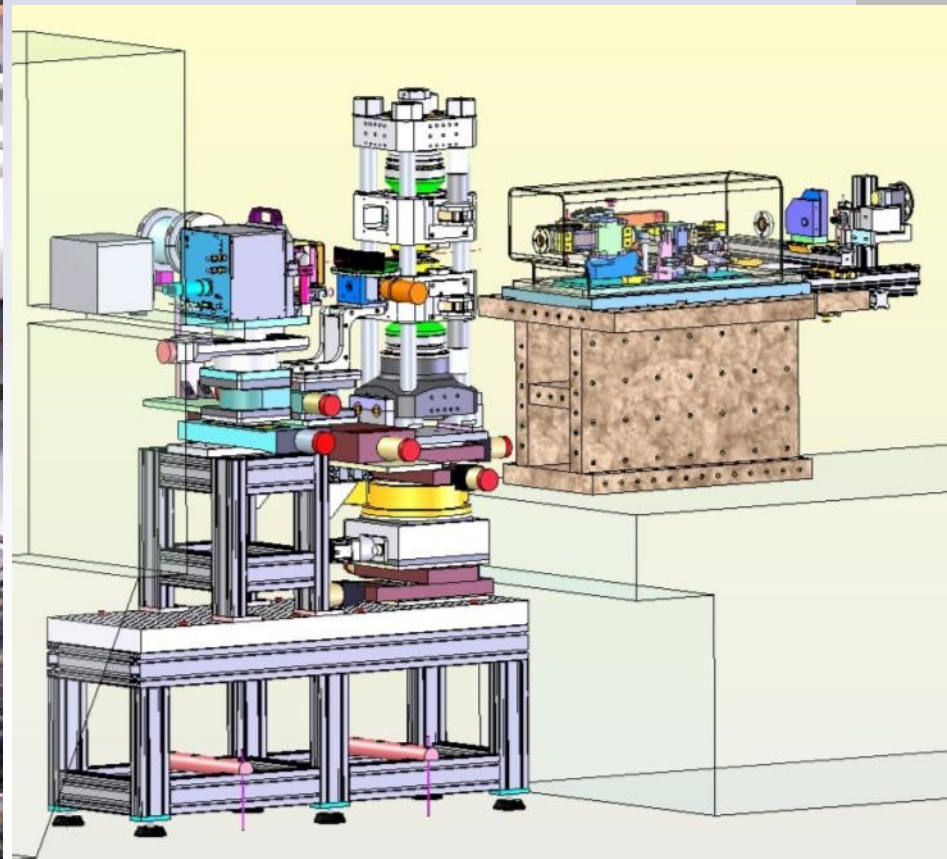
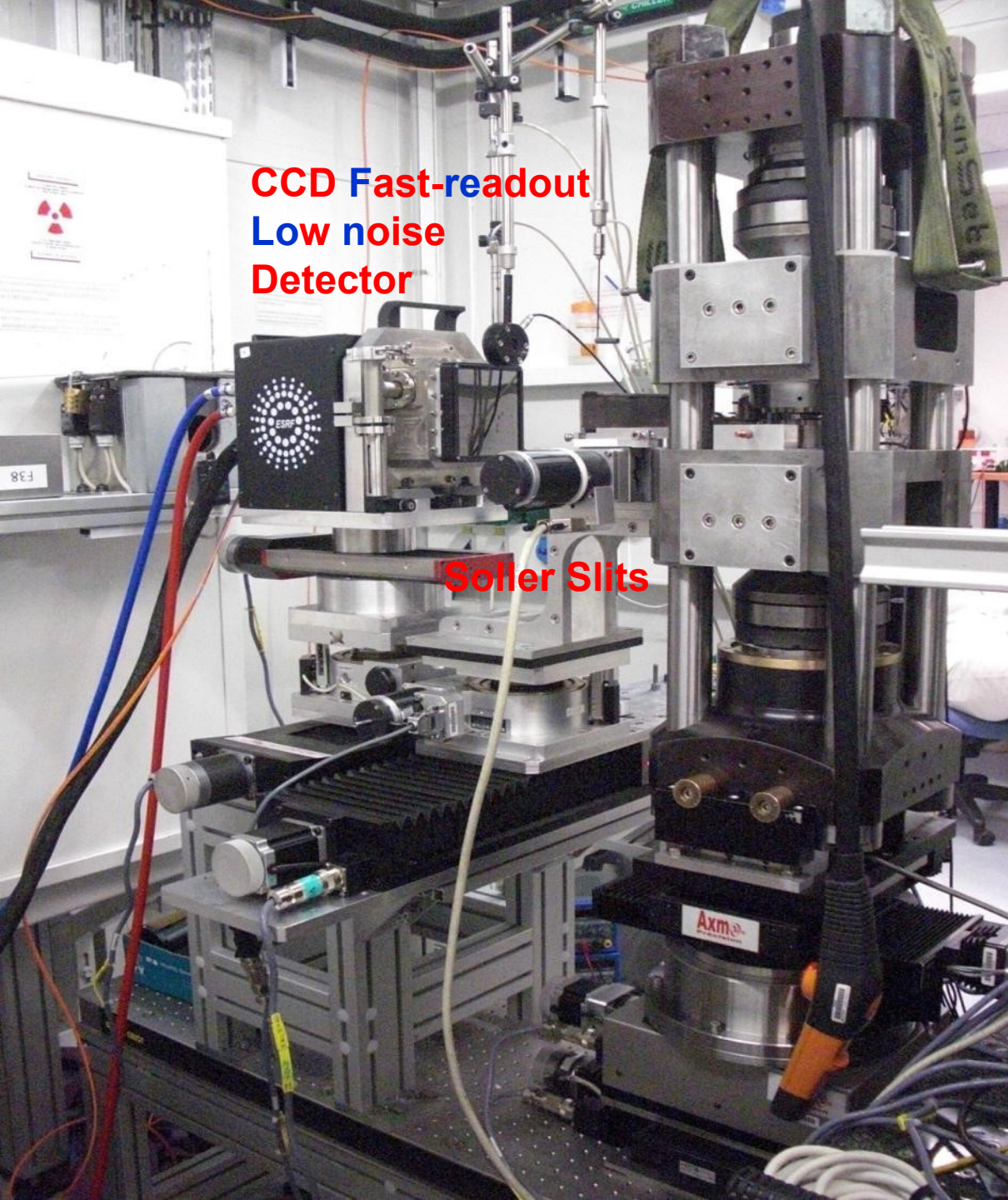
- Phase distribution - content
- Orientation (texture)
- Crystallite size/microstrain
- Degree of crystallinity



CCD Fast-readout  
Low noise  
Detector

Soller Slits

The corresponding set-up  
at ID27





This **new rotational tomography PE press** will provide **new scientific opportunities** for **original and unique studies** of phase transitions, density, crystallization and deformation in **extreme P/T/Stress conditions** with these 2 tomographies...

**Absorption  
Tomography**

(large beam, fast)

- Density
- Microstructure
- Porosity
- Defects
- Phase content

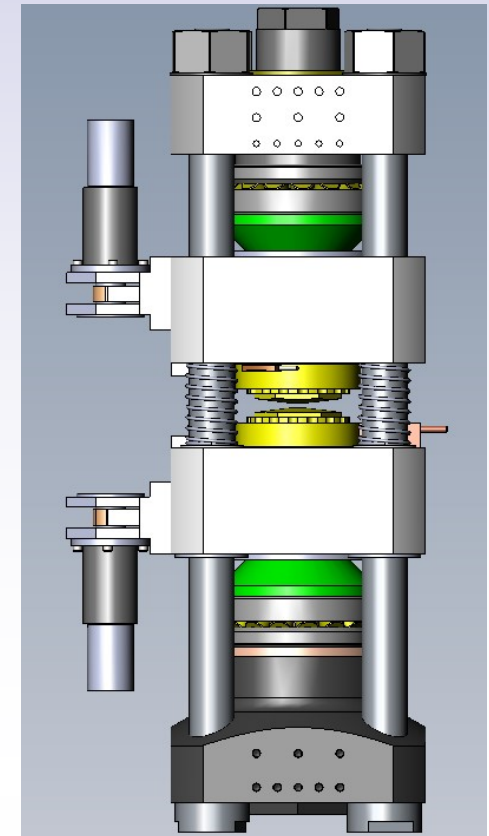
+

**Diffraction  
Tomography**

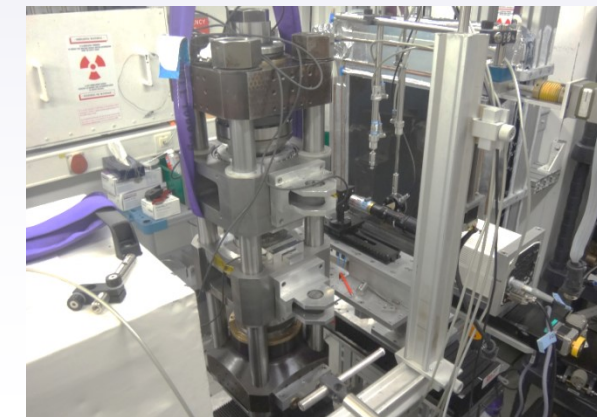
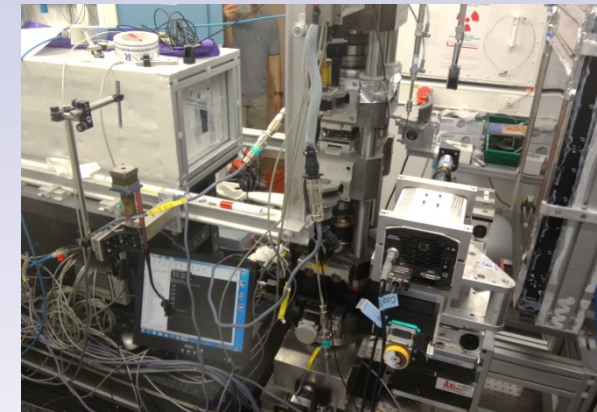
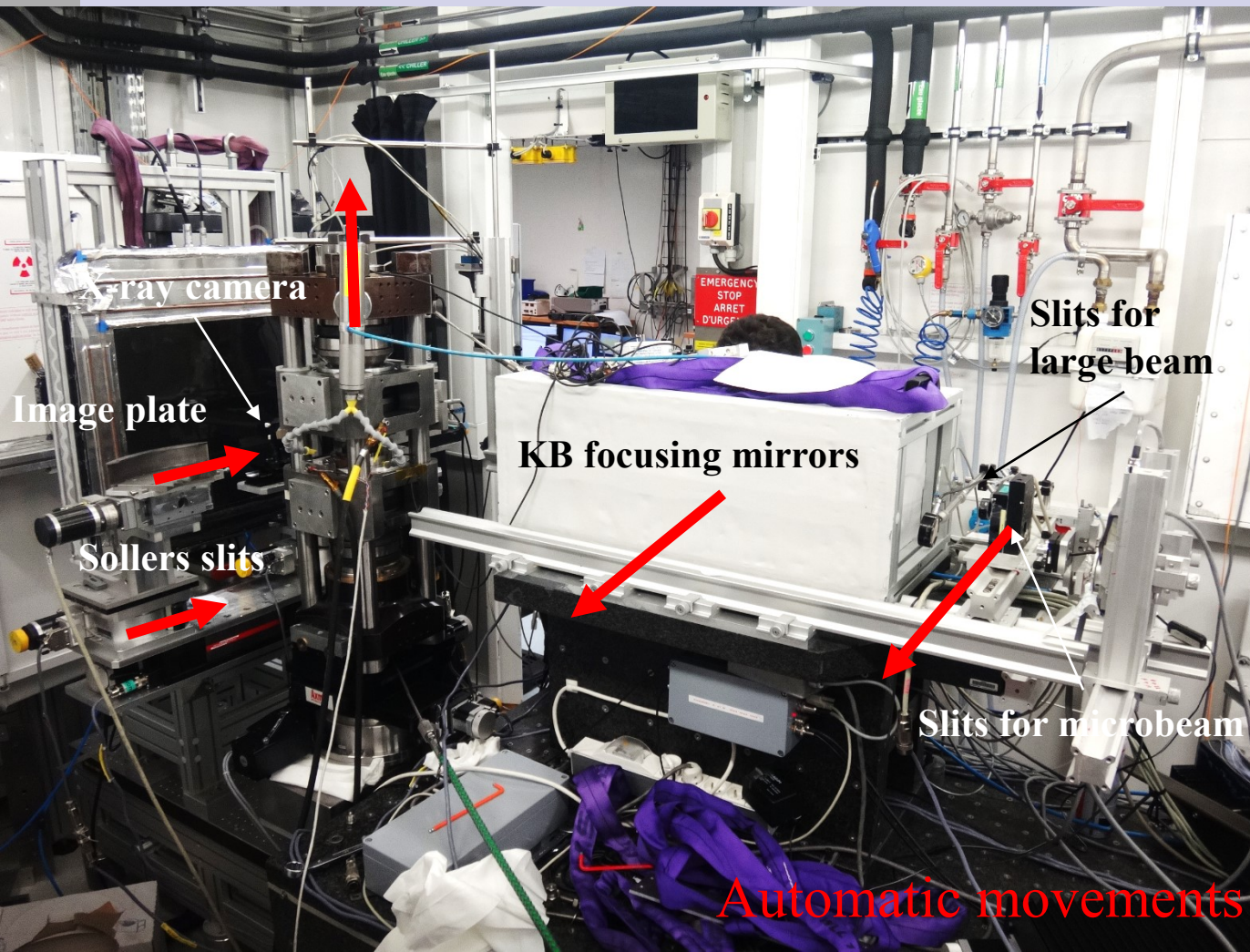
(micro beam, long)

- Phase distribution - content
- Orientation (texture)
- Crystallite size/microstrain
- Degree of crystallinity

**High  
P/T/stress  
PE cell**

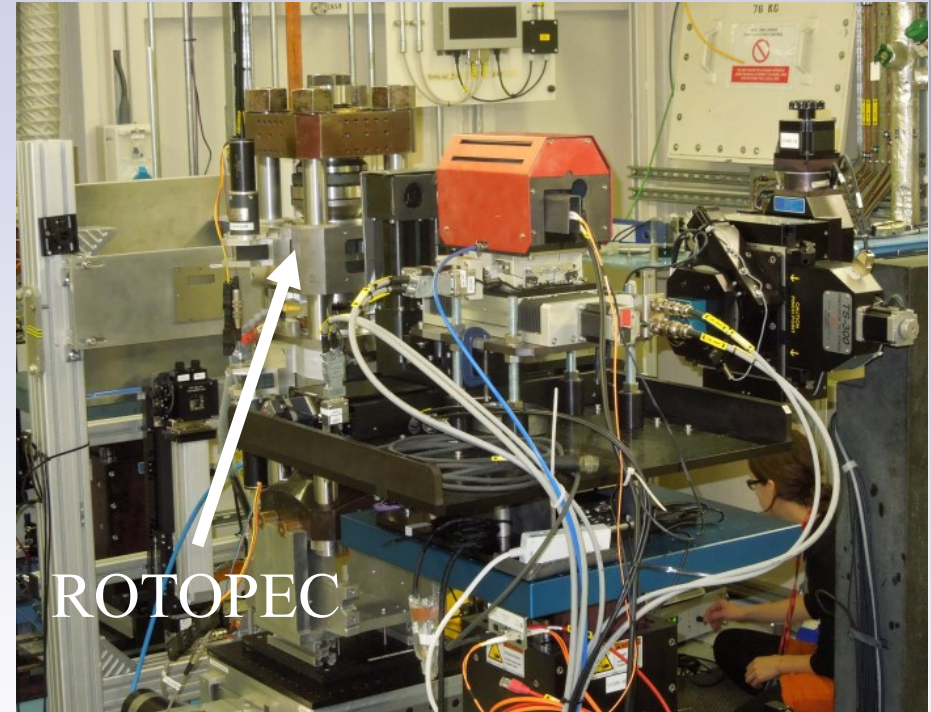
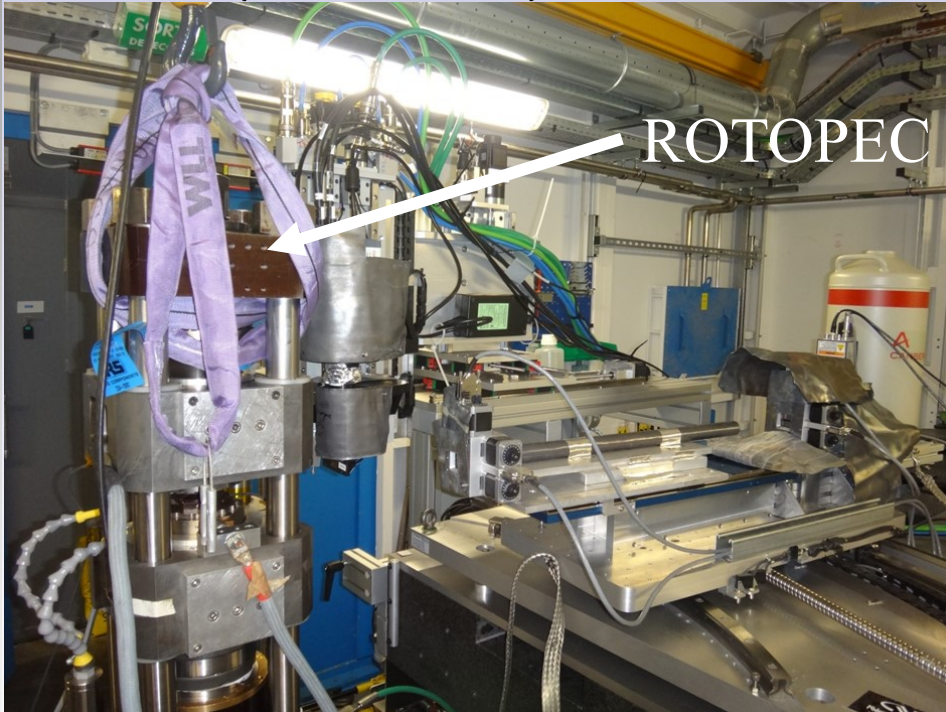


It is now possible to **switch** during an experiment between the **2 tomographies (absorption and diffraction)** by keeping the **same sample under HPHT**. It is done automatically (without going into the X-ray hutch) via automatic movements and in less **than a minute !!**





Our portable rotopec press has been **easily and successfully adapted** to various multi-modal **synchrotron experimental set-up**, so these experiments are possible not only at beamline ID27 (ESRF), but also at **PSICHE** (SOLEIL), and **I12** (DIAMOND) **beamlines**



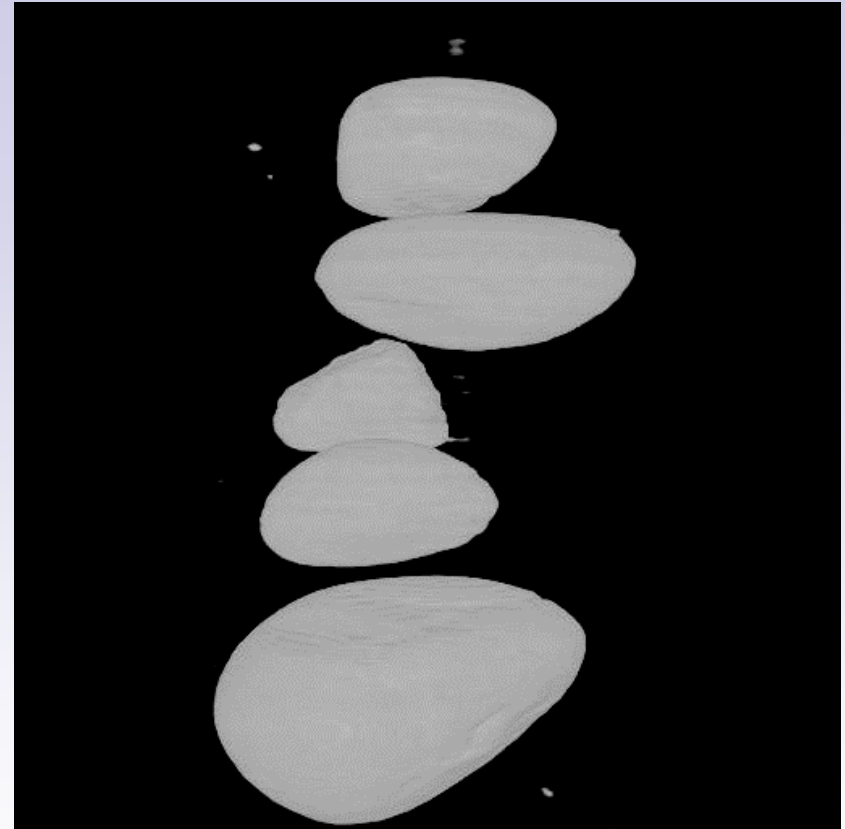


# 3 recent studies with **this novel** rotoPEc portable press

- **EOS of non-crystalline materials** : feasibility of accurately measuring **sample volume** as a function of **P and T**
- **Time resolved 3-D imaging of melt migration under extreme conditions**
- Measurements of **shear modulus** and **attenuation coefficient**  $Q_G^{-1}$  under **mantle conditions** and **seismic frequencies**

## EOS of non-crystalline materials : feasibility of accurately measuring sample volume as a function of $P$ and $T$

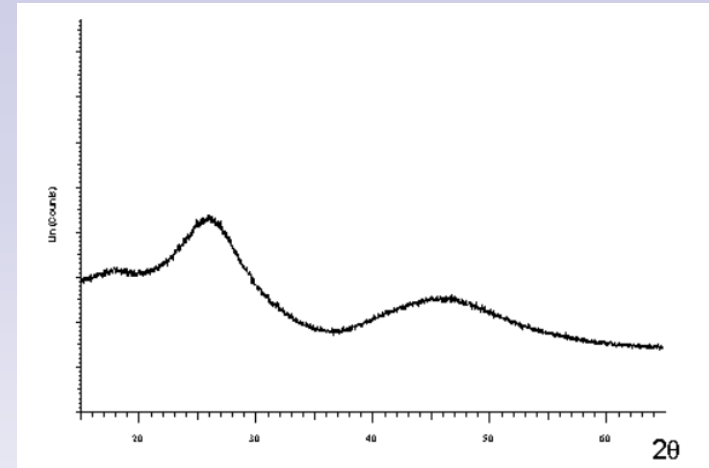
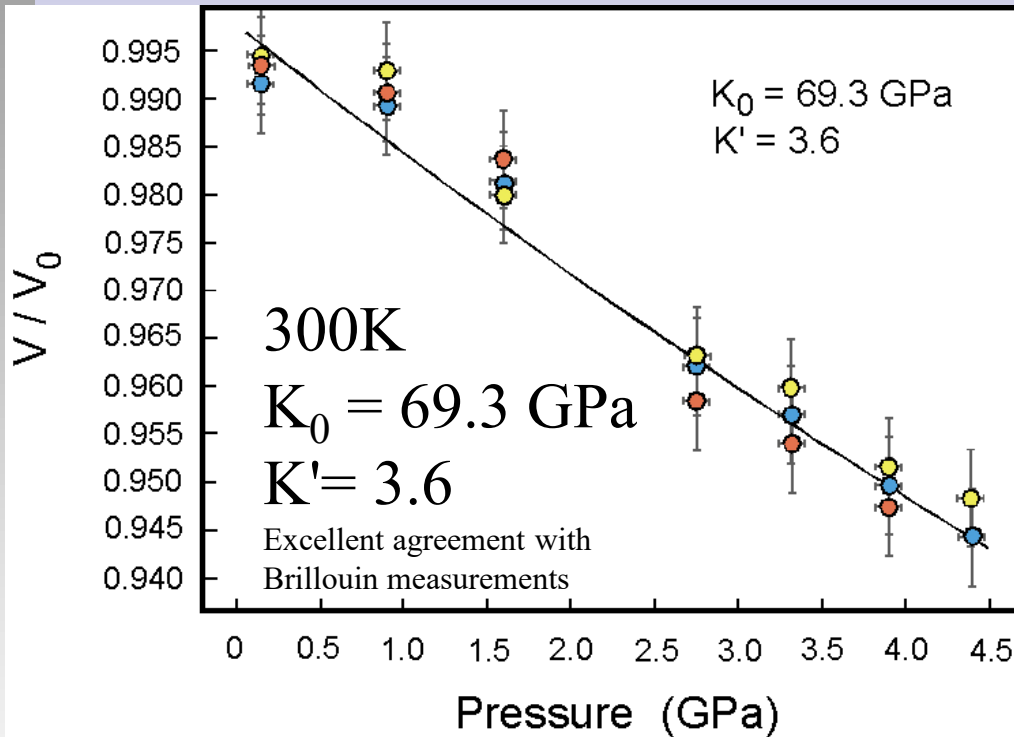
In this experiment at **SOLEIL**, the sample consisted of **vitreous olivine spheres** (diameter about  $800\ \mu\text{m}$ ) embedded in **BN powder**, hot pressed at **various  $P$  and  $T$**



3D rendering of the **five small olivine particles**, from which one can extract the **volume** of the sphere for each  **$P$  and  $T$**



## EOS of non-crystalline materials : feasibility of accurately measuring sample volume as a function of $P$ and $T$



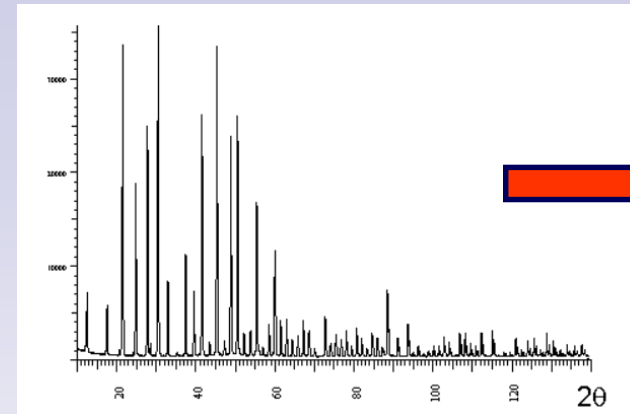
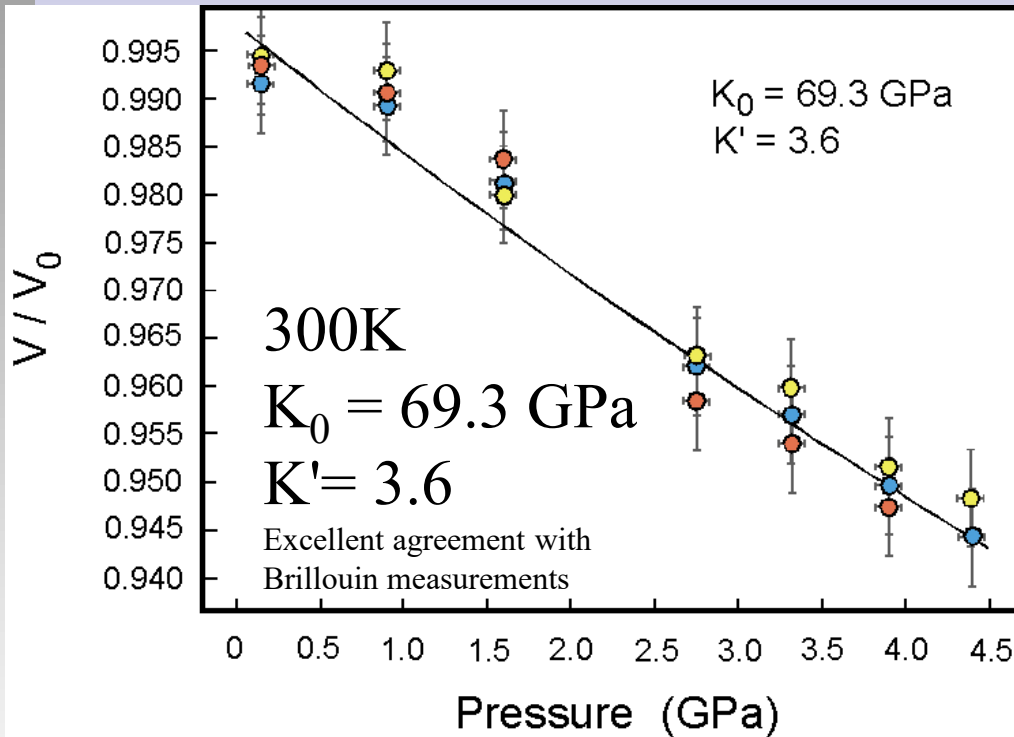
This study justifies **our technique** as a **useful tool** in directly measuring volume changes of **noncrystalline materials** as a function of both  $P$  and  $T$



*Talk of J.P. PERRILLAT will detail this study !!*



## EOS of non-crystalline materials : feasibility of accurately measuring sample volume as a function of $P$ and $T$

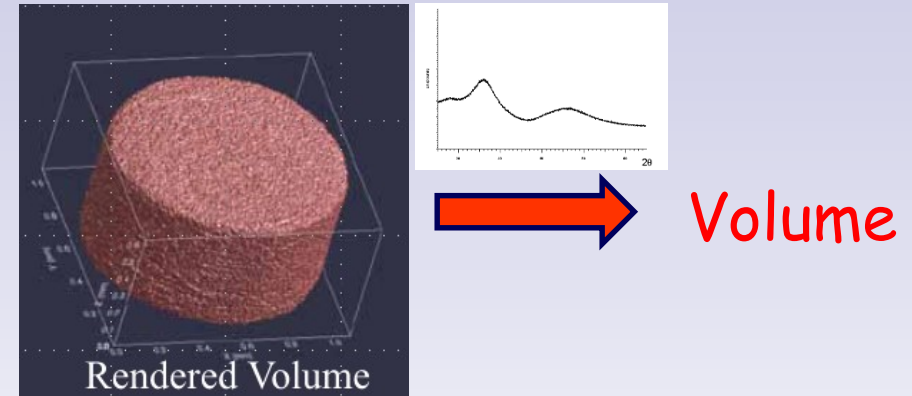
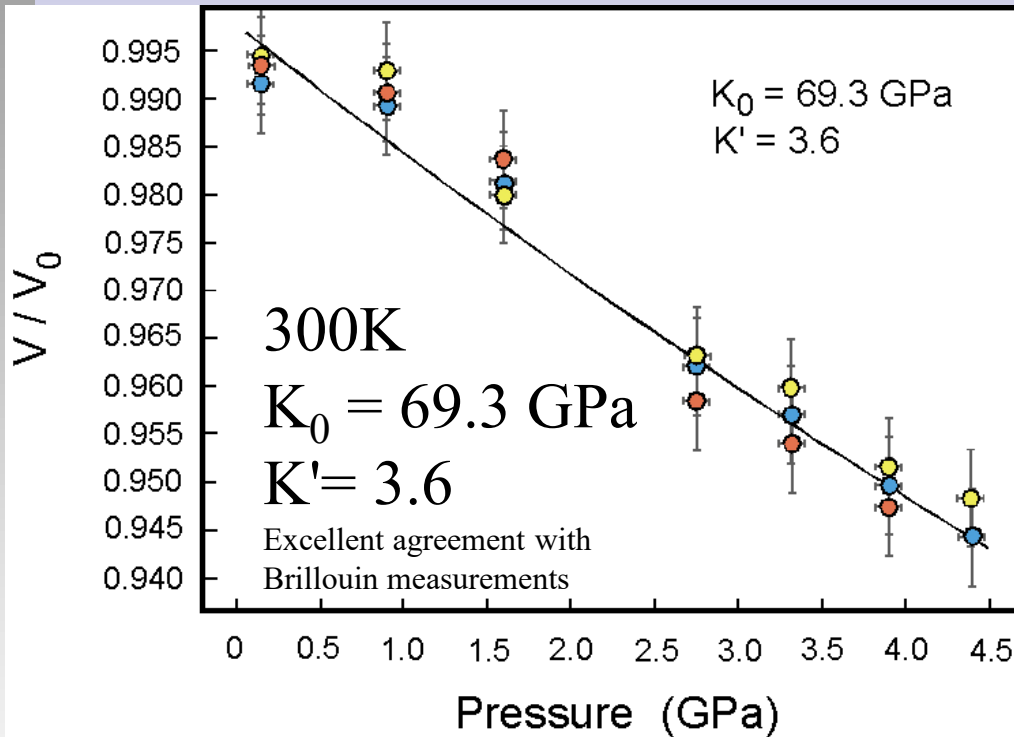


Cell parameter,  
 hence  $V$

Unlike crystalline materials, whose specific density can be evaluated using x-ray diffraction, densities of noncrystalline materials glasses and melts are traditionally more difficult to determine under extreme conditions



# EOS of non-crystalline materials : feasibility of accurately measuring sample volume as a function of $P$ and $T$



Our rotopec provides now a method to address this long-standing difficulty!

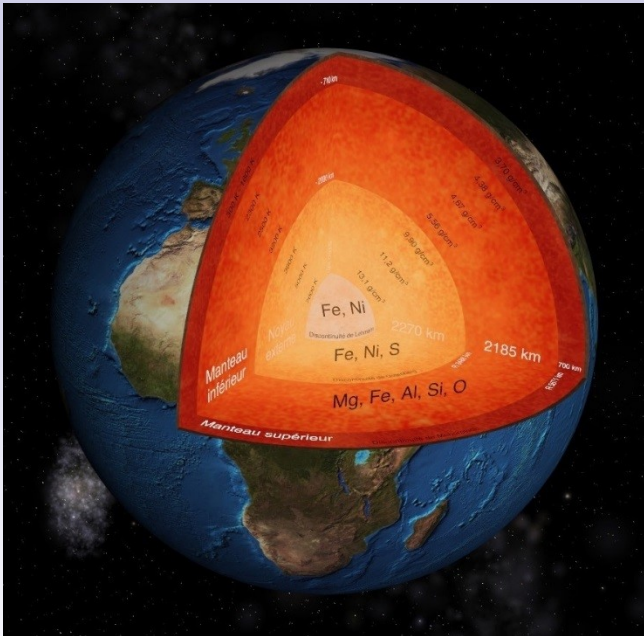


# 3 recent studies with **this novel** rotoPEc portable press

- **EOS of non-crystalline materials** : feasibility of accurately measuring **sample volume** as a function of **P and T**
- **Time resolved 3-D imaging of melt migration under extreme conditions**
- Measurements of **shear modulus** and **attenuation coefficient**  $Q_G^{-1}$  under **mantle conditions** and **seismic frequencies**



# Melt migration in planetary deep interiors



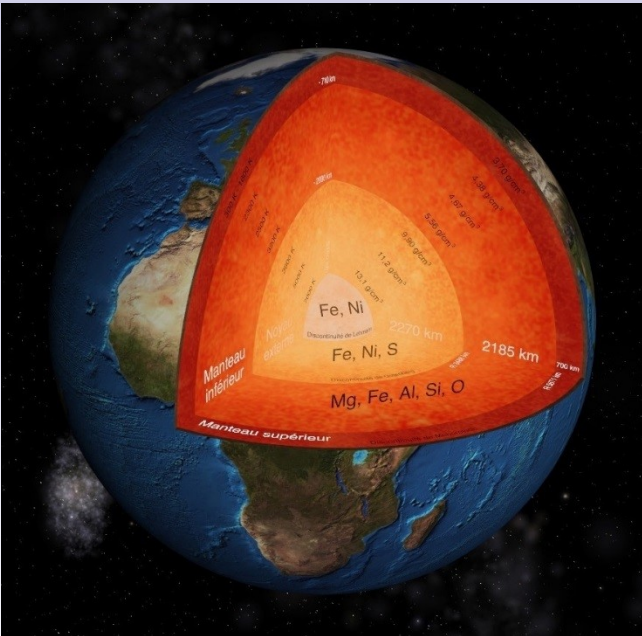
Many geological processes involve **extraction of low degree partial melts** and several questions arise :

At what point **during progressive melting** does the **melt phase become mobile** ?

How does **this partial melt move** through the remaining solid phase, and through the **rest of the Earth** ?

**Melt release** and **melt transport processes** are important because they influence the **geochemistry of the melt phase**. In fact, the **geochemical signature of melts** is typically used to understand where **they formed in planetary interiors** and how

# Melt migration in planetary deep interiors



Many geological processes involve **extraction of low degree partial melts** and several questions arise :

At what point **during progressive melting** does the **melt phase become mobile** ?

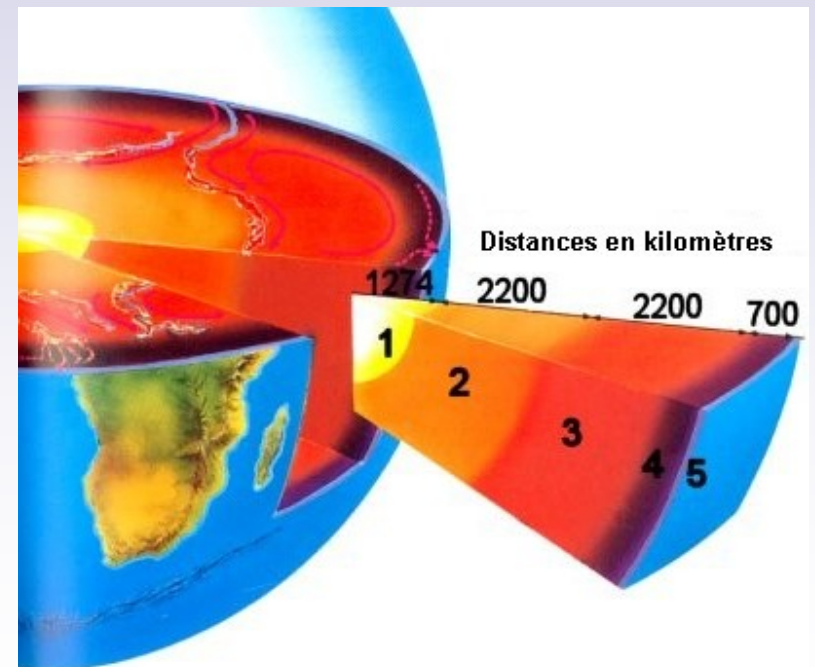
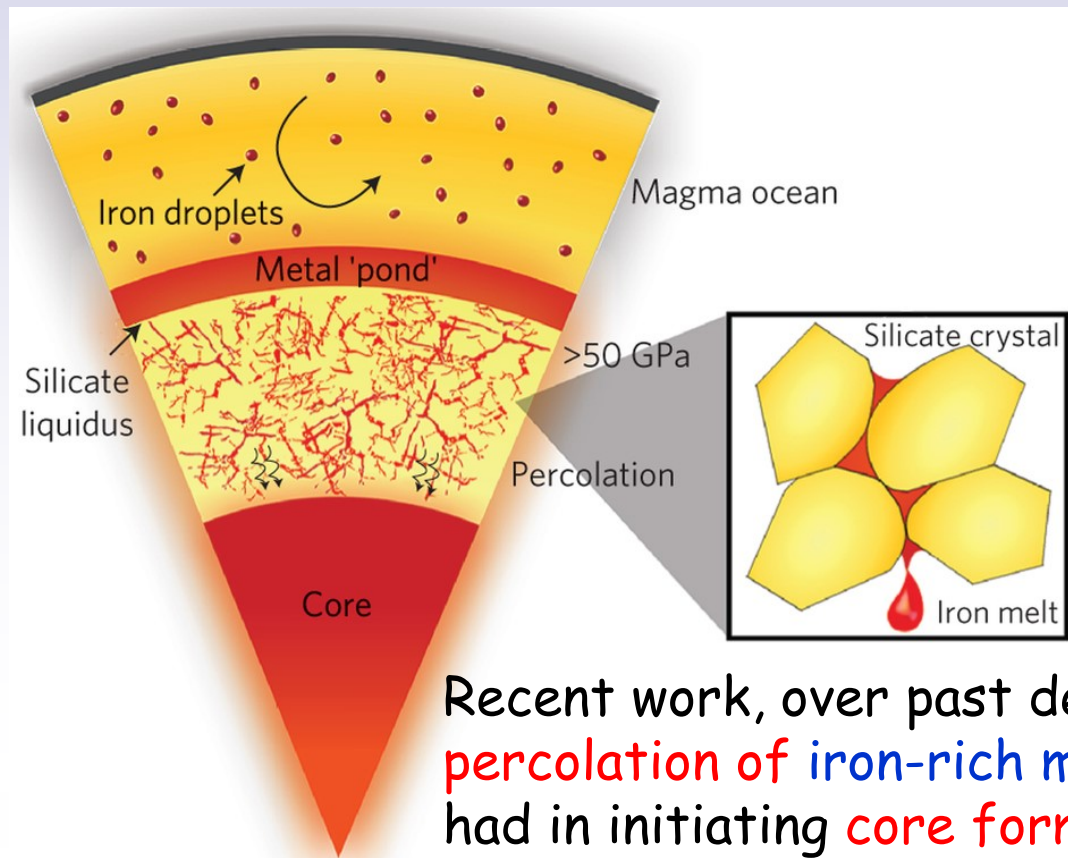
How does **this partial melt move** through the remaining solid phase, and through the **rest of the Earth** ?

We need to **understand melt formation and migration**  
At present, however, our **understanding of these processes** under extreme conditions of planetary interiors **is limited**

# Melt migration in planetary deep interiors

## Core formation ?

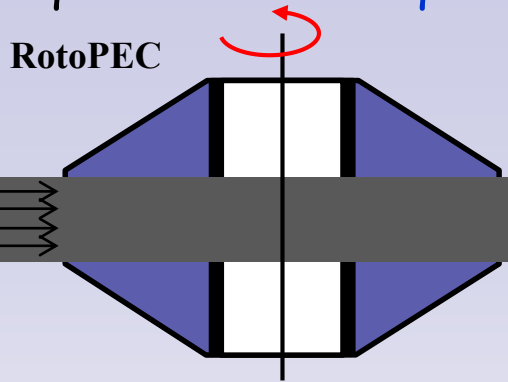
Our understanding of how Earth segregated to **form a metallic core surrounded by a thick shell of silicate** is constrained by our knowledge of metal-silicate segregation mechanisms under high  $P, T$  and stress



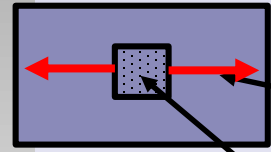
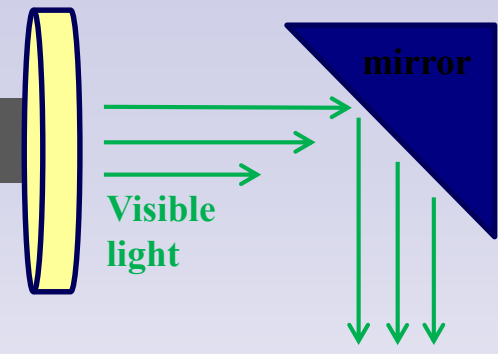
Recent work, over past decade, has highlighted the role that **percolation of iron-rich melts through solid silicate** could have had in initiating **core formation**



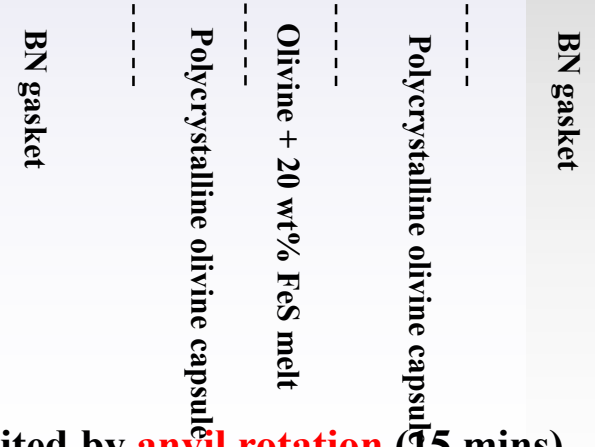
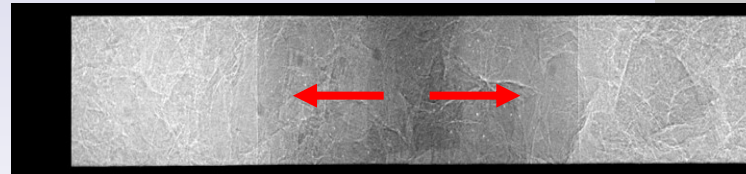
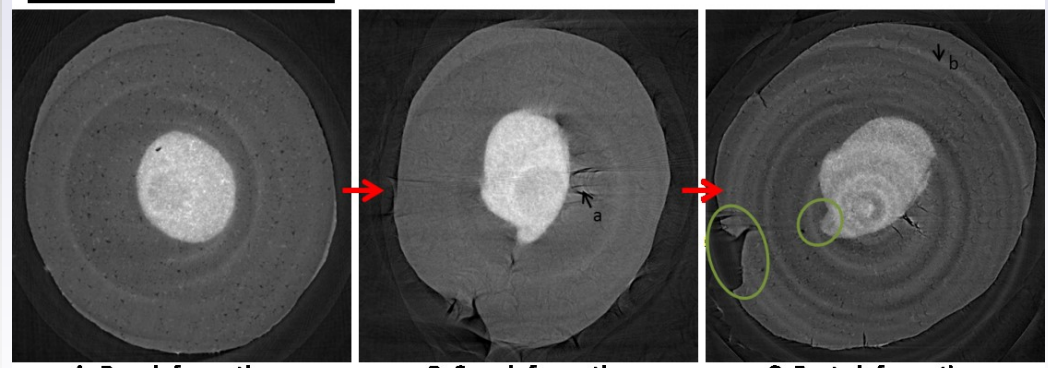
# Visualising melt migration under extreme conditions (HPHT, large deformation) to answer this question : how quickly can planets form metallic cores?



YAG Scintillator



polycrystalline olivine  
olivine + 20% Fe-S melt



Our experiments performed at DIAMOND aimed at investigating the mobility of Fe-S rich melts: Fe-S melts should migrate under extreme conditions from melt-rich central region of sample assembly into surrounding, melt-free polycrystalline olivine

- Scan time limited by anvil rotation (15 mins)
- But samples stable over this time
- Approx 2µm voxel size achievable;

DLS3 time series

hours at max P-T

100rpm deformation  
34°

75°

123°



27926

9

27930\_f4

17

27942

19

27950

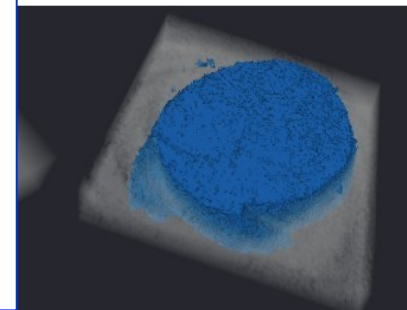
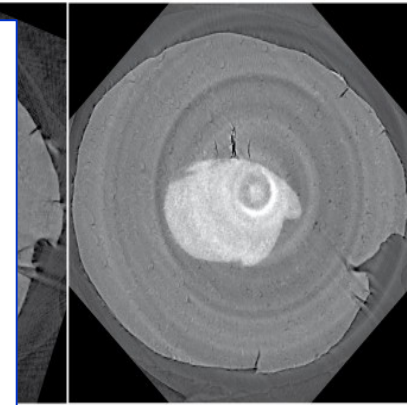
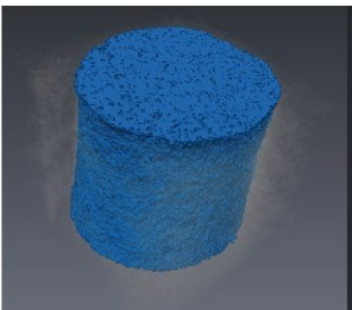
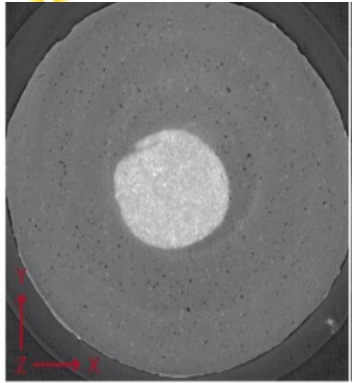
23

27967

For this study, please look :

J. Philippe et al. (2016) Rotating tomography Paris-Edinburgh cell: a novel portable press for micro-tomographic 4-D imaging at extreme pressure/temperature/stress

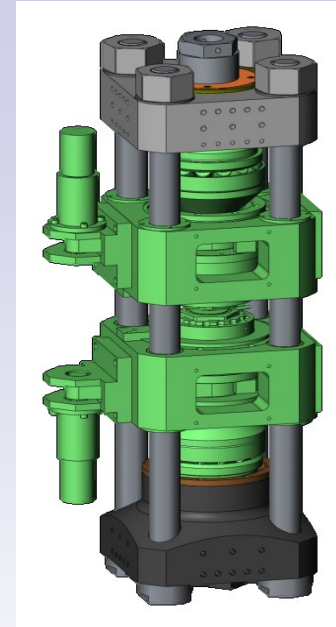
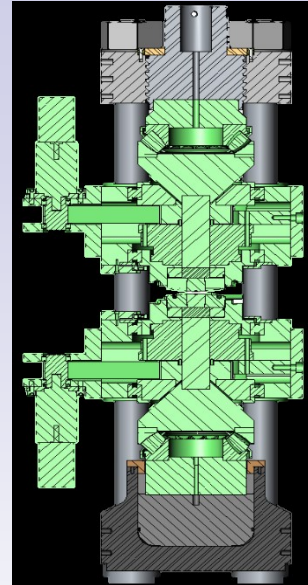
M. Berg et al. (2018) Rapid core formation in terrestrial planets by percolative flow: in-situ imaging of metallic melt under high pressure/temperature conditions. *Frontiers in Earth Science*, 6, 77.



3-D rendering of the melt-rich central part of the sample during progressive annealing under P and then deformation : **deformation enhance melt migration**

# Melt migration in planetary deep interiors

	distance / $\mu\text{m}$	time / s	speed / $\mu\text{ms}^{-1}$
M1a short	300	20	15.0
M1a mid	900	20	45.0
M1a long	1150	20	57.5
M1b short	300	20	15.0
M1b long	800	20	40.0
M2 2d	350	540	0.6
M2 short	400	540	0.7
M2 mid	500	540	0.9
M2 long	1500	540	2.8



Clearly this is all **very preliminary**

However, we can show that we can **extract now with our rotopec press** meaningful data on how melts move and melt migration velocities under extreme P/T/Stress conditions



# 3 recent studies with **this novel** rotoPEc portable press

- **EOS of non-crystalline materials** : feasibility of accurately measuring **sample volume** as a function of **P and T**
- **Time resolved 3-D imaging of melt migration under extreme conditions**
- Measurements of **shear modulus** and **attenuation coefficient**  $Q_G^{-1}$  under **mantle conditions** and **seismic frequencies**

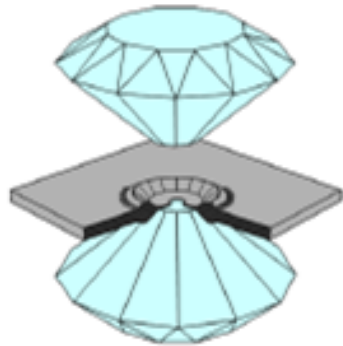




# Conventional experimental methods for the determination of sound velocities in geological materials

Coll. J.P. Perrillat

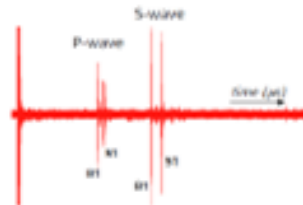
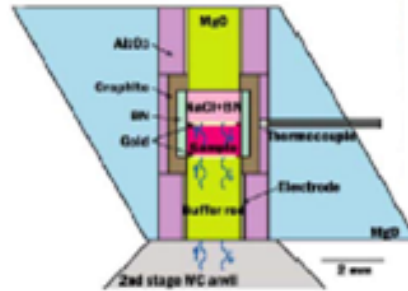
## Static compression measurements



X-Ray & neutron Diffraction

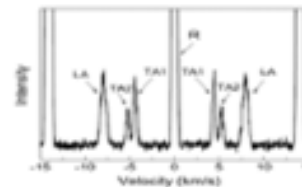
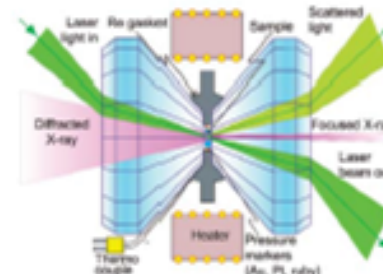
$V(P, T)$   
Bulk modulus ( $K$ )

## Ultrasonic methods



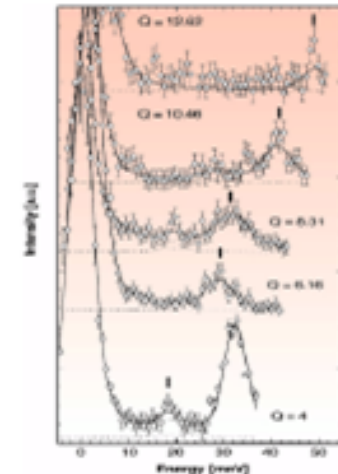
$V_p - V_s$   
Bulk modulus ( $K$ )  
Shear modulus ( $G$ )

## Brillouin Light scattering



$V_p - V_s$   
Bulk modulus ( $K$ )  
Shear modulus ( $G$ )

## Inelastic X-ray scattering



$V_p - V_s$   
Bulk modulus ( $K$ )  
Shear modulus ( $G$ )

frequency

MHz

GHz

THz

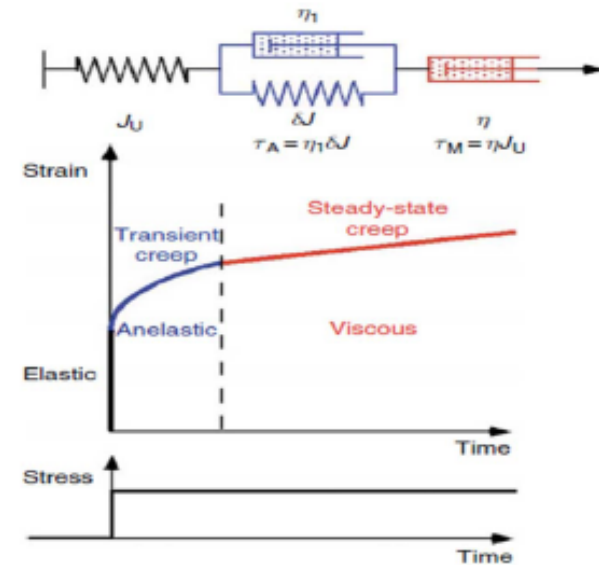
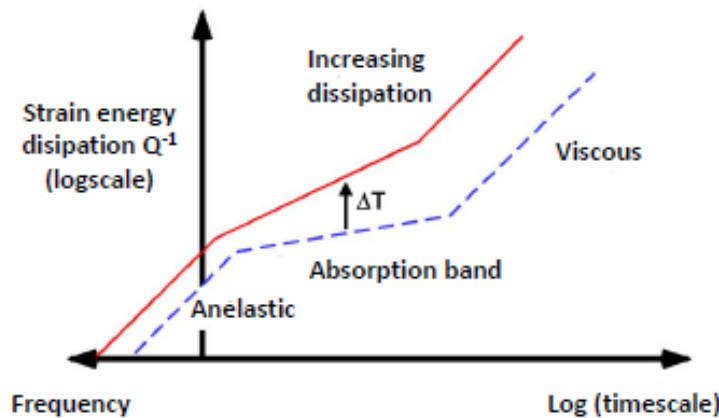
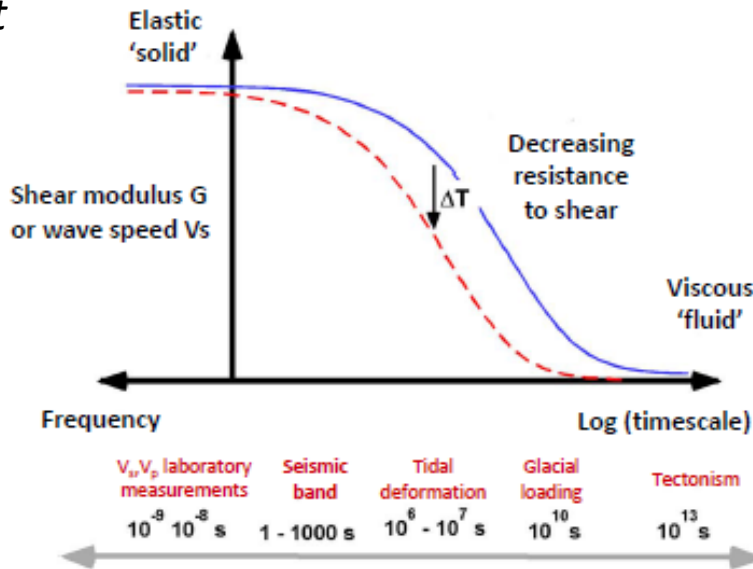
...but in fact teleseismic waves propagate at lower frequencies mHz - Hz





# Mantle rocks are visco-elastic solids, with anelasticity

Coll. J.P. Perrillat



Jackson [2005]

**Crystallographic mechanisms:**  
 motion of dislocations,  
 grain boundary sliding,  
 phase transitions, melting, etc...

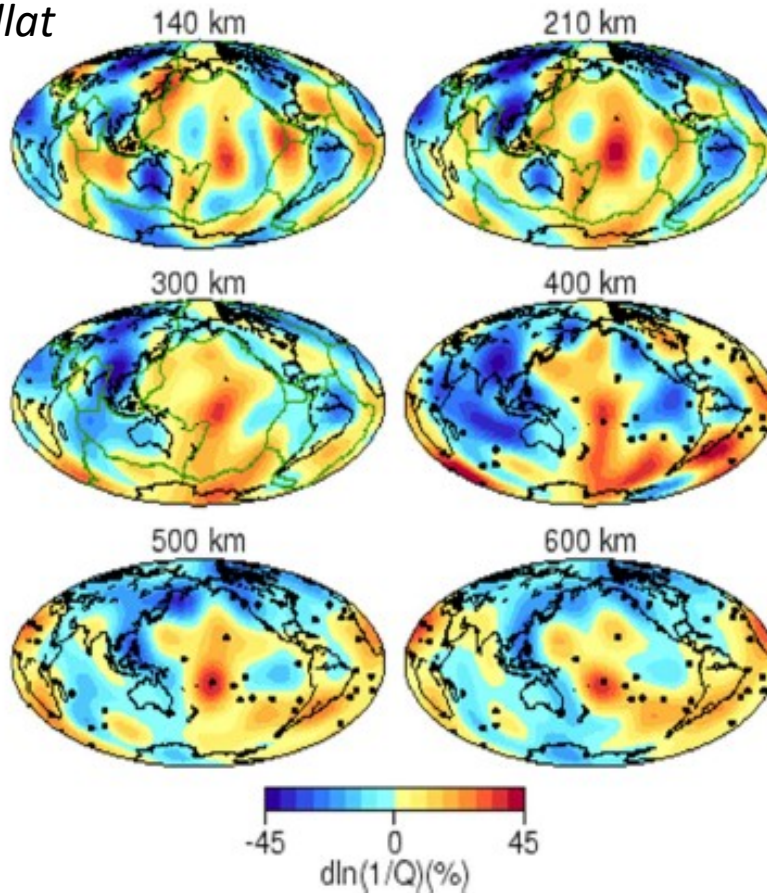
**Attenuation of seismic waves**  
 (« intrinsic attenuation »)



# Measurements of **shear modulus** and **attenuation coefficient $Q_G^{-1}$** under **mantle conditions** and **seismic frequencies**

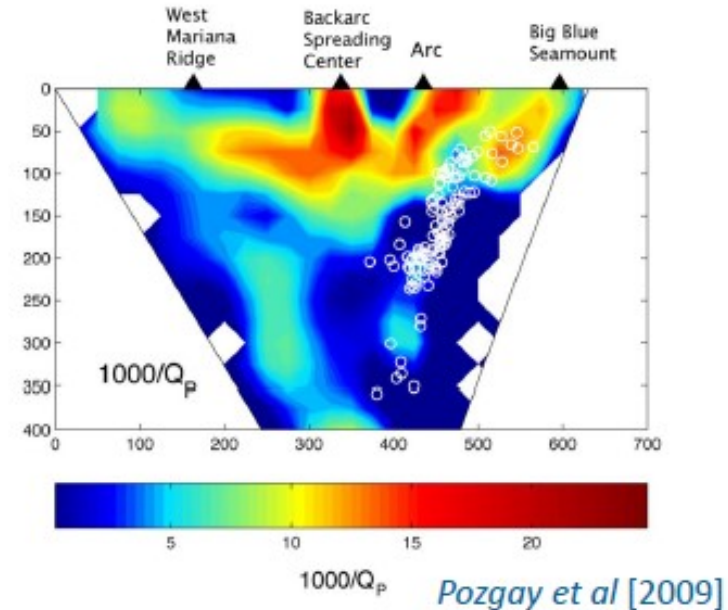
Coll. J.P. Perrillat

## Global 3D models



Gung & Romanowicz [2004]

## Regional studies



Attenuation furnishes important constraints on temperature, water/melt content, grain sizes... however we lack a mineral database for attenuation coefficient at mantle conditions and seismic frequencies because up to now we lacked an experimental method for these HPHT measurements !

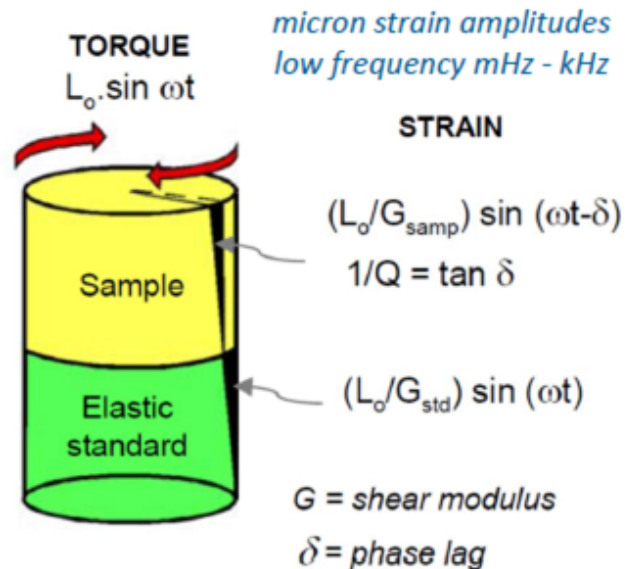
Interpretation of **seismic profiles** and **tomographic models of the Earth's mantle** requires HPHT exploring the viscoelastic behaviour of rocks / minerals at **low strains** and **seismic frequencies**



Coll. J.P. Perrillat

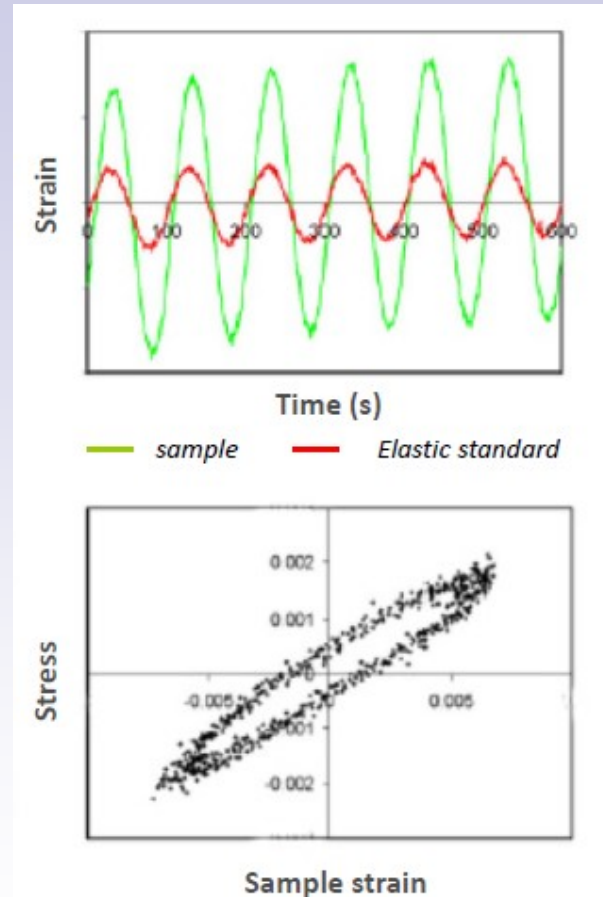
# The torsional forced-oscillation method, a cyclic loading technique

## Torsional forced-oscillation method



Shear modulus  $G_{\text{samp}} = \text{amplitude ratio} \times G_{\text{std}}$

Attenuation  $Q_G^{-1} = \text{tangent of the phase lag}$   
area of the stress - strain ellipse



The **shear modulus** is derived from the **amplitude ratio** and the **attenuation** of the sample is determined from the **phase lag** between the **deformation of the elastic standard and the sample**





# Torsional forced-oscillation method : the next step

Coll. J.P. Perrillat

Increasing the P-T range of G and  $Q_G^{-1}$  measurements to mantle conditions

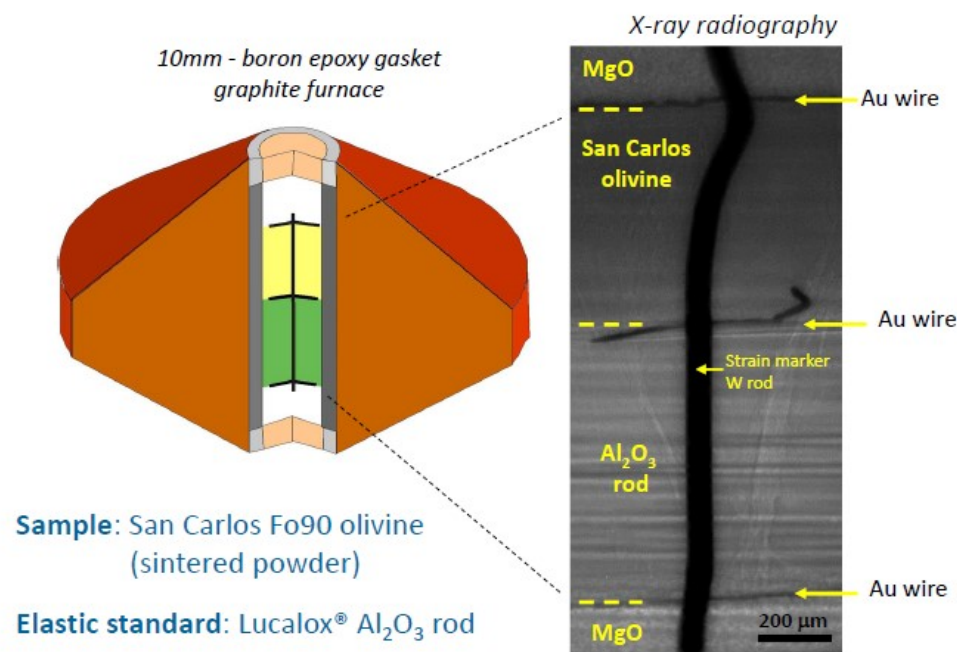
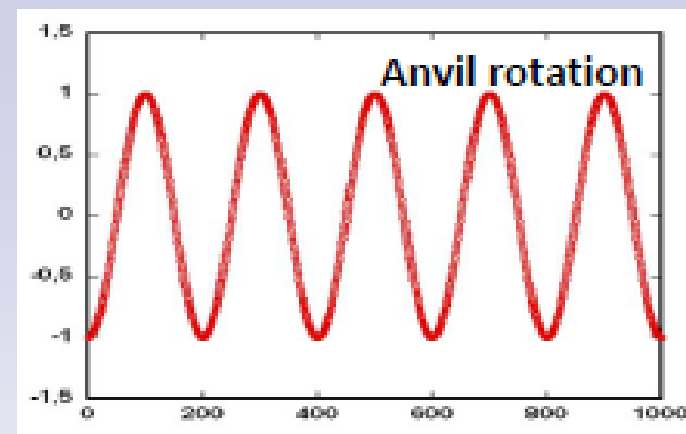
## Requirements:

- High pressure device capable of applying a sinusoidal torque on the sample under load

= "RoToPEc" module

- A technique to measure the sample/standard strain with high spatial & time resolution

= X-ray radiography (synchrotron)



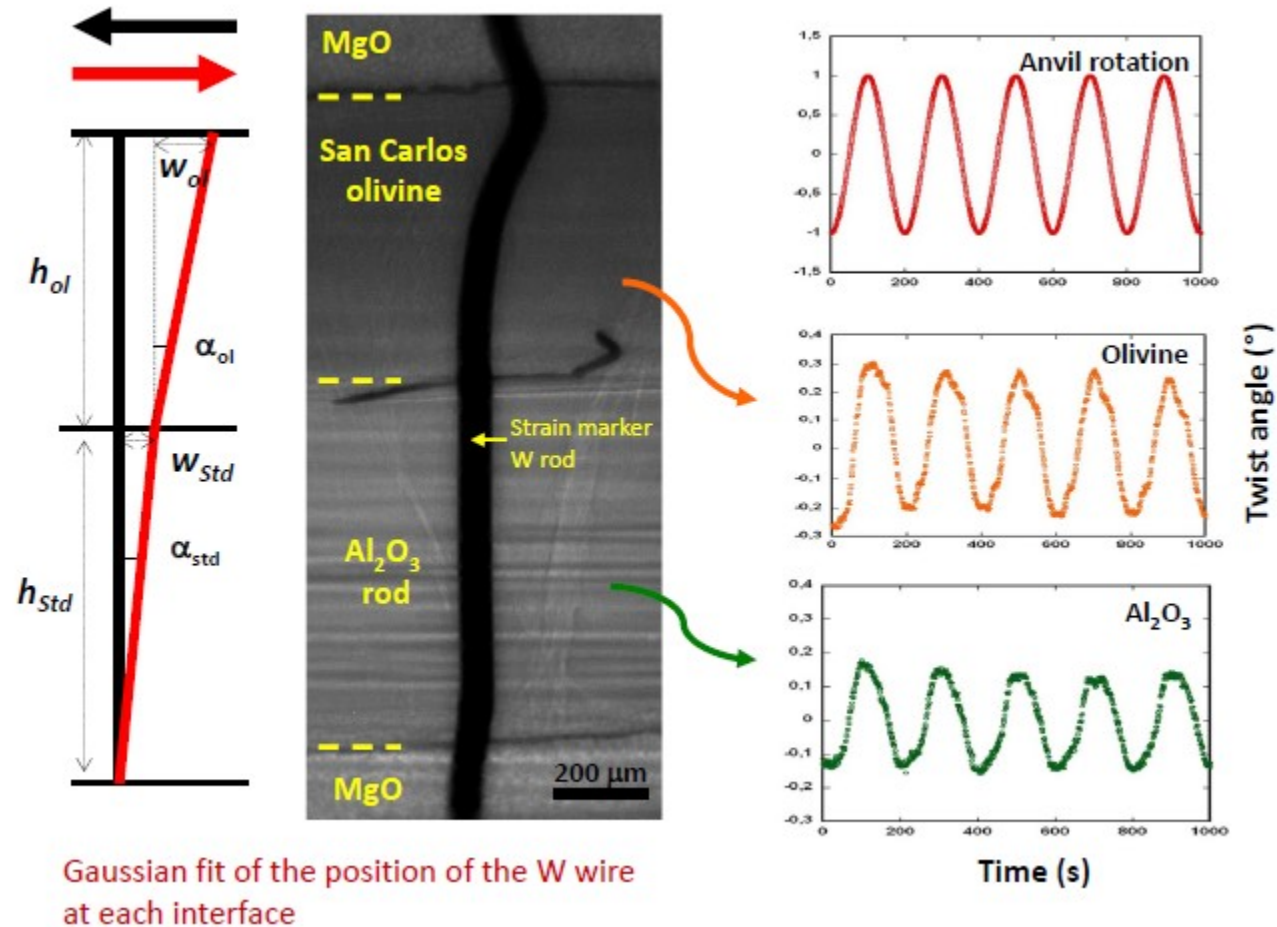


# Measurements of **shear modulus** and **attenuation coefficient $Q_G^{-1}$** under **mantle conditions** and **seismic frequencies**

Coll. J.P. Perrillat

A sinusoidal shear was set to the sample by the **periodic rotation** of the bottom anvil of the RoToPEc module by  **$0.5 - 5^\circ$**  at periods of **100 - 1000 s**. While the sample was under cyclic loading, **X-ray images** were recorded with a time resolution as low as **100ms** in order to **monitor the deformation** of both the sample and elastic standard

Data are still processing but the **feasibility has been clearly demonstrated...**

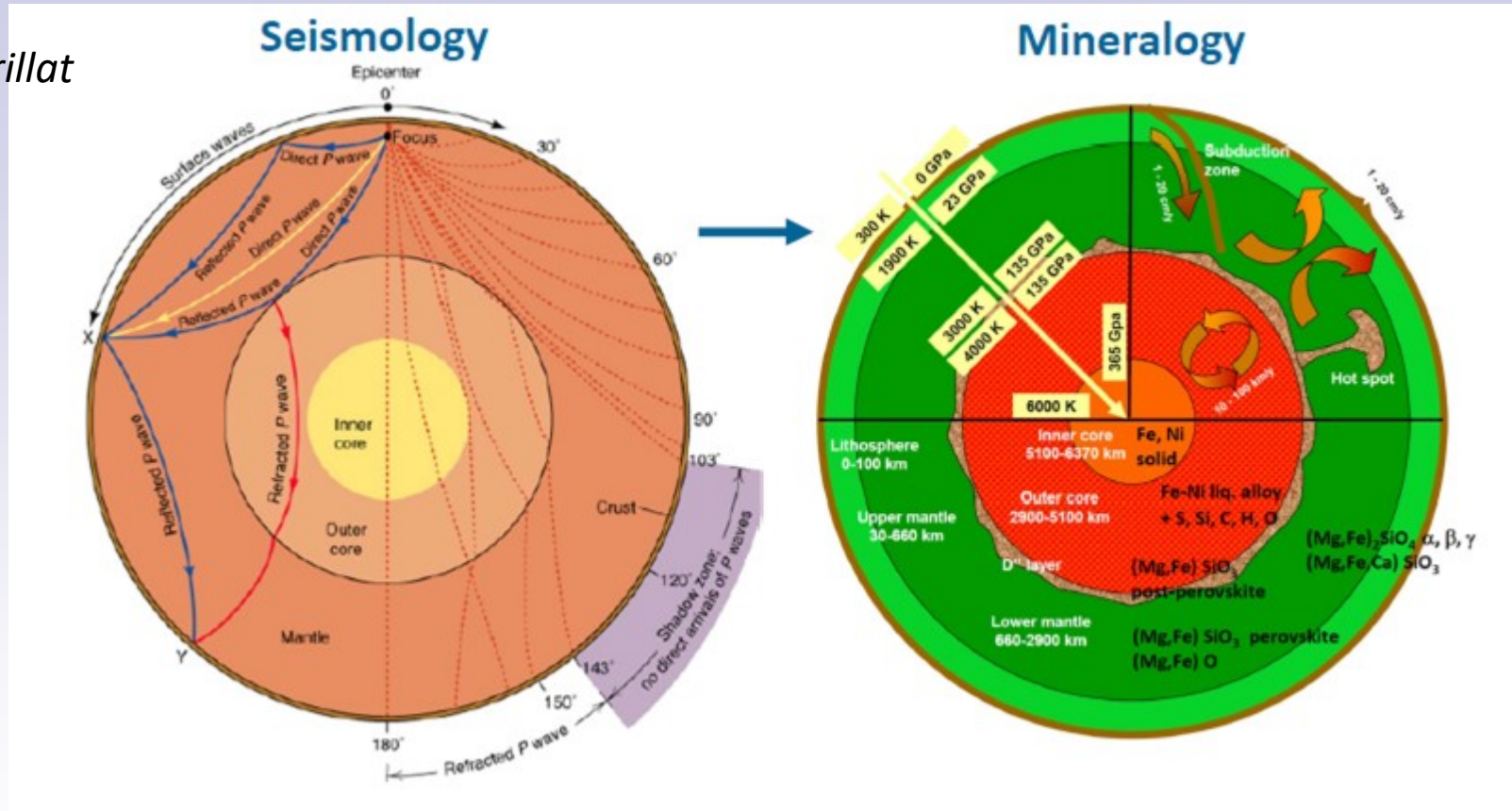


ESRF Experimental report n° ES 33



Coll. J.P. Perrillat

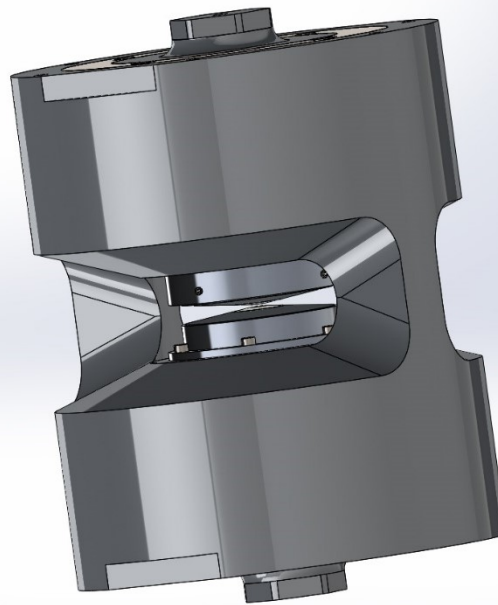
Measurements of **shear modulus** and **attenuation coefficient  $Q_G^{-1}$**  under **mantle conditions** and **seismic frequencies**



**Attenuation and shear wave dispersion of Earth materials** can be investigated at **mantle conditions** and **seismic frequencies**



# PERSPECTIVES



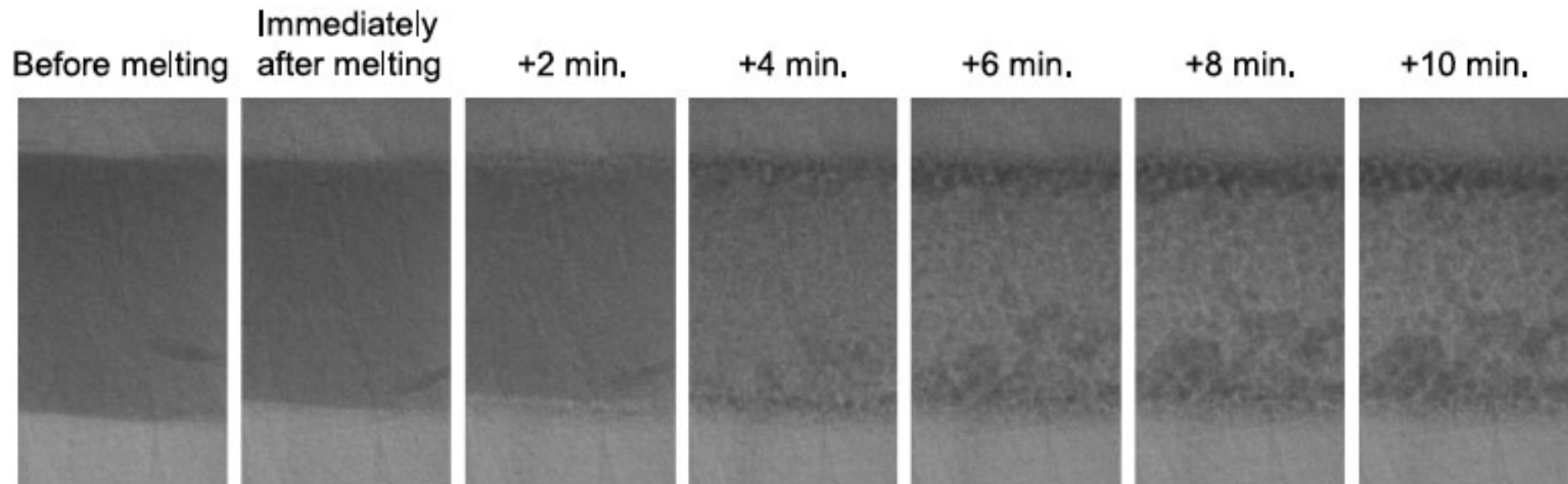
Development of a new **Ultrafast Tomography Paris-Edinburgh Cell**  
**UTOPEC**

Our current acquisition times of the order of 10 minutes per tomogram **represent a significant limitation for dynamic studies of samples** that evolve at a rate that is **greater than this acquisition time**, or for samples which are mechanically **unstable under HP and HT** conditions

**Fast Processes**



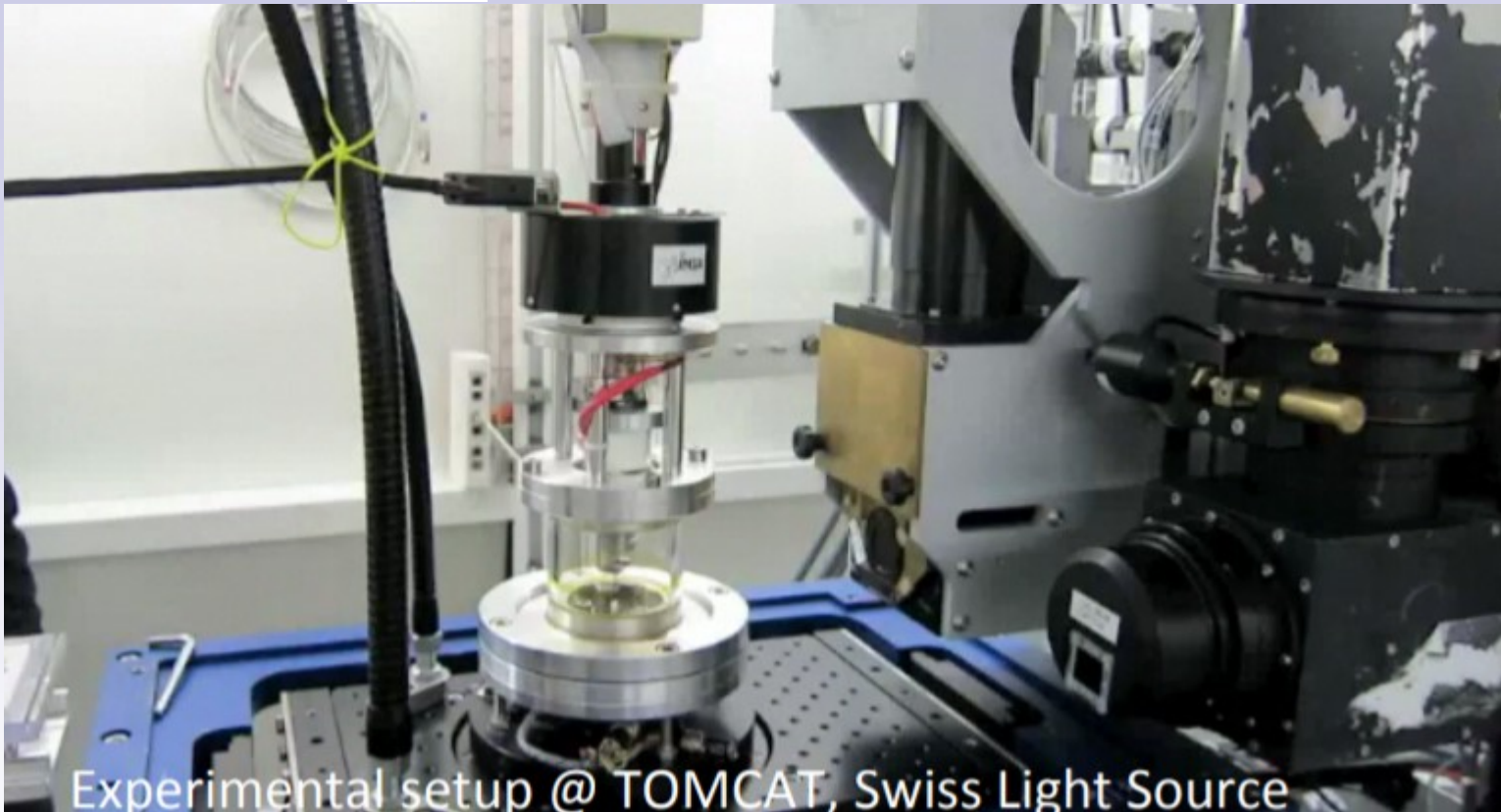
We need **Fast tomography**



200  $\mu\text{m}$

demixion of a liquid within 10 minutes

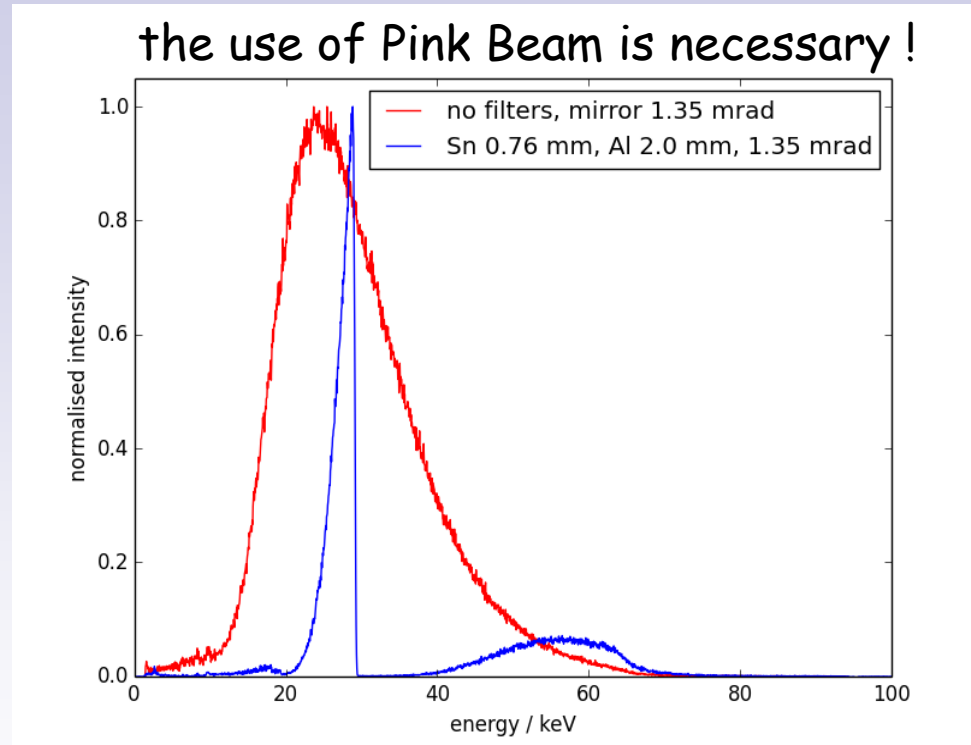
- Fast detector



Recent developments in **synchrotron tomography** have greatly reduced the **acquisition time** that can be expected, and **second or even sub-second tomography** is possible. For that **Fast detectors are now capable of kHz frame rate**. This enables increasingly fast **dynamic processes** to be studied in situ such as **fracture**, **solidification** or **phase transformations**

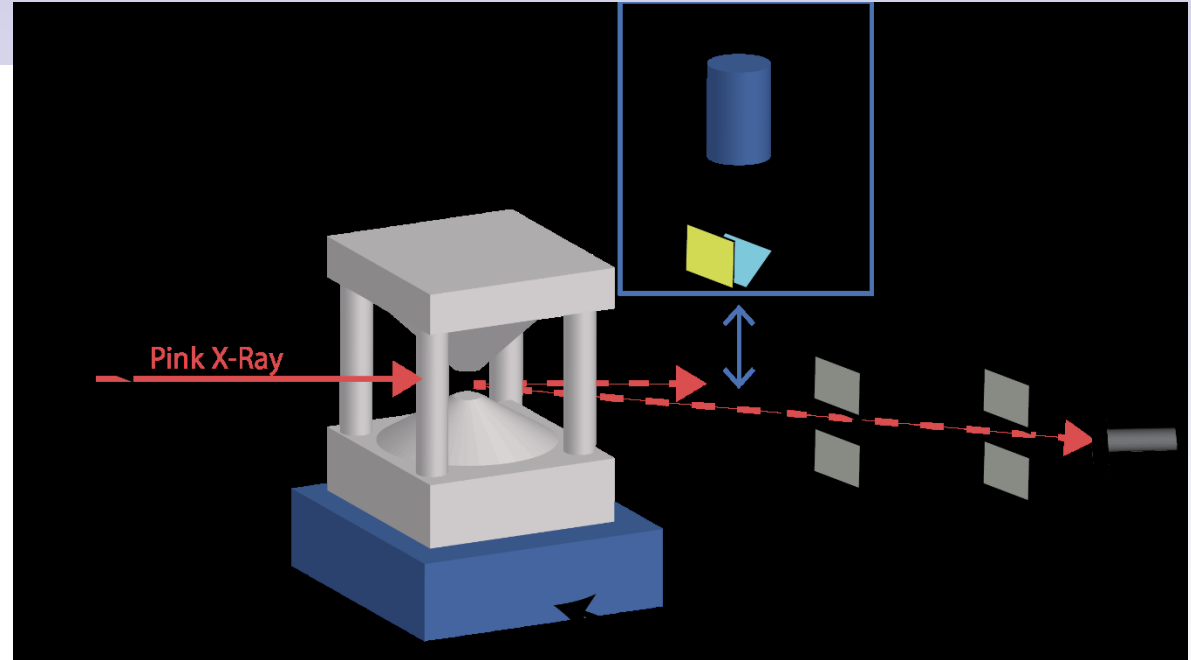
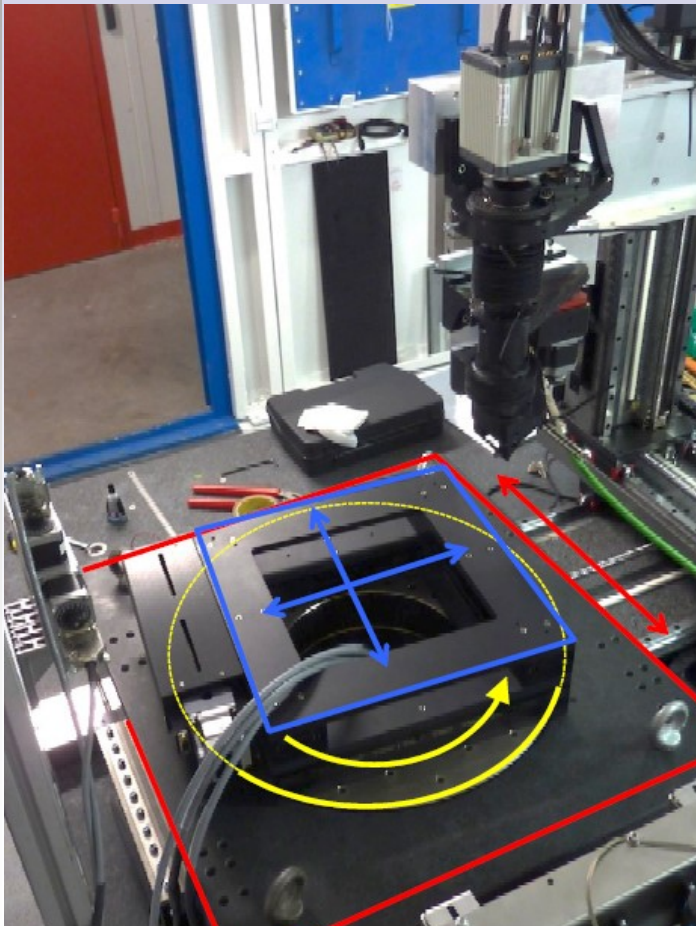


- Fast detector 😊
- High Brilliance of synchrotron source 😊



Crystal monochromator optics on a synchrotron beamline typically transmit **less than 0.1% of the incident spectrum**, and so greatly **decrease the flux available**. For this reason, **monochromatic beams are less suitable for dynamic studies**.  
By using **high-flux wide-bandwidth X-ray illumination (pink beam)**, the exposure time per **radiograph can be reduced accordingly**

- Fast detector 😊
- High Brilliance of synchrotron source 😊
- Fast (and precise) rotation of the press 😊

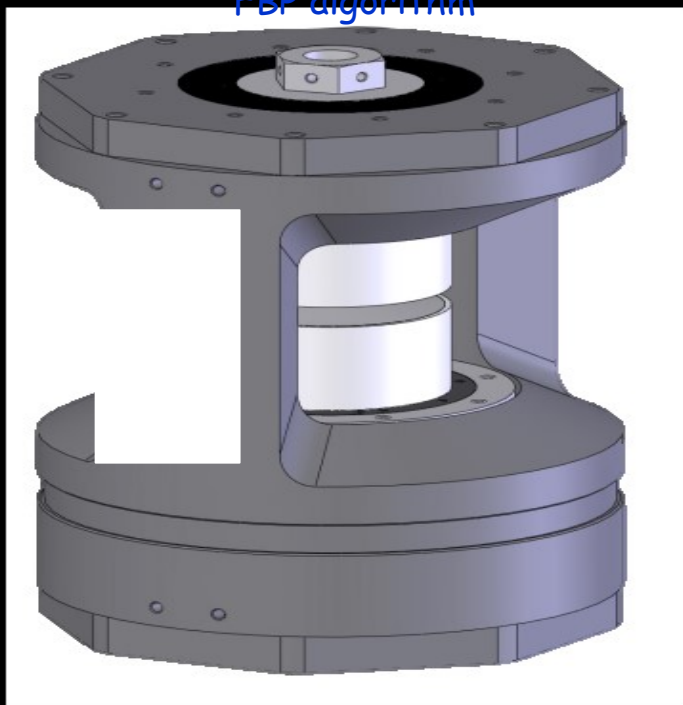


The PSICHE tomograph is based on a **high-precision highload-capacity rotation**. It has a maximum **load capacity of more than 50 kg**, and is capable of rotating **at 60 rpm with an eccentricity of less than 150 nm**

- Fast detector 😊
- High Brilliance of synchrotron source 😊
- Fast (and precise) rotation of the press 😊
- A new PE press with severe requirements

Since now we would like to rotate the press itself!

FBP algorithm



170 mm

The new press should allow working up to 15 GPa and 2000 K. So we need at least a press with 80 tons capacity and 60mm anvils diameter

In order to increase the quality of the reconstructed images, the angular opening will be maximized using a two-column system, like VX, but with a 165° angular opening (!! ) in order to reduce the reconstruction artefacts currently observed

The press should also have a smaller diameter (170 mm) which will allow a smaller sample detector distance to be used. This is particularly important, as the limiting spatial resolution deteriorates with increasing sample-detector distance

The weight of the press should be less than 50 kgs





Rotary Union

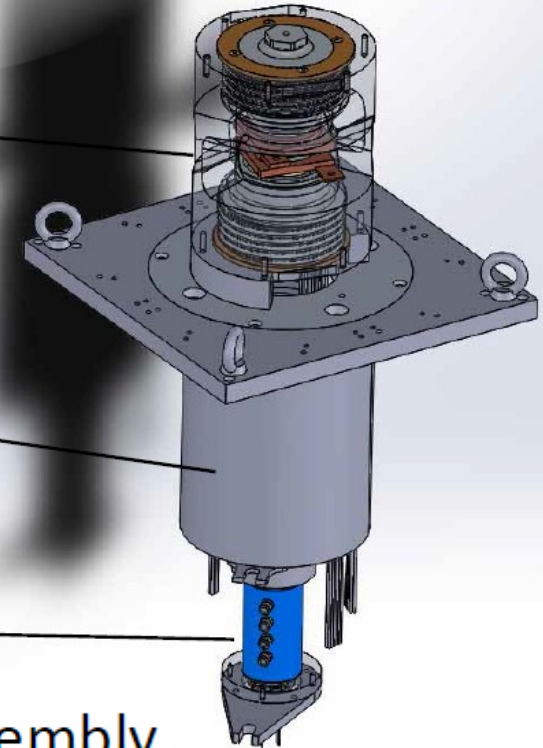
Coll. G. Hamel

its adaptation  
to **PSICHE**  
beamline

power (300 A) and signal rotary joint

hydraulic (cooling) rotary joint

UToPEc



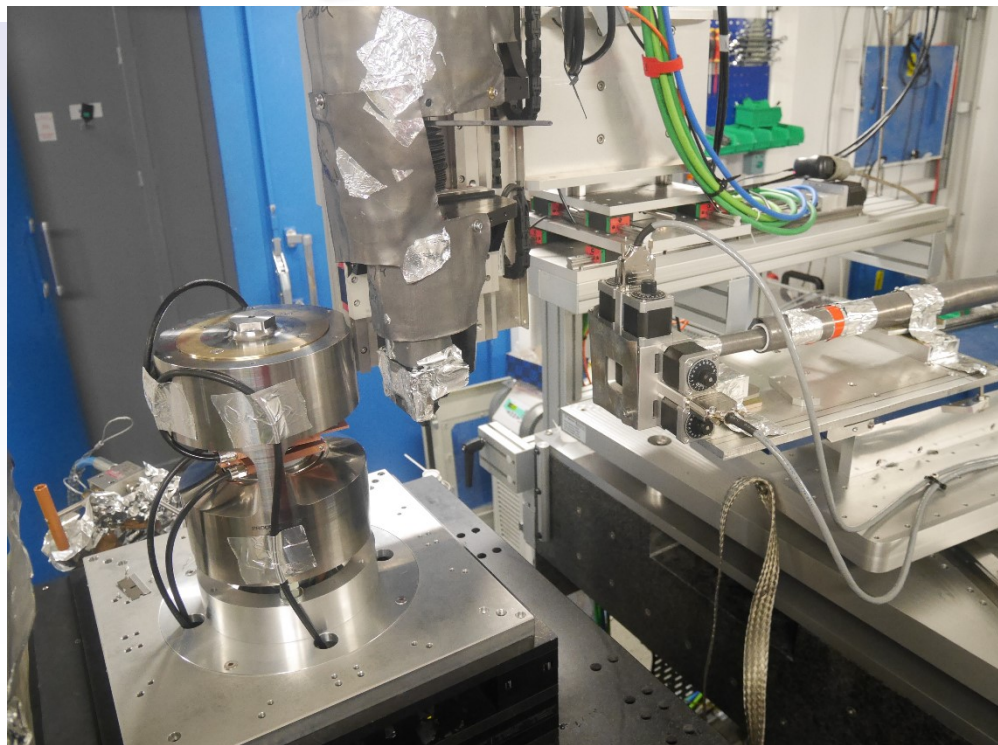
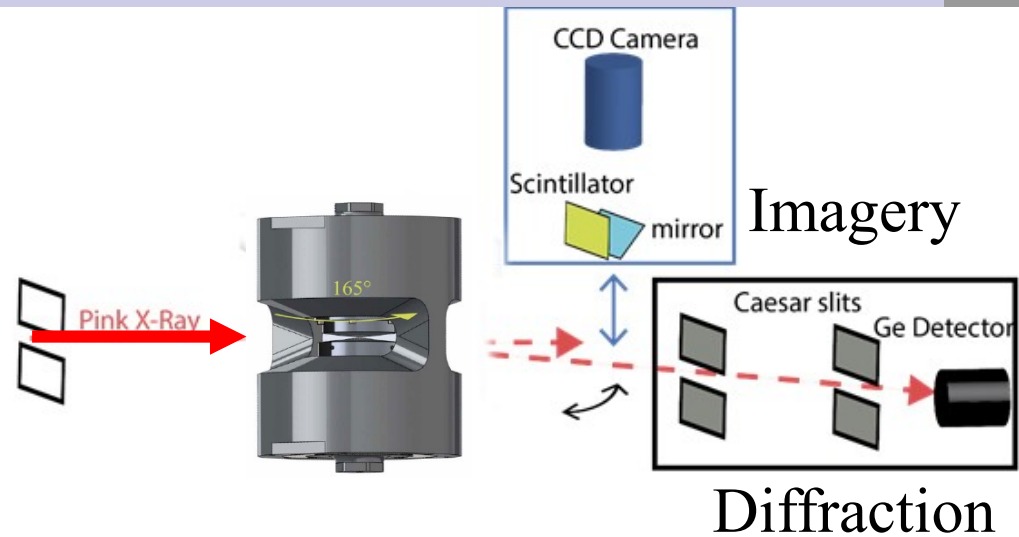
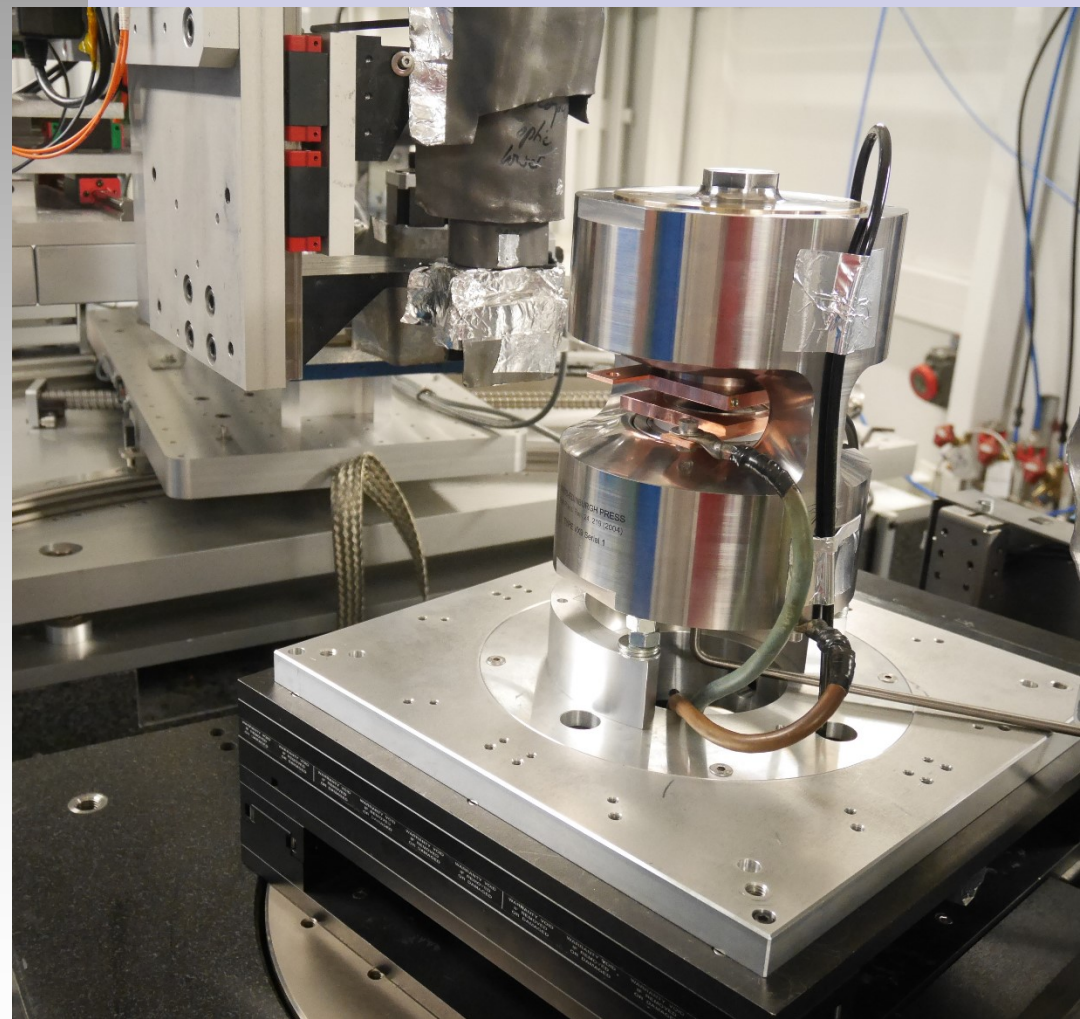
Rotary joints assembly

Basically we can rotate **the press continuously**

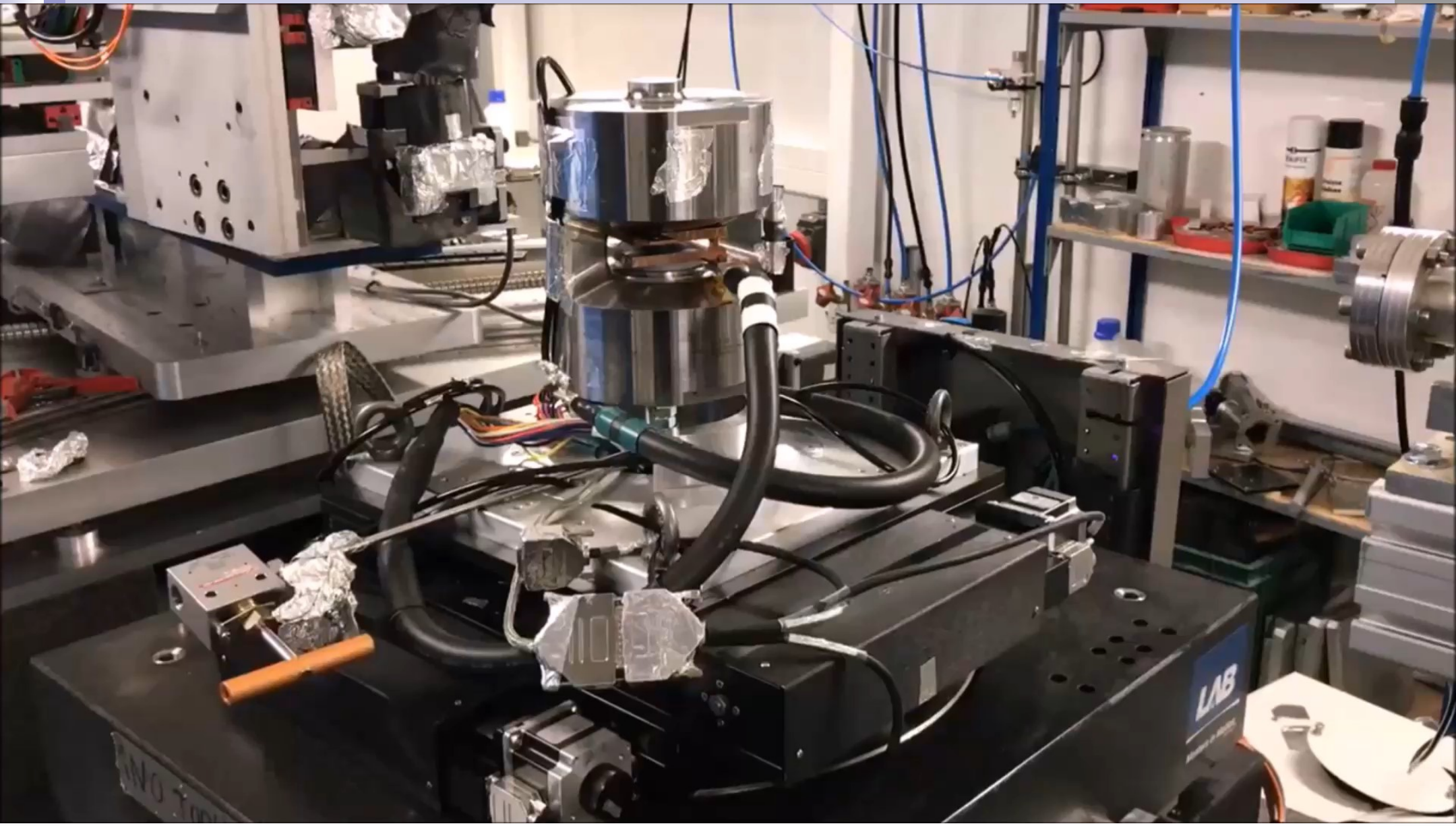
With this improvement, it is possible to acquire in principle **a full tomogram in 0.5 s**, corresponding to **the maximum rotation speed of the stage**

**UltraFast acquisitions: < 1 tomo / s**

# UTOPEC

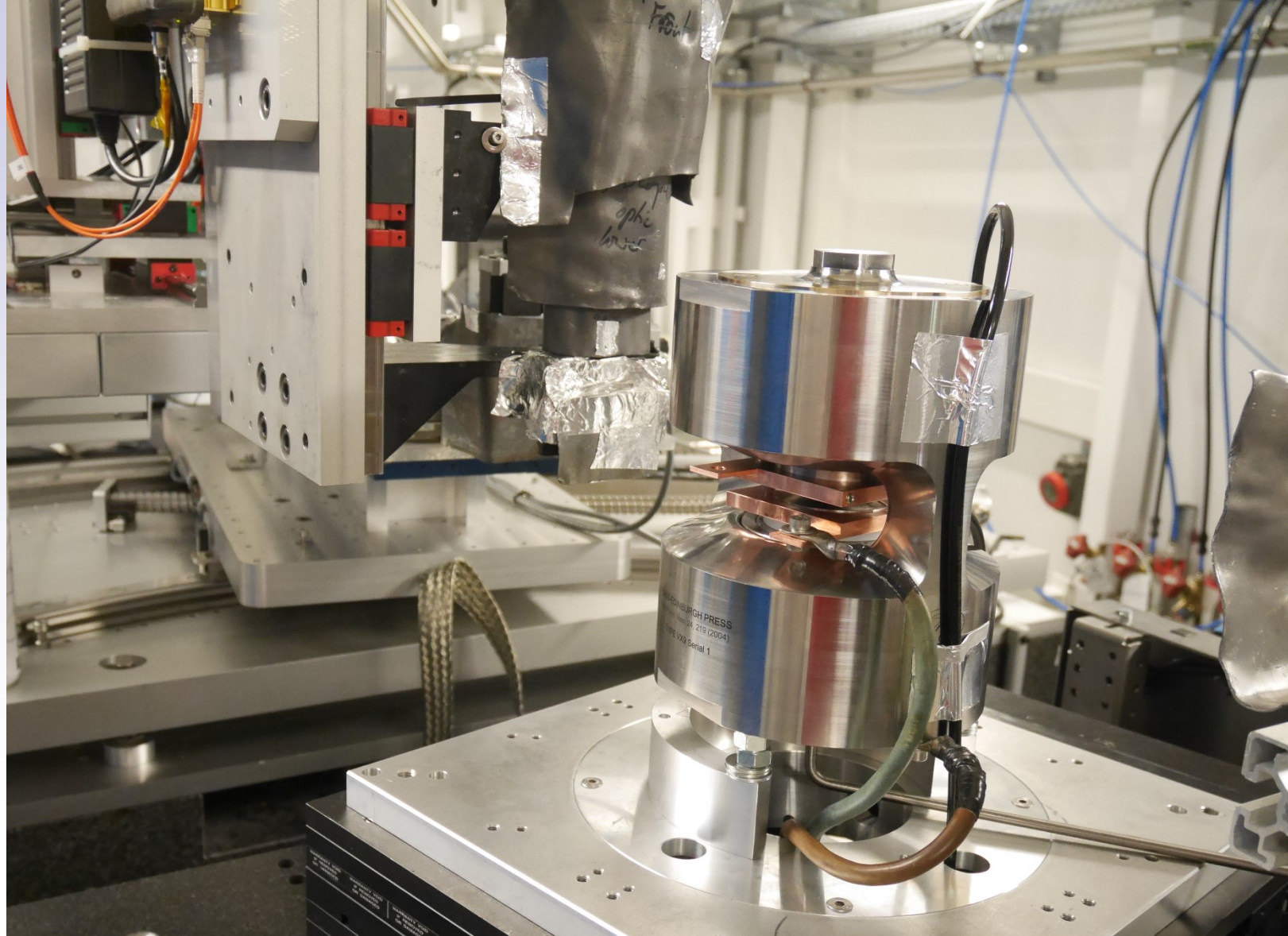




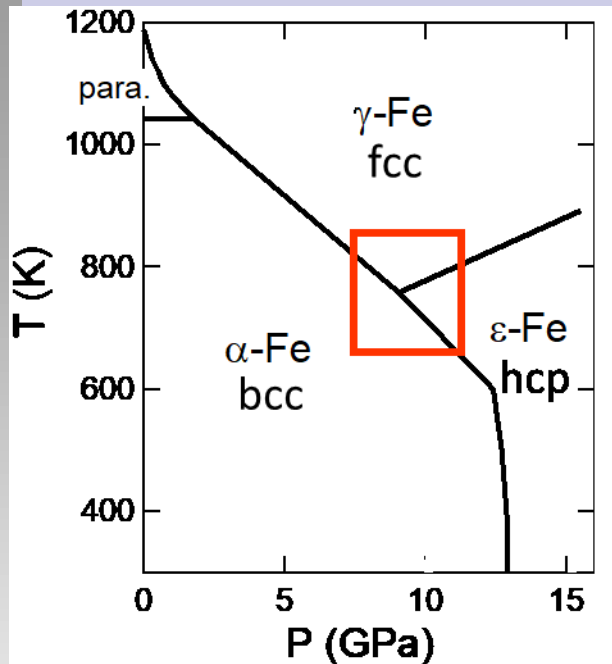




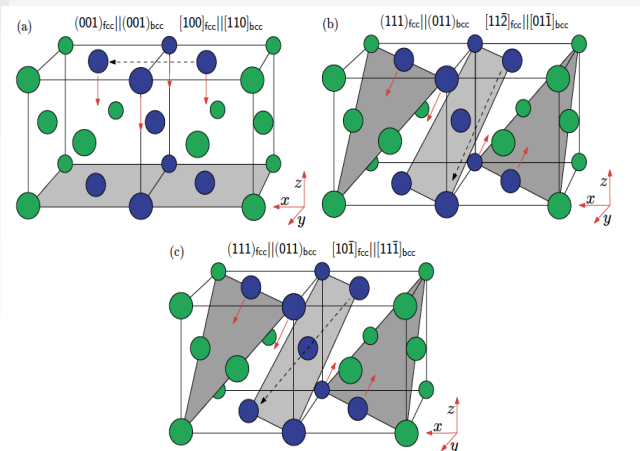
# 2 recent FAST HPHT Tomography studies



# Martensitic Transitions of Iron at HP-HT



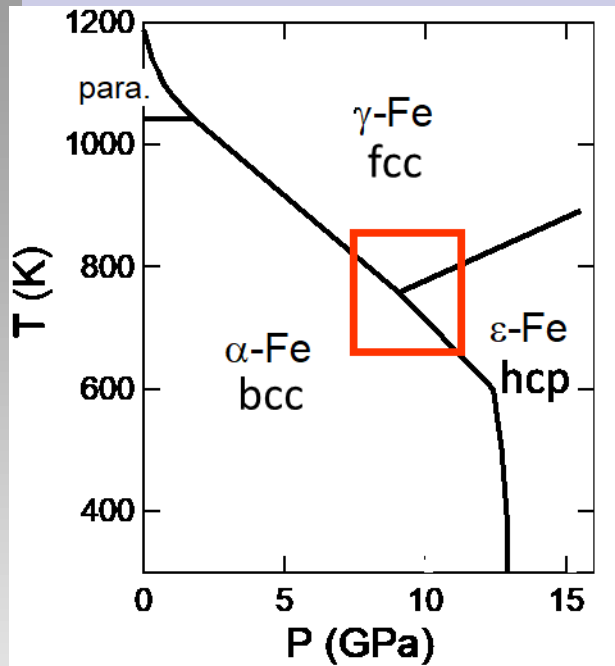
**AIM** : We studied the mechanisms involved in the HP HT martensitic transformations in iron. These martensitic transitions have profound implications in Fe-based materials technology as well as in planetary science with iron being the main component of terrestrial planetary core. By imaging in situ the orientation of the coexisting phases, we are trying to determine the orientations relations and habit planes for both transitions



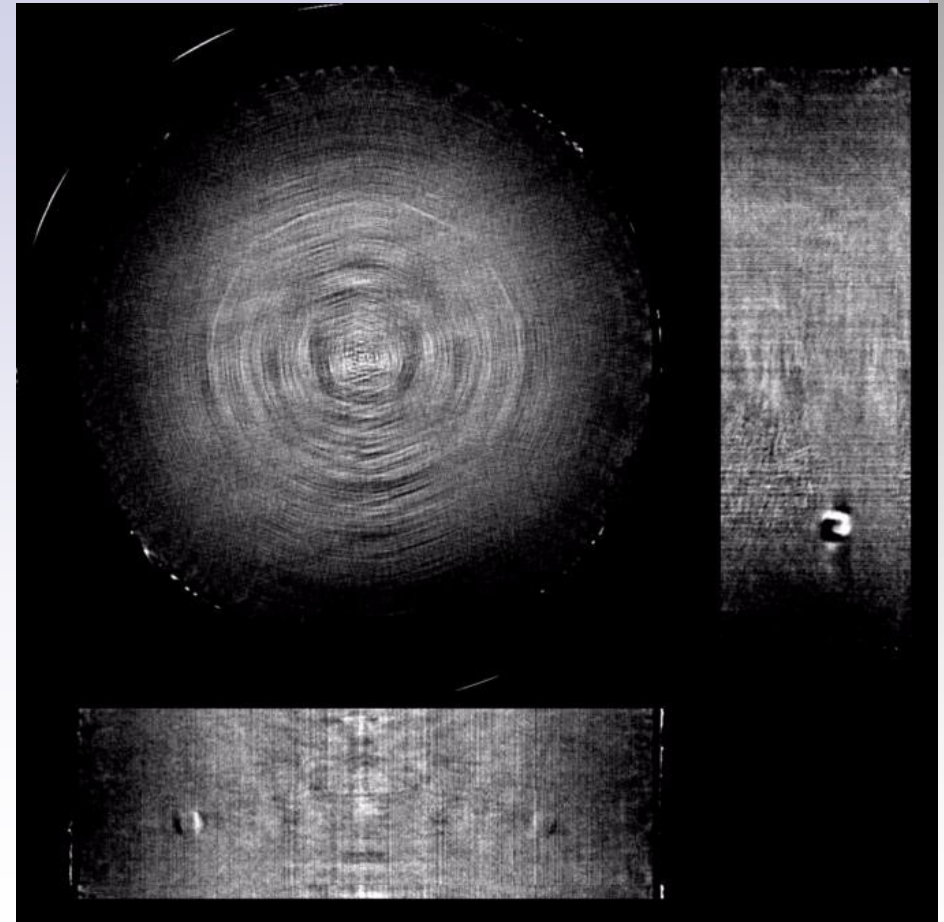
In the literature, three mechanisms of transformation have been proposed involving different orientation relationship between the fcc parent and the bcc phase and our study will be able to discriminate these mechanisms

Coll. E. Boulard, A. Deawele et P. Loubeyre (CEA)

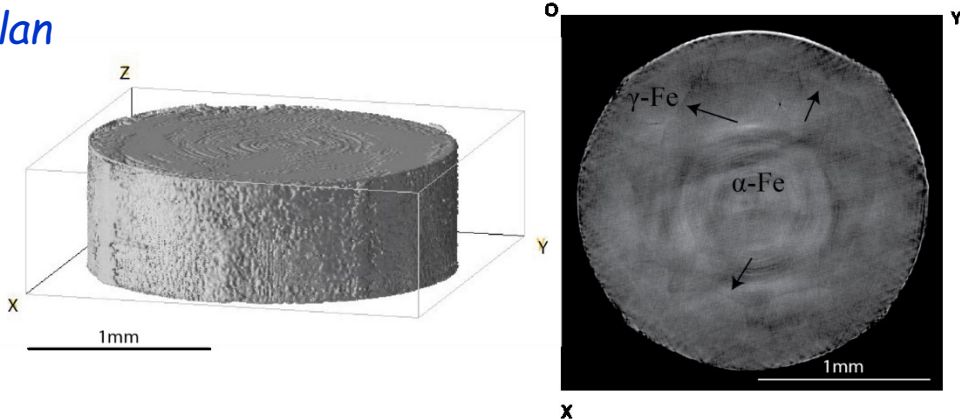
# Martensitic Transitions of Iron at HP-HT



the **gamma phase** arrives by the edge



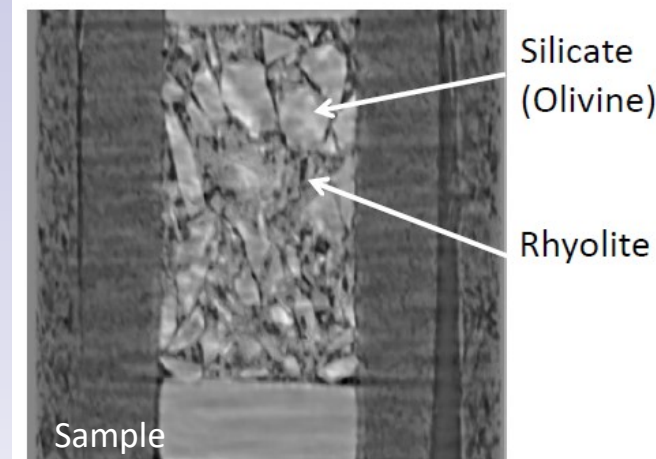
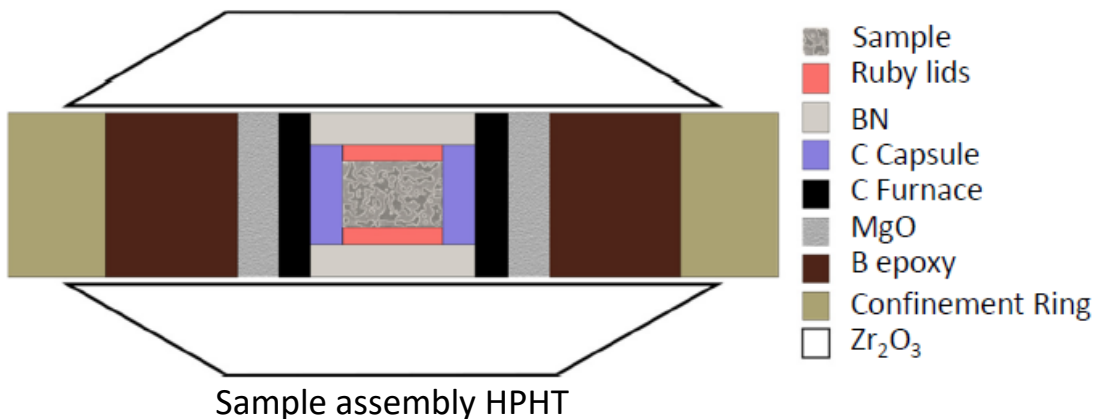
the **coexistence of the bcc phase and the fcc phase** at the transition and thus the **determination of the habit plan**



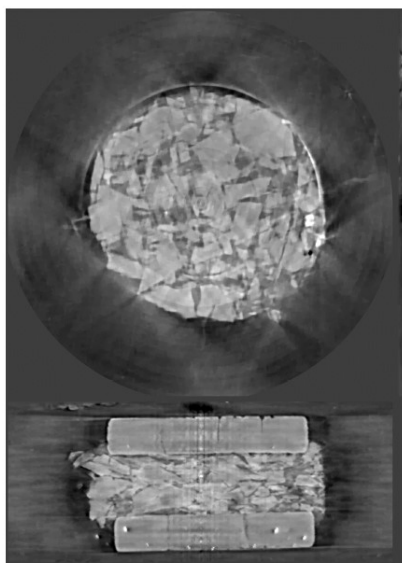
Our experiments provided **for the first time information** on the **orientations relations and habit planes** for  $\alpha$ - $\gamma$  and  $\alpha$ - $\epsilon$  transitions **in iron under HPHT**



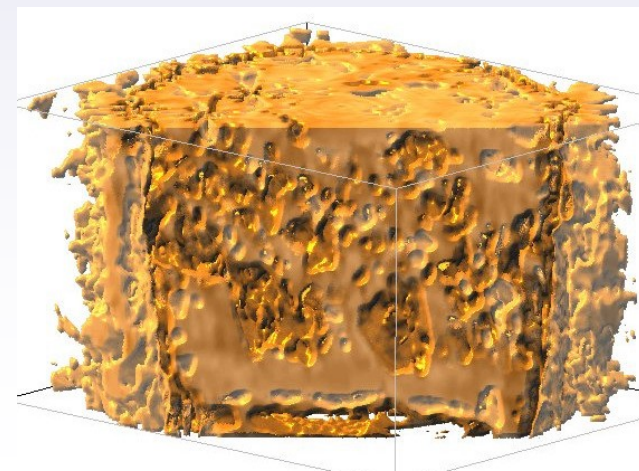
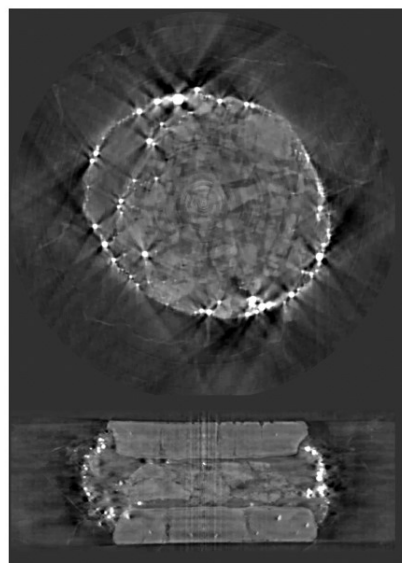
# Transport of magma under mantle conditions



3 GPa - 1200 K



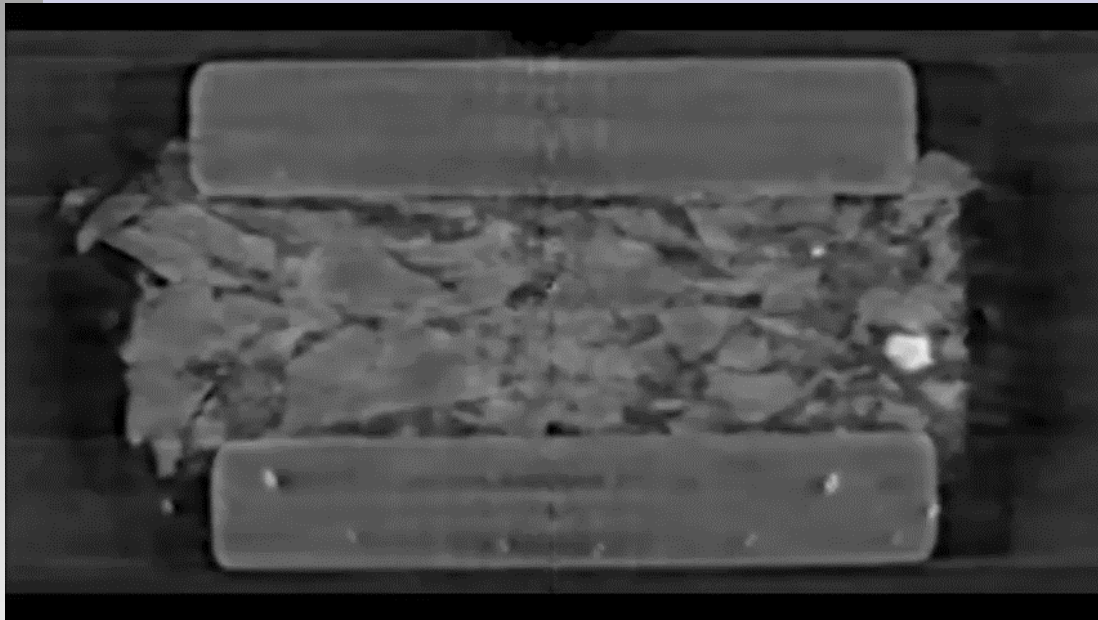
3 GPa - 1600 K



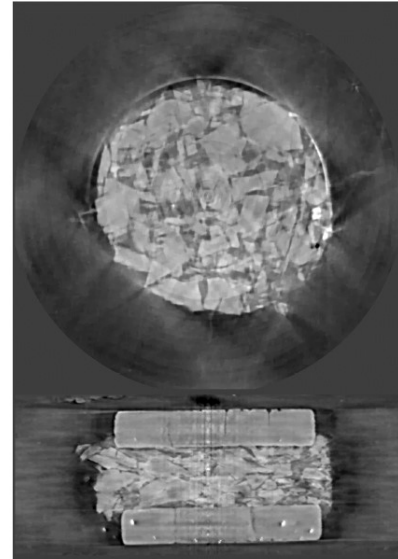
This study  
made it  
possible to  
**image** high P-  
T liquids

Coll. Eglantine Boulard

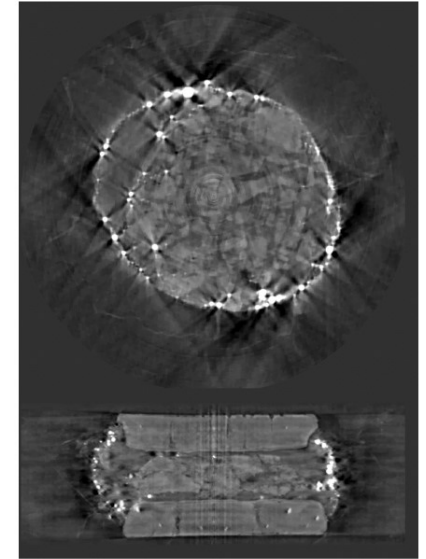
# Transport of magma under mantle conditions



3 GPa - 1200 K

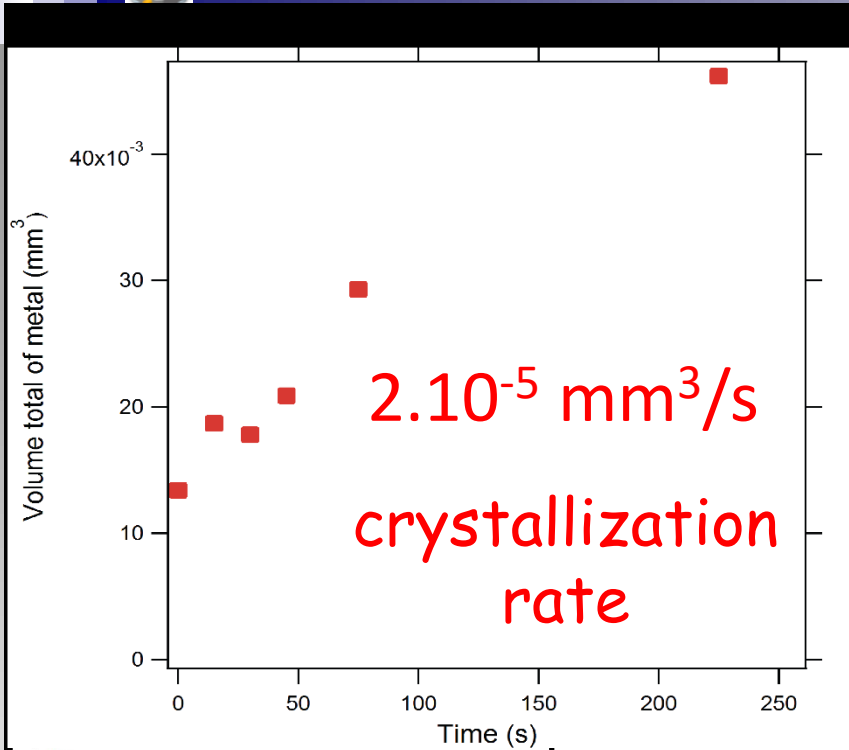


3 GPa - 1600 K

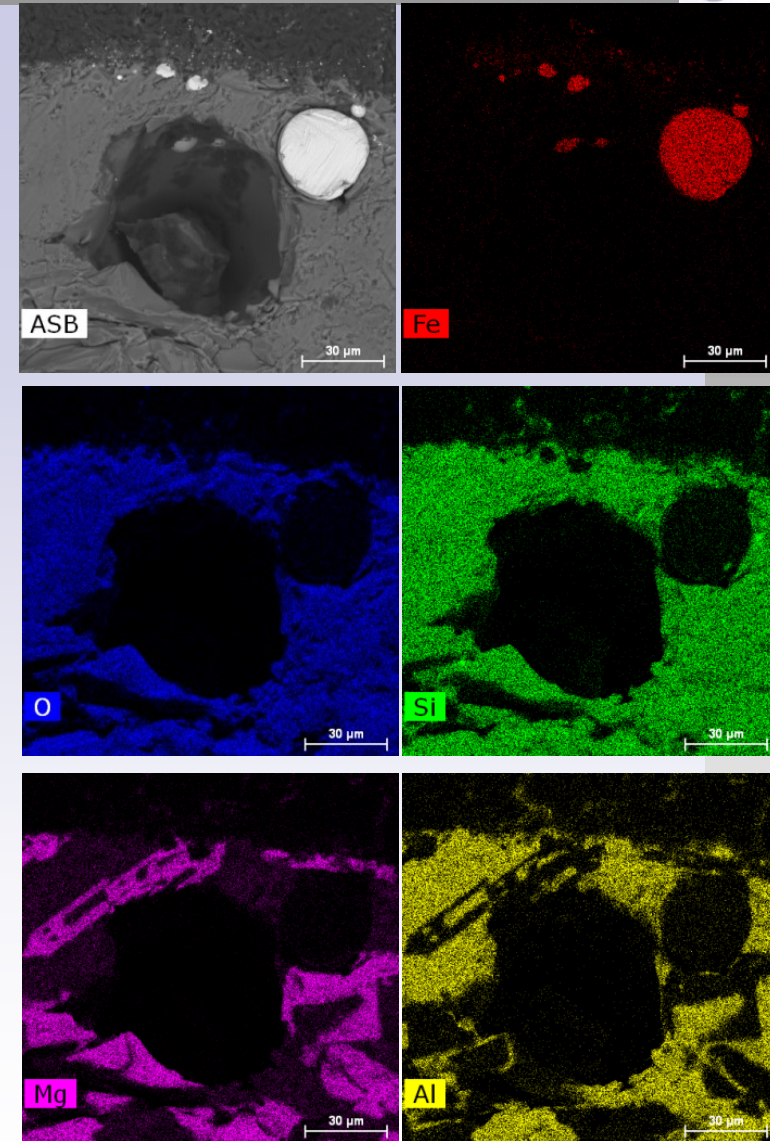


At 3 GPa and above 1100 K, strong deformation of the sample chamber is observed as its volume is reduced **drastically** due to **melting of the rhyolite material**. In a few minutes and with increasing temperature the melt **reorganized itself toward the border of the capsule**. When **temperature reached 1600 K**, we observed **rapid and progressive demixing of liquid iron from the melt** that precipitate **as iron metallic spherules**, probably due to reduction of iron in the melt by the graphite capsule. **The fast acquisition time allow us to demonstrate the possibility to measure the iron crystallization rate in situ**

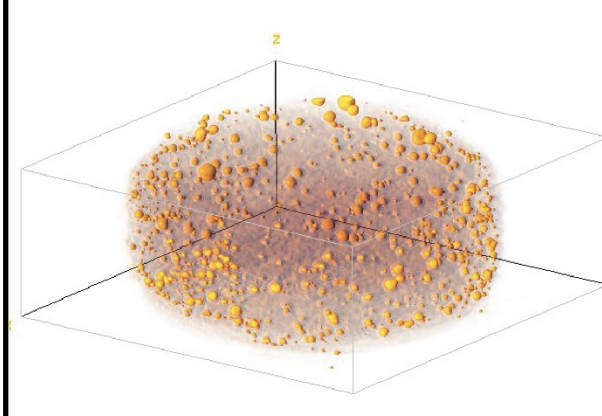




Coll. E. Boulard



SEM confirms that spherules are only Fe



We can obtain quantitative data with our new fast tomography technique at HPHT



New scientific possibilities coming with the ESRF upgrade will allow a better resolution (spatial and temporal) ! It will provide **new scientific opportunities for original and unique studies** of phase transitions, density, crystallization and deformation in **extreme P/T/Stress conditions**

**Absorption**  
Tomography

(large beam, fast)

- Density
- Microstructure
- Porosity
- Defects
- Phase content

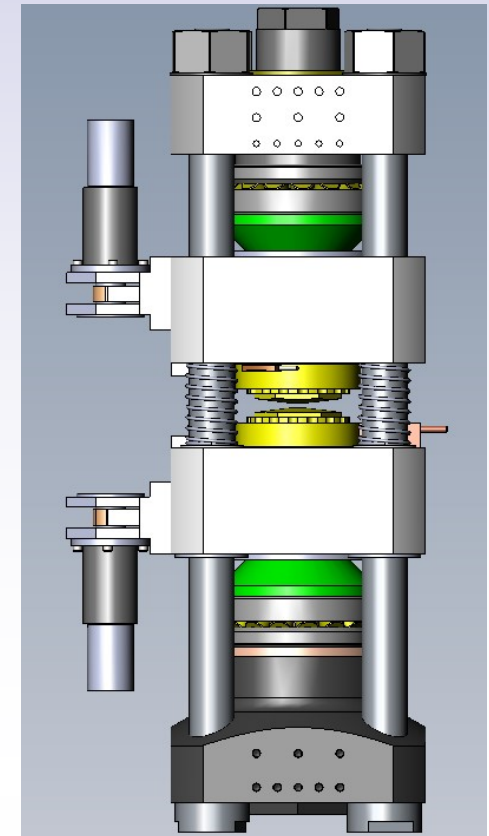
+

**Diffraction**  
Tomography

(micro beam, long)

- Phase distribution - content
- Orientation (texture)
- Crystallite size/microstrain
- Degree of crystallinity

**High**  
**P/T/stress**  
**PE cell**



# Bibliography

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S. Klotz, et al. *High Press. Res.*, 24, 219-223 (2004).

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## On tomography with PE press :

J. Philippe et al. (2016) Rotating tomography Paris-Edinburgh cell: a novel portable press for micro-tomographic 4-D imaging at extreme pressure/temperature/stress conditions, *High Pressure Research*, doi:/10.1080/08957959.2016.1221951.

M. Alvarez-Murga et al. (2017) Development of synchrotron X-ray micro-tomography under extreme conditions of pressure and temperature, *Journal of Synchrotron Radiation*, 24, 240-247.

M. Berg et al. (2018) Rapid core formation in terrestrial planets by percolative flow: in-situ imaging of metallic melt under high pressure/temperature conditions. *Frontiers in Earth Science*, 6, 77.

E. Boulard et al. (2018) High-speed tomography under extreme conditions at the PSICHE beamline. *J. Synchrotron Rad.* 25, 818-825.

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**Univ. of Edinburgh** : G. Bromiley, M. Berg (and **Cambridge** : S. Redfern)

**ESRF** : M. Mezouar, M. Alvarez-Murga, S. Bauchau, G. Garbarino

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**DIAMOND** : R. Atwood

**SOLEIL** : N. Guignot, A. King, J.P. Deslandes, J.P. Itié

